

IAEA CRP “Beryllium” final meeting, Vienna 2016

Beryllium-related PSI-studies at IPP Garching

Thomas Schwarz-Selinger

Plasma Edge and Wall Division

- Experimental equipment used/still in use for Be related studies

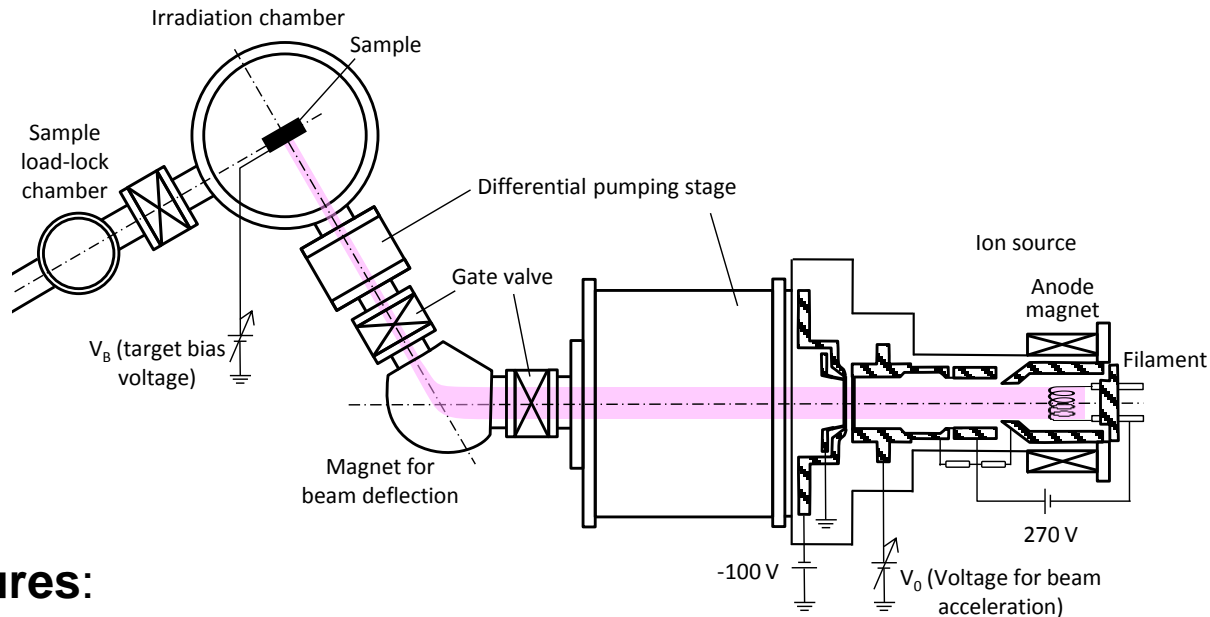
- Recent projects:
 - D retention in and release from Be mixed materials
 - H/D isotope exchange experiments
 - Cavity probes in PISCES and JET
 - JET marker tiles and wall inserts
 - Cross sections for ion beam analysis
 - Modelling material migration

- List of recent publications

Experimental facilities at IPP Garching used for Beryllium studies

- UHV preparation chamber (Be thermal vapour deposition) connected to dual ion beam setup and in-situ Photoelectron Spectroscopy (XPS)
- High current ion beam setup “HCS” with in situ mass balance
- Thermal desorption spectroscopy experiment “TESS”
- “RKS” ion beam analysis setup with Rutherford Backscattering Spectroscopy, Nuclear Reaction Analysis and Elastic Recoil Detection Analysis
- “SAK” ion beam analysis setup with air tight glove box equipped with Rutherford Backscattering Spectroscopy and Nuclear Reaction Analysis

D implantation with mass separated ion beam

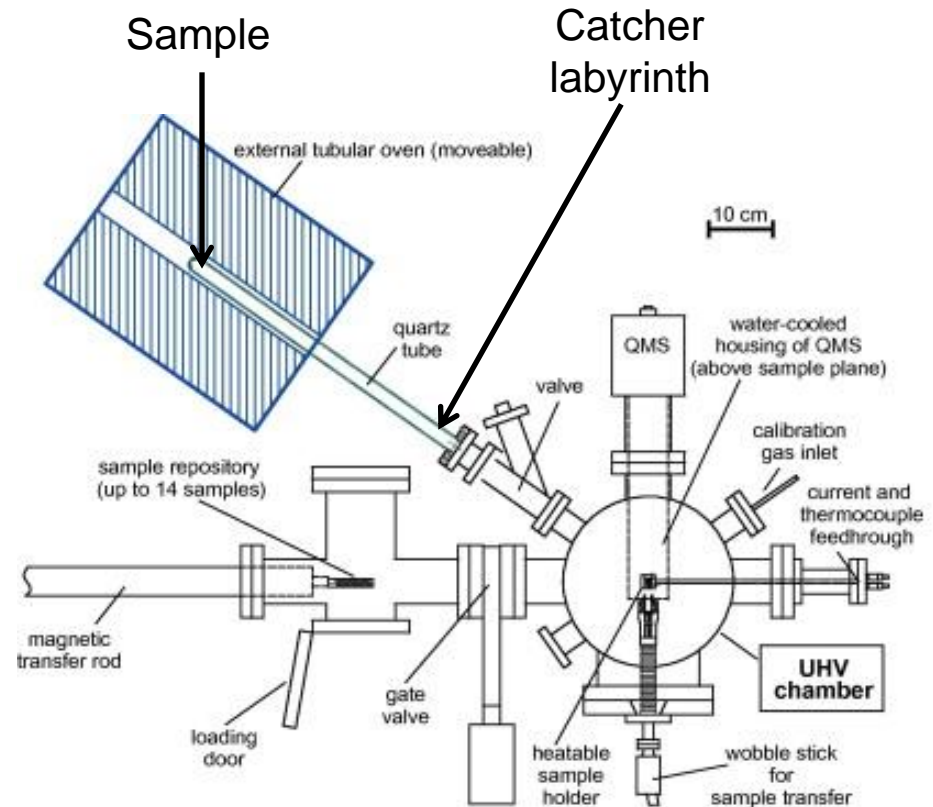



Features:

- D_3^+ ; D energy 30 eV/D – 2 keV / D
- D flux / fluence: $\sim 10^{19}$ D/m²s / $< 10^{25}$ D/m²
- Sample heating by e-bombardment < 1000 K
- Decommissioned, upgraded and duplicated in 2015/2016
- New in-situ mass balance in operation in autumn 2016, in-situ IBA > 2017
- No Be related work anymore

TPD or long term annealing at the TESS facility

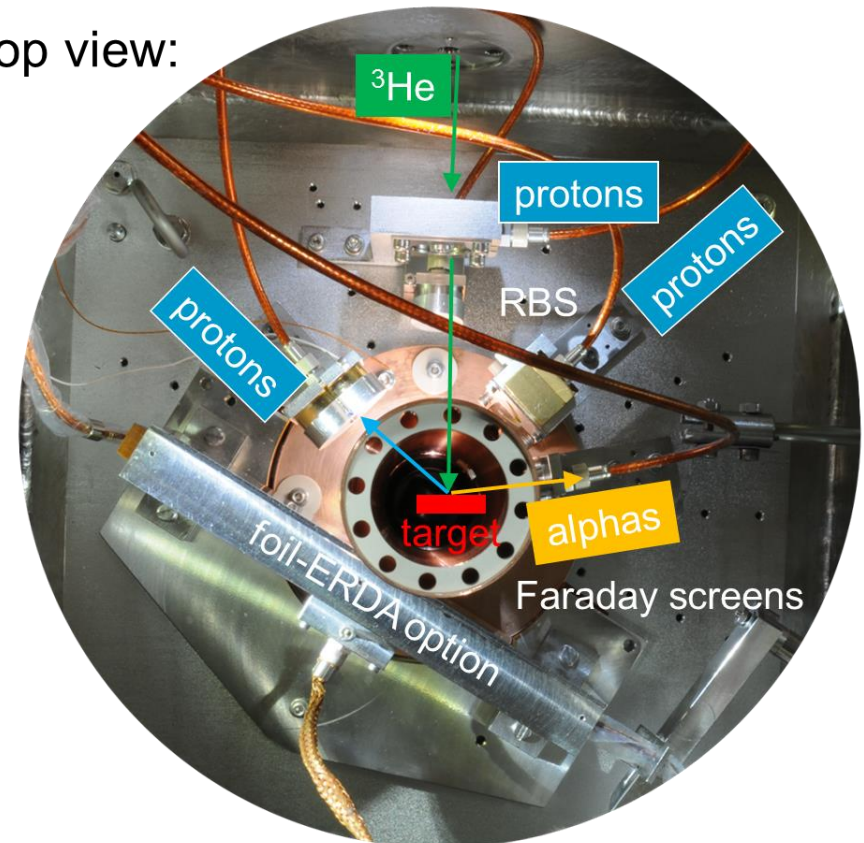
- Used for “Normal” TPD of Be mixed TVA layers provided by National Institute for Laser, Plasma and Radiation Physics, Bucharest and implanted at “IPP HCS”
- Typically sample heated up to 1000 K with a ramp rate of 0.25 K/s.
- UHV and tubular oven allows long-term outgassing for days and weeks
- Be safety: Dedicated quartz tube for Be studies with “catcher labyrinth”
- Operational but Be activities on TVA thin films stopped in 2014 due to risk of flaking



-  built their own TDS setup

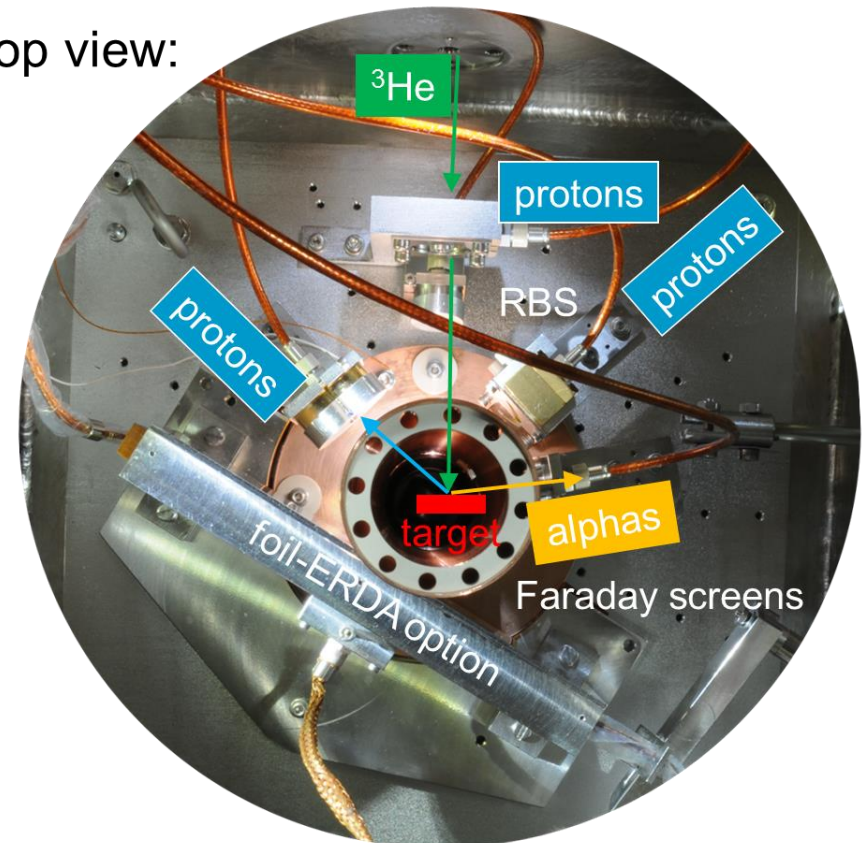
- In addition to
 - RBS and
 - NRA also
 - Elastic Recoil Detection Analysis (ERDA) for H, D and He detection
- Only “dust free samples”

Top view:



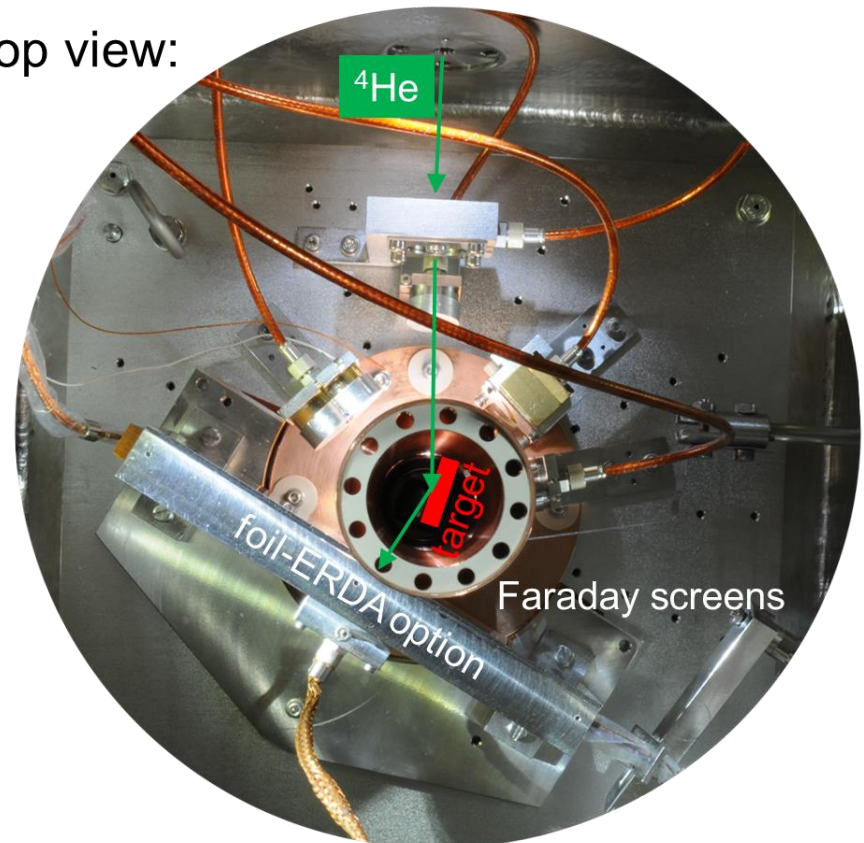
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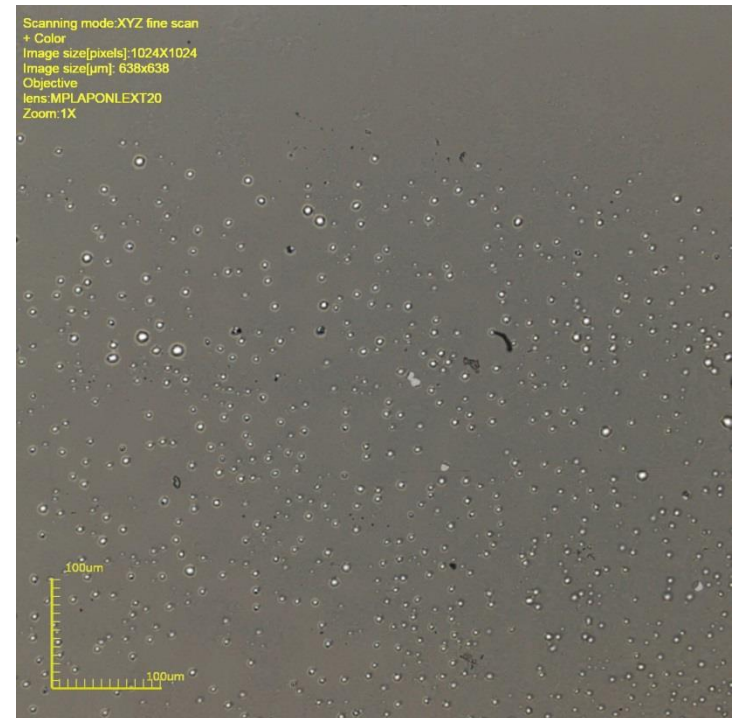
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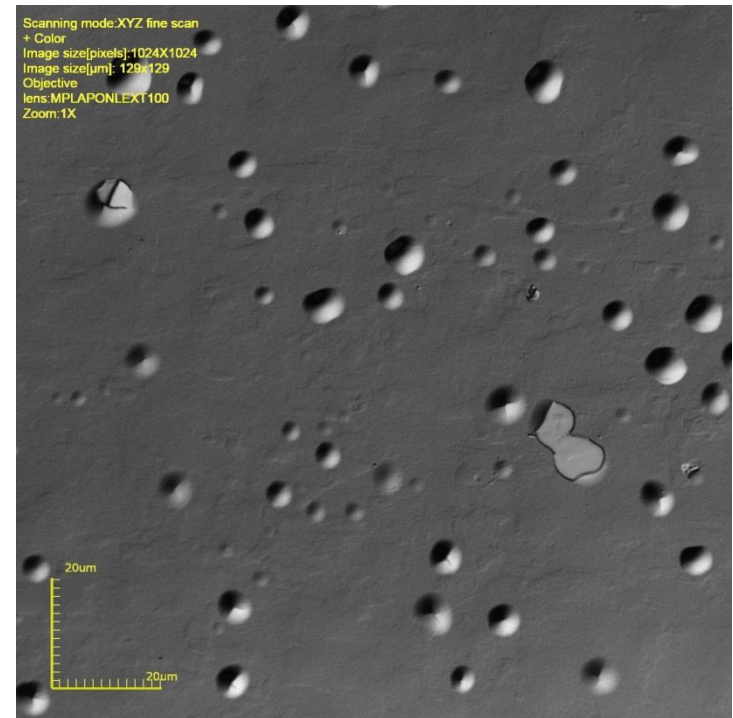
- In addition to
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- Dangerous with thin films!

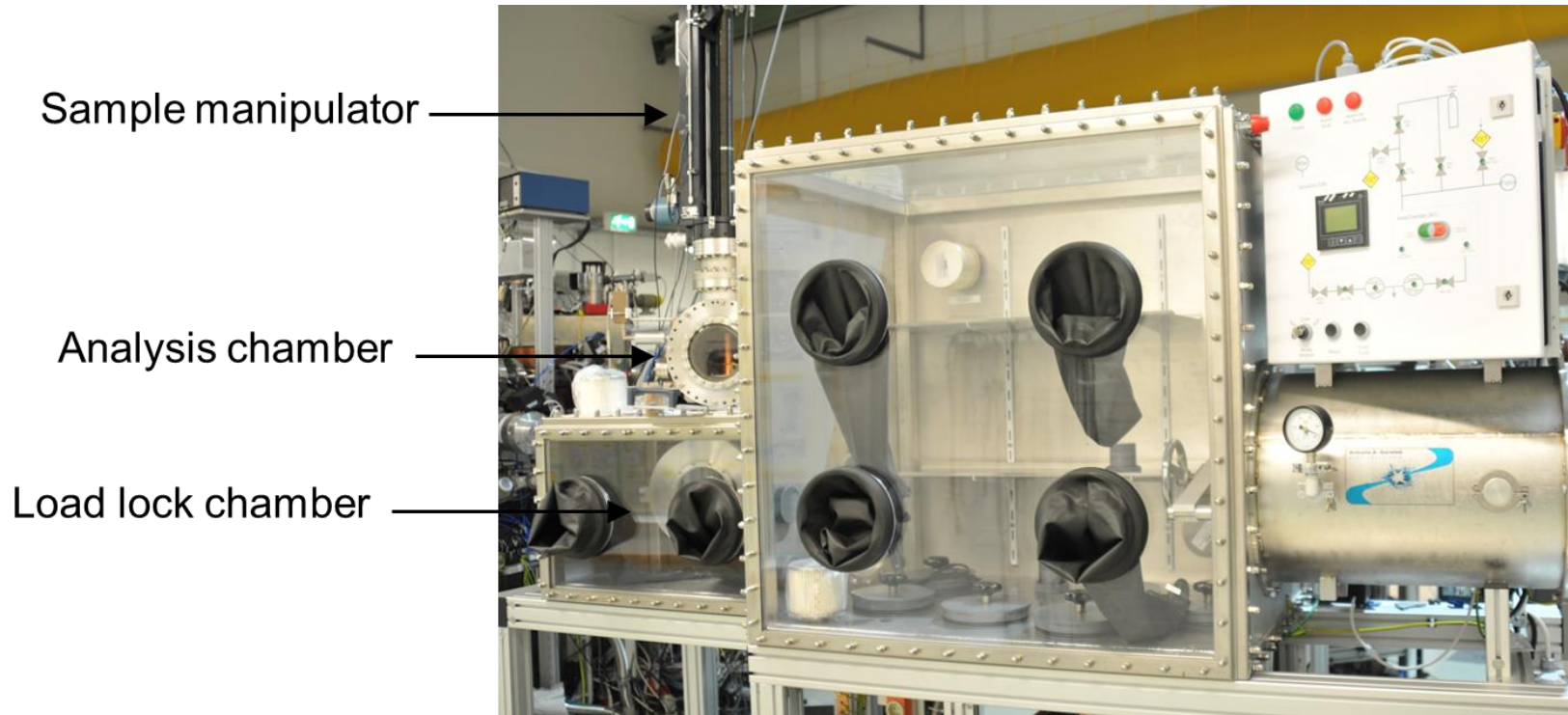
≈ 250 nm Be on W



- In addition to
 - RBS and
 - NRA also
 - Elastic Recoil Detection Analysis (ERDA) for H, D and He detection
- Only “dust free samples”
- Dangerous with thin films!

≈ 250 nm Be on W





- Dedicated chamber for Be contaminated Jet samples (Tritium < 1 GBq)
- Rutherford Backscattering Spectroscopy (typically protons or Helium) and Nuclear Reaction Analysis (^3He for D, ^4He , p for N and D for ^3He)
- Regular swipe sampling

D retention in / release from Beryllium-related mixed material layers

- Work performed under framework of F4E contracts: F4E-OPE-080 / -347 “Deuterium retention and outgassing experiment”, towards an evaluation of the tritium removal operation in ITER (baking at 240 °C for the first wall / 350 °C for the divertor).
- Completed and published since the last CRP

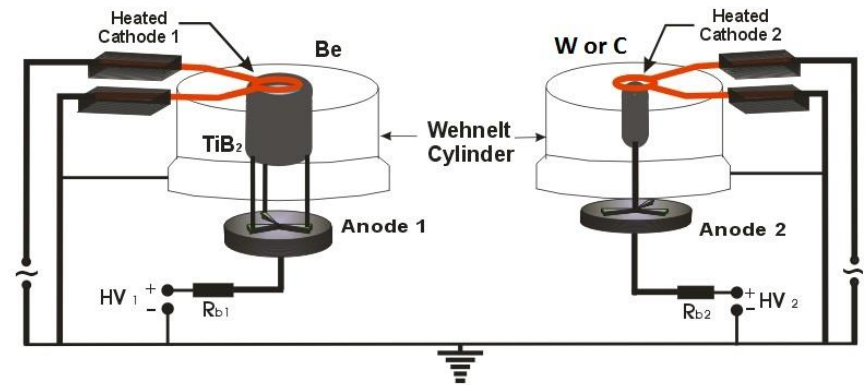
Work strategy

- Be containing samples to be investigated: **Be, Be-W, Be-C and Be-O** mixed layers prepared by **Thermionic Vacuum Arc (TVA)** deposition method at National Institute for Laser, Plasma and Radiation Physics, Bucharest.
- Layer characterization, **D ion implantations** and subsequent outgassing experiments at IPP Garching.
 - Rutherford Backscattering Spectrometry (RBS) using 2.0 MeV $^4\text{He}^+$ for the layer characterization.
 - D loading to the layers in the High Current Ion Source device.
 - D retention is analysed by **Nuclear Reaction Analysis (NRA)** using 800 keV $^3\text{He}^+$ ($\text{D}(^3\text{He}, \text{p})^4\text{He}$ reaction).
 - TDS and **long-term outgassing** experiments in TESS facility.

Layer preparation by Thermionic Vacuum Arc (TVA) deposition

National Institute for Laser, Plasma and Radiation Physics, Bucharest

- Be-W and Be-C mixed layers are prepared by simultaneous deposition of Be and W or C using dual target source configuration.
- Gas inlet system successfully developed - Be-O mixed layer can be deposited by oxygen injection during Be deposition.
- Each deposition batch typically consists of 15 **silicon** and 2 **graphite** substrates – Silicon is for the outgassing experiment and Graphite is for the layer characterization.



Schematic view of TVA setup (inside the vacuum chamber)

Layer thickness and compositions determined by RBS

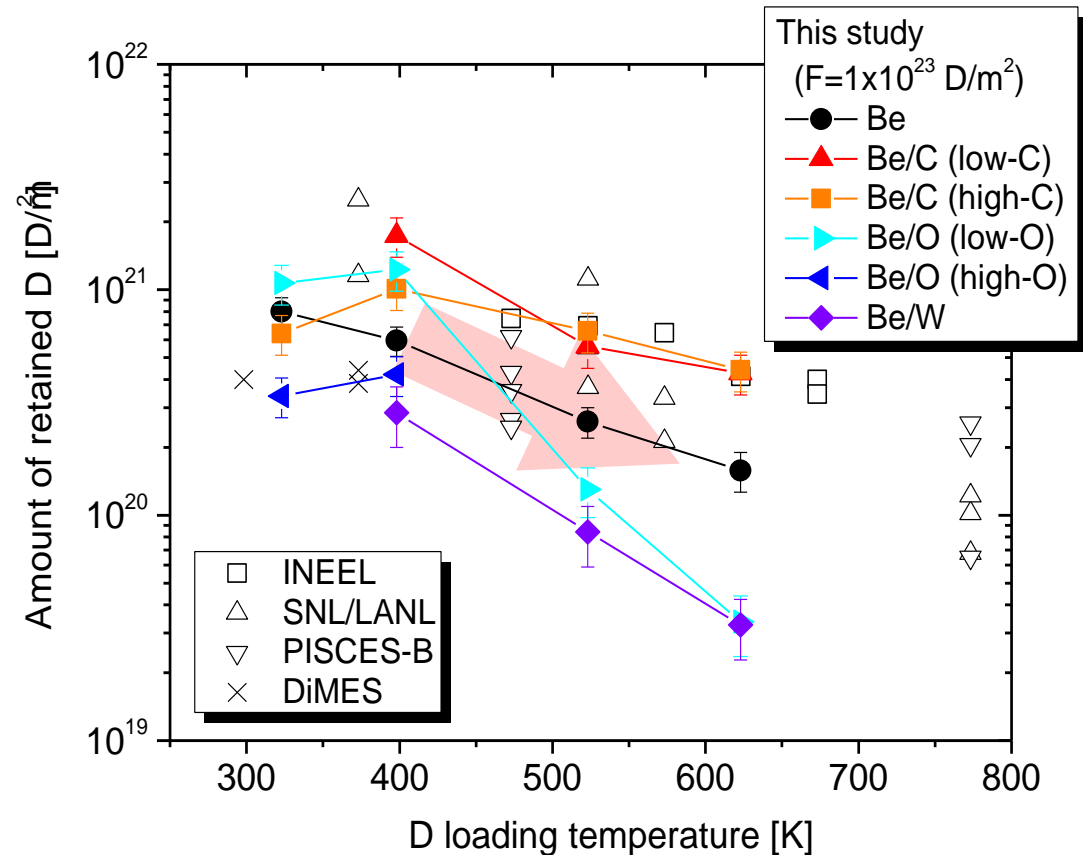
| Layer type | Layer areal density | Layer stoichiometry |
|----------------|--|---------------------------------------|
| Be | 6.9×10^{18} at./cm ² | (O impurity ≤ 1 at.%) |
| Be/W | 4.3×10^{18} at./cm ² | W: 6 ± 2 at.% (O: ~ 10 at.%) |
| Be/C (low C) | 6.2×10^{18} at./cm ² | C: 13 ± 1 at.% (O: ~ 7 at.%) |
| Be/C (high C) | 4.6×10^{18} at./cm ² | C: 50 ± 1 at.% (O: ~ 8 at.%) |
| Be/O (low O) | 4.1×10^{18} at./cm ² | O: $\sim 6 \pm 1$ at.% |
| Be/O (high O)* | 2.6×10^{18} at./cm ² | O: $\sim 50 \pm 1$ at.% |

Layer thickness 340 – 570 nm

D retention in Be-containing layers

Amount of D retention in Be-containing layers vs. implantation temperatures

- D retention in general decreases with the D loading temperature (above ~400 K).
- ✓ The absolute amount varies depending on the experimental condition.
- ✓ Reduction is limited in the case of Be-C mixed cases.
- ✓ Retention in the ITER first wall temperature range is expected to be $10^{20} - 10^{21} \text{ m}^{-2}$.

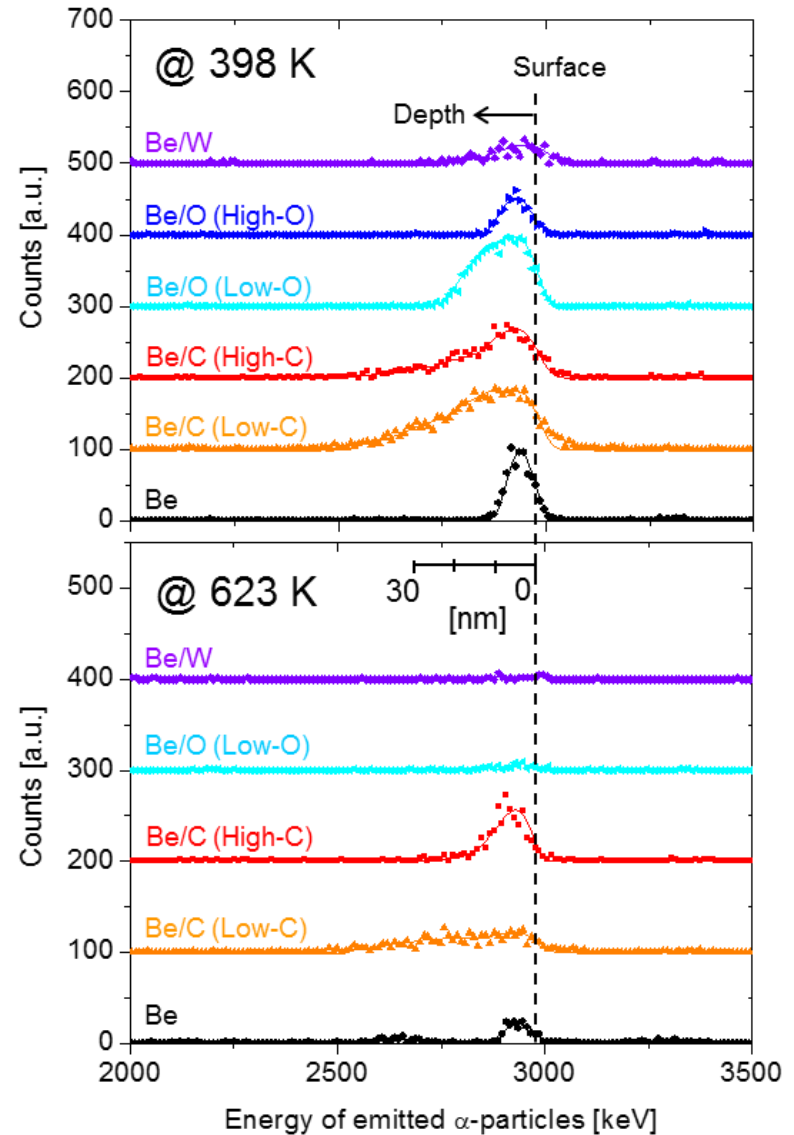


Amount of D retention in each Be-containing mixed material layer as a function of implantation temperature, together with some literature data of pure Be after plasma exposure obtained at SNL/LANL-TPE, PISCES-B, INEEL and DiMES for comparison.

D concentration in Be-containing layers

Depth profile of D concentration in Be-containing layers

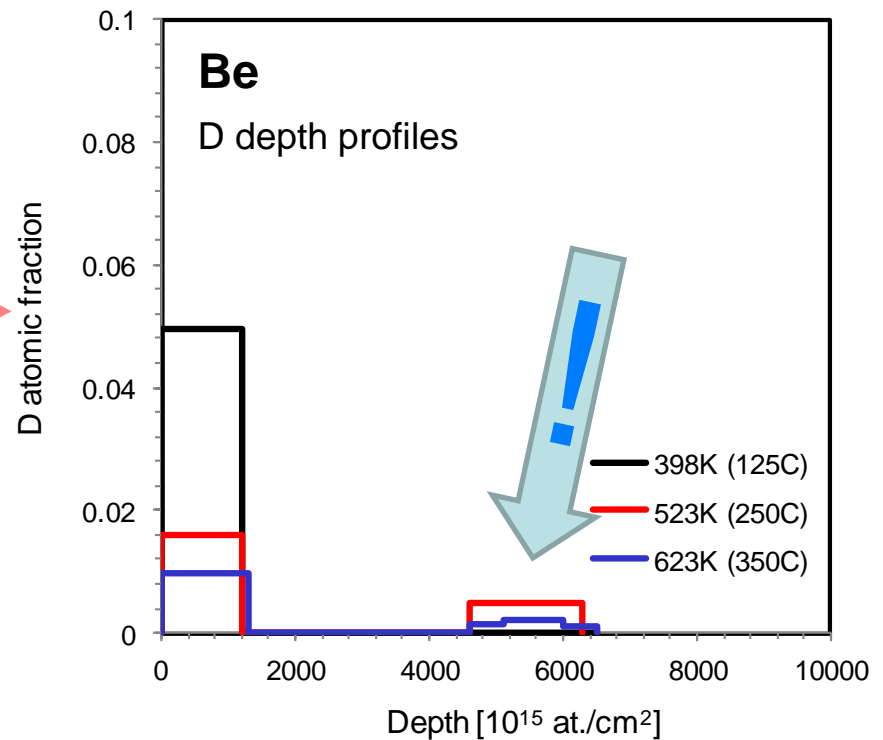
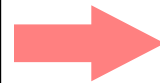
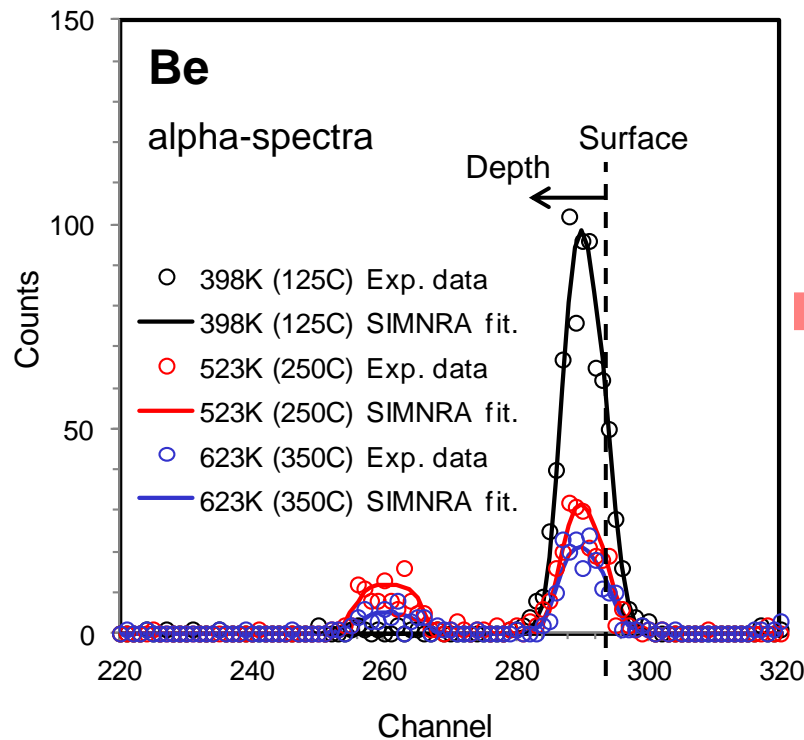
- Determination of depth profile of D concentration by analyzing the profile of alpha particles emitted from $D(^3\text{He}, \alpha)\text{H}$ reaction.
- C-rich D-C layer keeps relatively high D concentration even at high temperature – likely due to the trapping by C-D chemical bonds.
- Why different penetration depths?
Different diffusion or simply morphology?



D retention in mixed materials

Deuterium depth profile in Be-containing layers

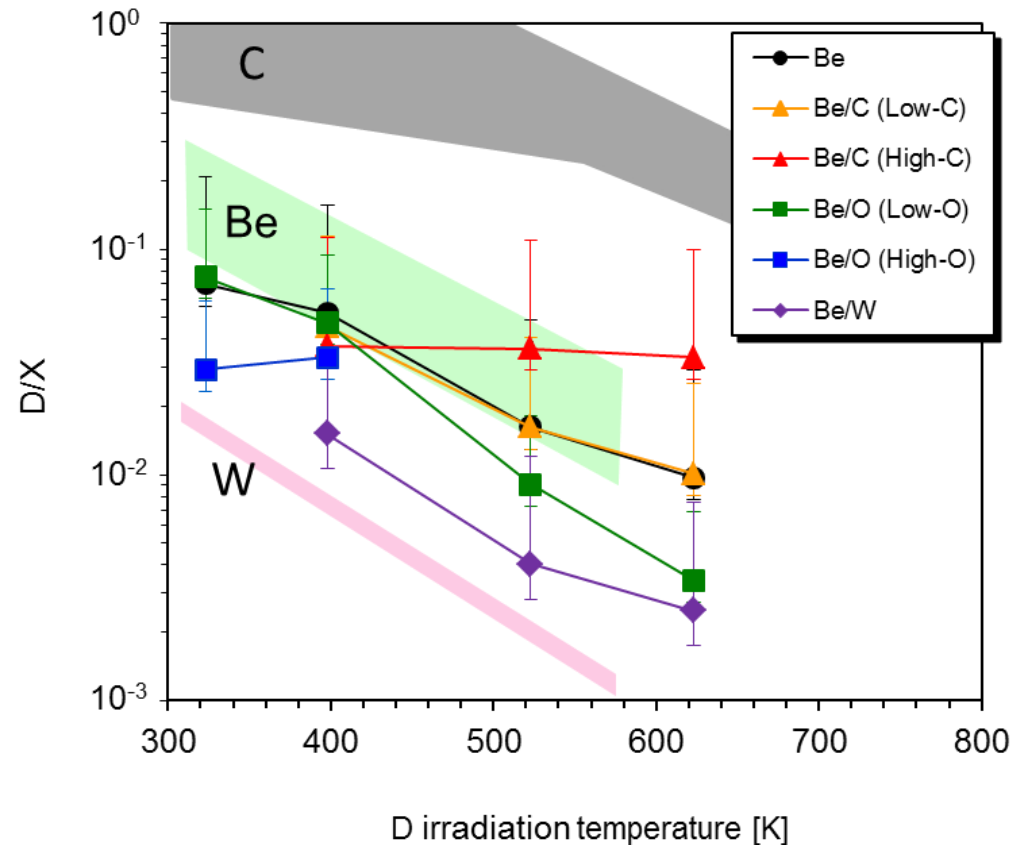
Determination of depth profile of D concentration by analyzing the profile of alpha particles emitted from $D(^3\text{He}, \alpha)\text{H}$ reaction



- D is dominantly retained in the implantation layer
- At elevated T, **some D migrates to the interface**

D concentration in Be-containing layers vs. implantation temperatures

- D/X in Be layer agrees with the data obtained from Be-D codeposition.
- C-rich D-C layer keeps relatively high D/X even at high temperature – probably due to the trapping by C-D chemical bonds.
- Low fraction of W, O impurity can **reduce** the D concentration compared to pure Be.



Maximum D concentration (shown as D/X) in each Be-containing mixed material layer as a function of implantation temperature. The areas labeled as C, Be or W indicate results from a data compilation of experimentally-obtained D/X values for the D concentration in “codeposition layers”

D retention in and release from Be-containing mixed material layers were investigated in view of ITER tritium removal schemes:

- Retention in the mixed layer decreases with increasing the D loading temperature, but the tendency has some variation depending on the admixed element and its amount.
- Admixed impurities seems not to lead to dramatic changes of absolute retention. Only for high C content case total retention and the release temperature increase.
- Removal efficiency by the wall baking procedure(s) will be limited if the wall temperature is already comparable to the baking temperature.

Related publications:

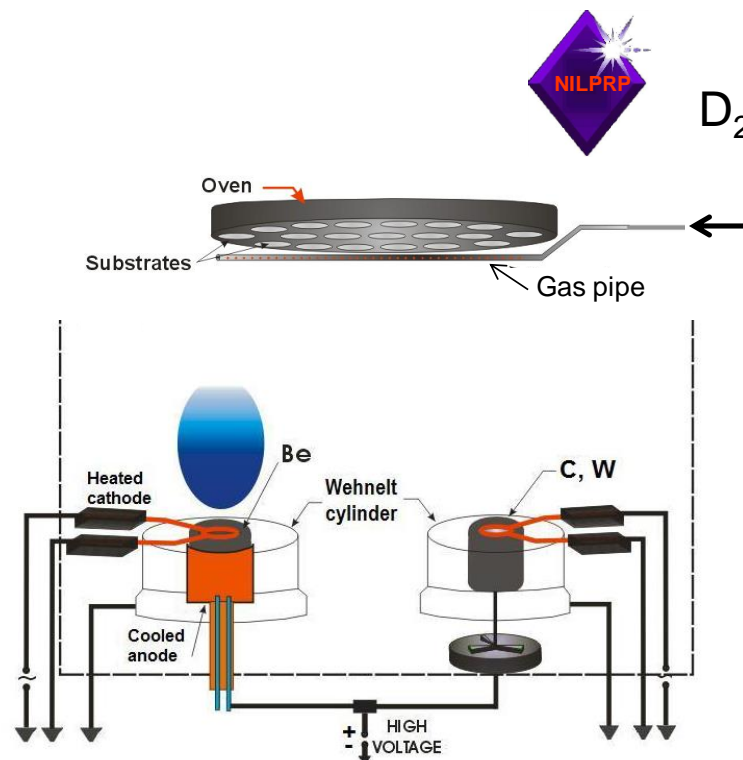
K. Sugiyama, J. Roth, A. Anghel C. Porosnicu, M. Baldwin, R. Doerner, K. Krieger, C.P. Lungu., J. Nucl. Matter. 415 (2011) S731

K. Sugiyama, C. Porosnicu, W. Jacob, J. Roth, Th. Dürbeck, I. Jepu, C.P. Lungu, J. Nucl. Matter. 438 (2013) S1113

K. Sugiyama, C. Porosnicu, W. Jacob, I. Jepu, C.P. Lungu, Nuclear Materials and Energy 6 (2016) 1

Preparation of “codeposition” by TVA deposition method at MEDC/NILPRP

- ✓ Layer deposition with TVA under D_2 gas atmosphere.
 - Analysis of N and D containing TVA films for EFDA/EUROfusion tasks
 - Mixed success:
 - D contents $\ll 10\%$
 - N containing films had more O than N
 - Very promising results with recent magnetron sputter device (as initially done shown by PISCES lab)
 - Analysis at other European labs (JSI Ljubljana, IST Lisbon)



D/H - Isotope exchange experiments bulk beryllium and Be:H and Be:D thin films

- Work performed at PISCES and IPP Garching
(T. Schwarz-Selinger, D. Nishishima, R. Doerner, unpublished)
- Work performed at PISCES, IPP Garching, MePHI, CEA
(D. Kogut et al. Phys. Scr. T167 (2016) 014062 (6pp))

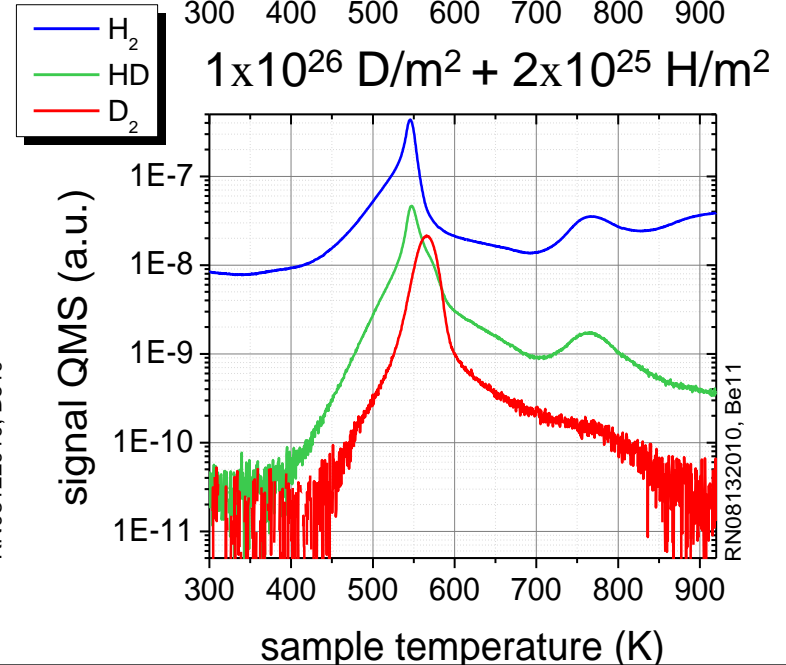
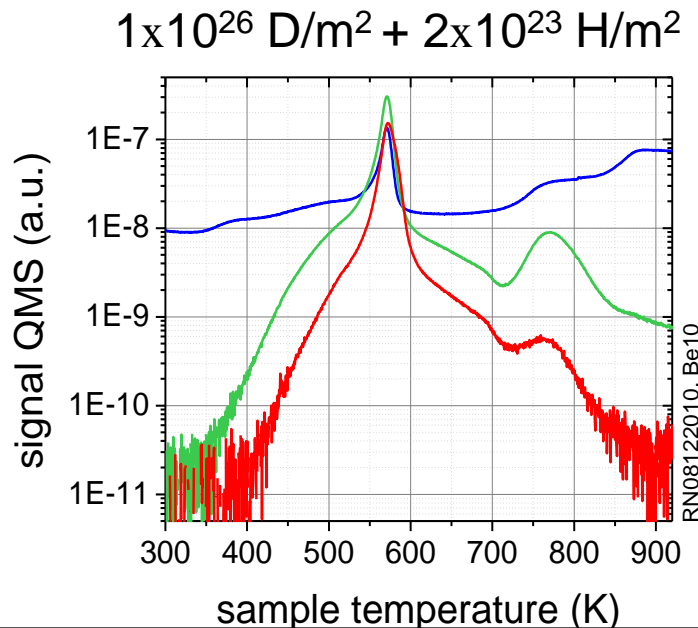
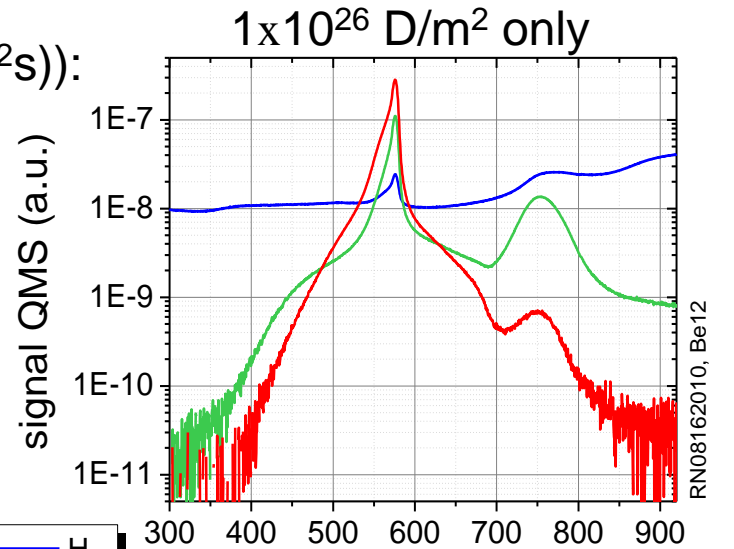
- In general: hydrogen retention in beryllium is even less understood as in tungsten
 - material grade (codeposits vs. bulk)
 - temperature
 - D energy
 - flux
 - fluence
 - seeding (He, N, Ar, ...)
- Investigations are hampered by
 - its toxicity
 - its reactivity
 - reduced IBA depth resolution and sensitivity
- Strategy:
 - exchange experiments in plasma-sprayed Be targets
 - effectiveness in thick codeposited layers?

} „de Temmerman scaling“: Nucl. Fusion 48+49 (2008)

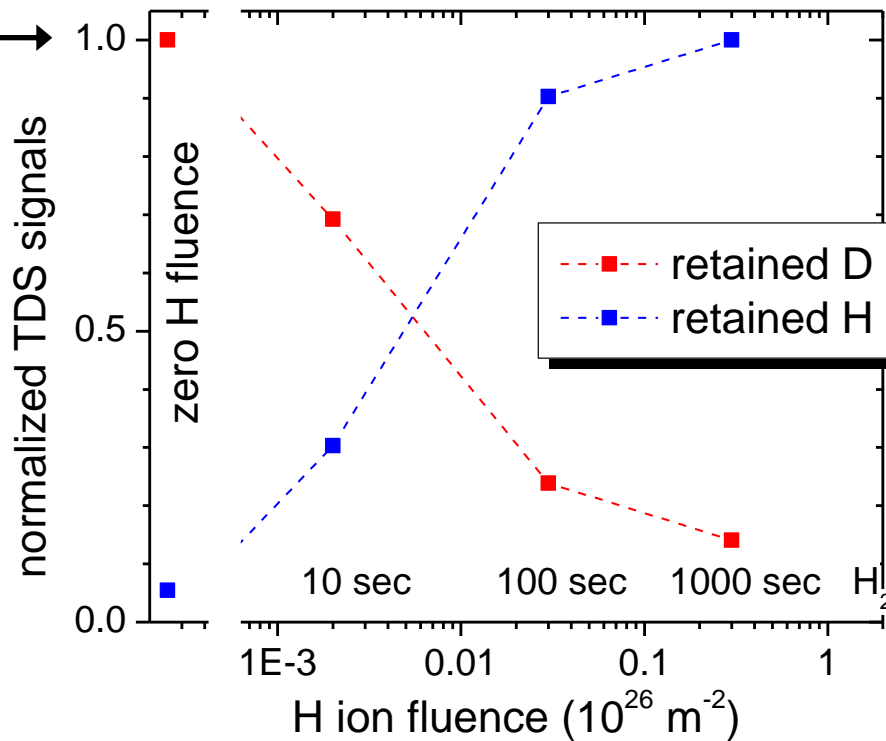
To check its applicability as fuel removal method

Divertor simulator PISCES-B (ion flux $2 \cdot 10^{22}$ D/(m²s)):

- sequence: D loading + pumping + H exchange
- bias: -100V (< 70eV), 330K
- polycrystalline Be target + Be coated wall
- TDS of samples (+OES)
- preferential depletion of binding states? (bubbles?)



$1.6 \cdot 10^{21} \text{ D/m}^2$ →



TDS

⇒ complete exchange: $\approx 10^{25} - 10^{26} \text{ ions/m}^2$

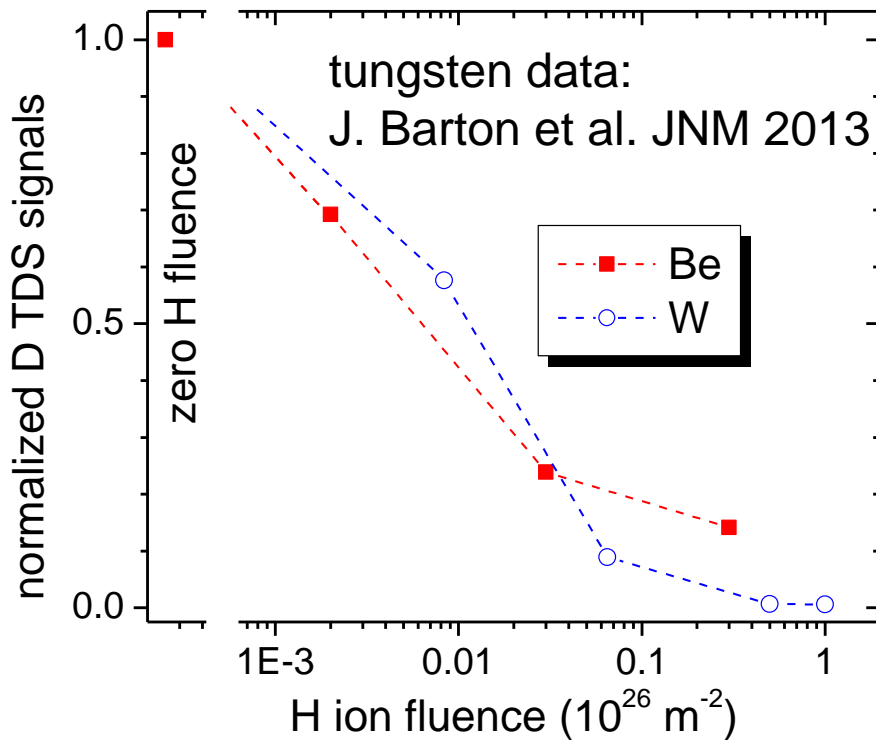
Φ_{exchange} for Be \gg than for W?

issues: - wall effect / recycling?

- D in depth?

- sputtering?

- ITER relevant: codeposits

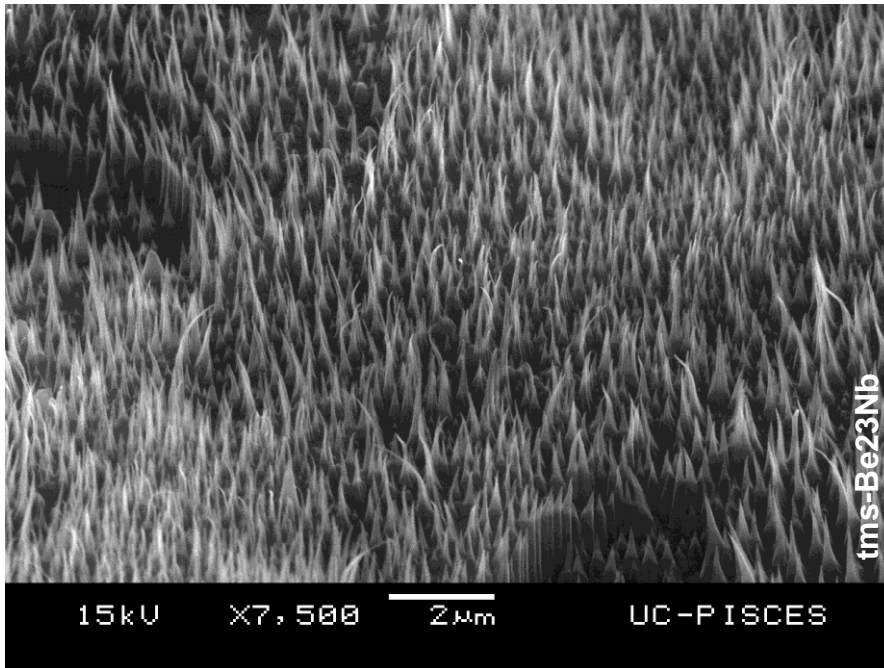


similar behavior in PISCES for Be and W!

Φ_{exchange} for Be \gg than for W?

issues:

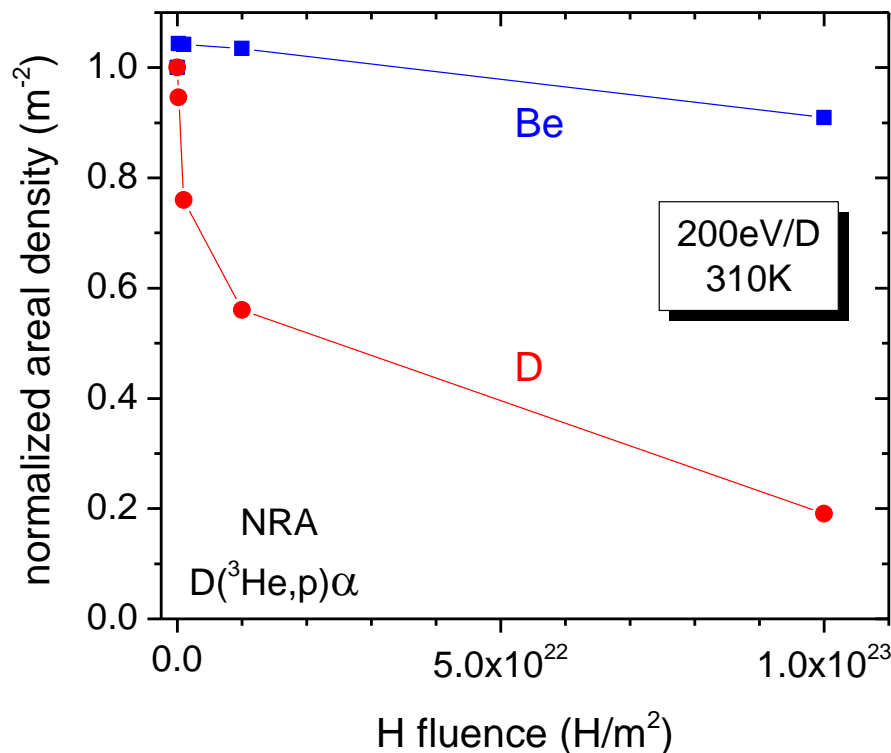
- wall effect / recycling?
- D in depth?
- sputtering?
- ITER relevant: codeposits



evolving surface morphology
does not allow to judge about that!

Better defined experiment for ion beam analysis (NRA for D + Be; ERD for H + D):

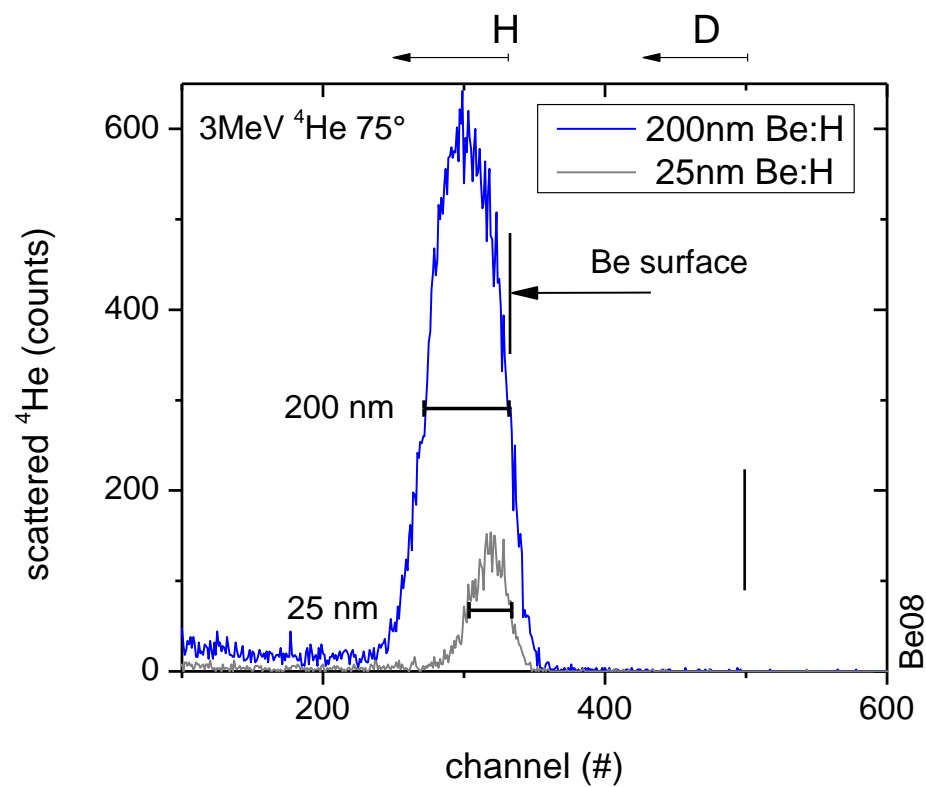
- 50 nm: PISCES-B Be:D codeposit (D/Be = 0.25) on polished tungsten
- sequentially exposed to H ion beam in Garching „HCS“



- exchange in near surface layers in Be as
 - effective
 - fast as in W!
- also for thick codeposits?

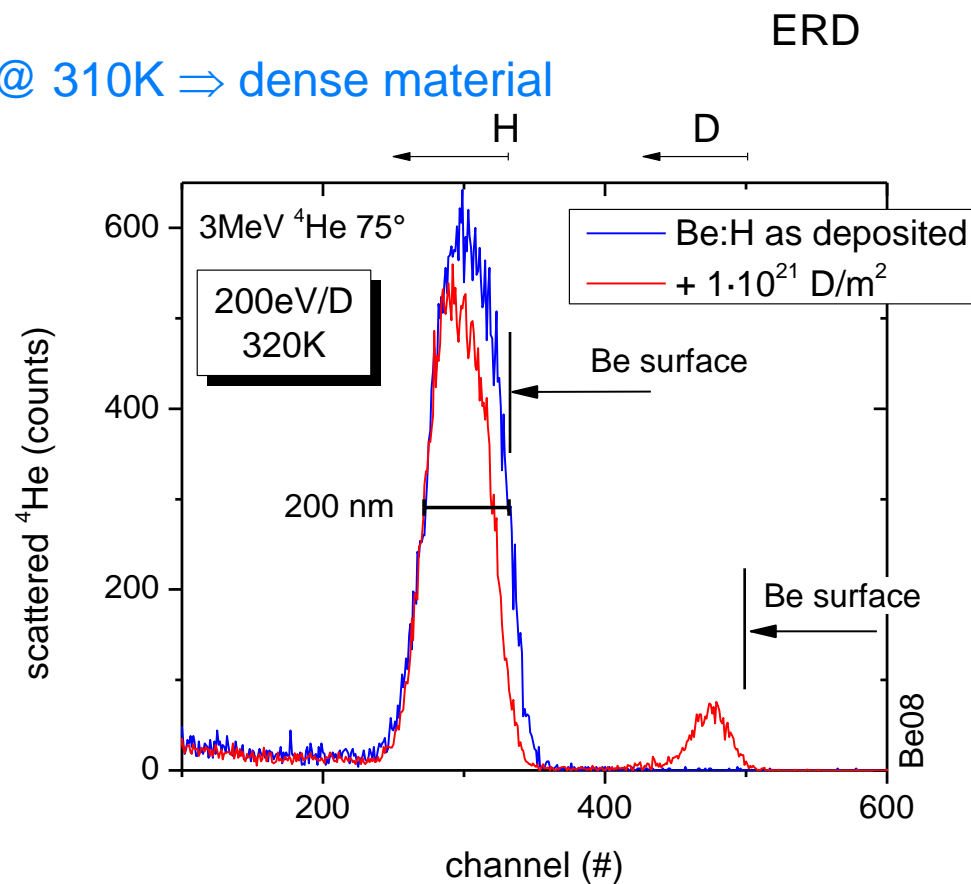
- ≈ 200 nm PISCES-B Be:H codeposit on polished tungsten (H/Be=0.2)

Elastic Recoil Detection (ERD)



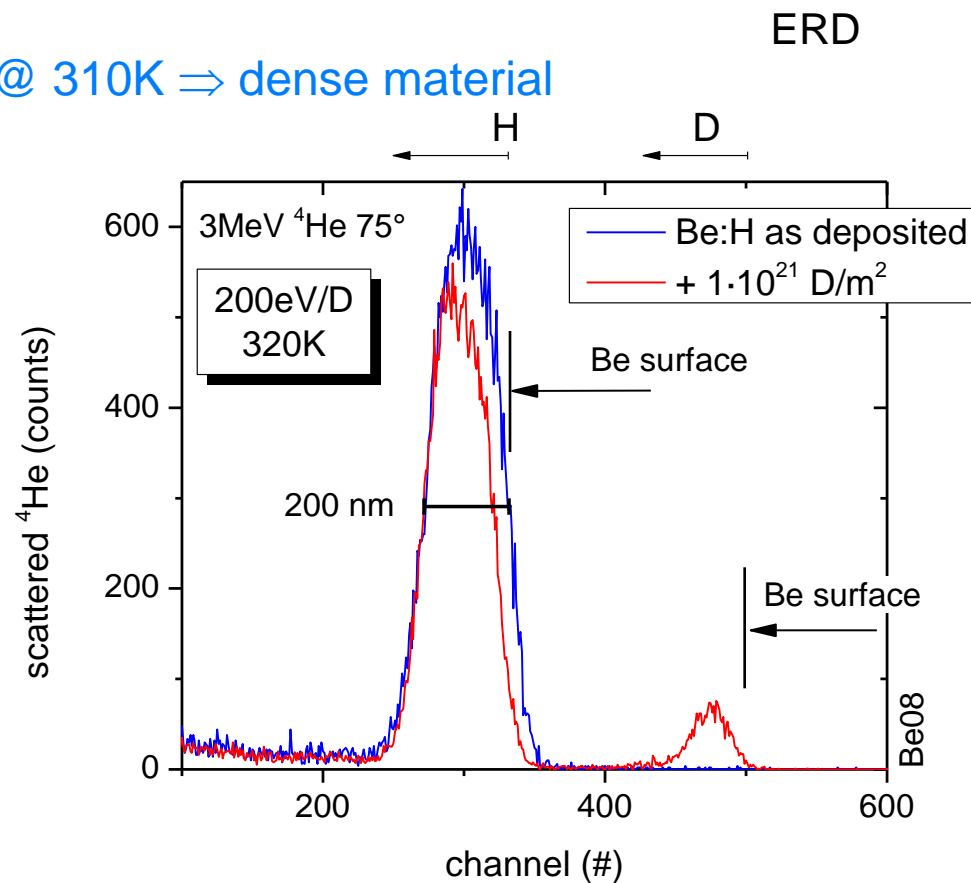
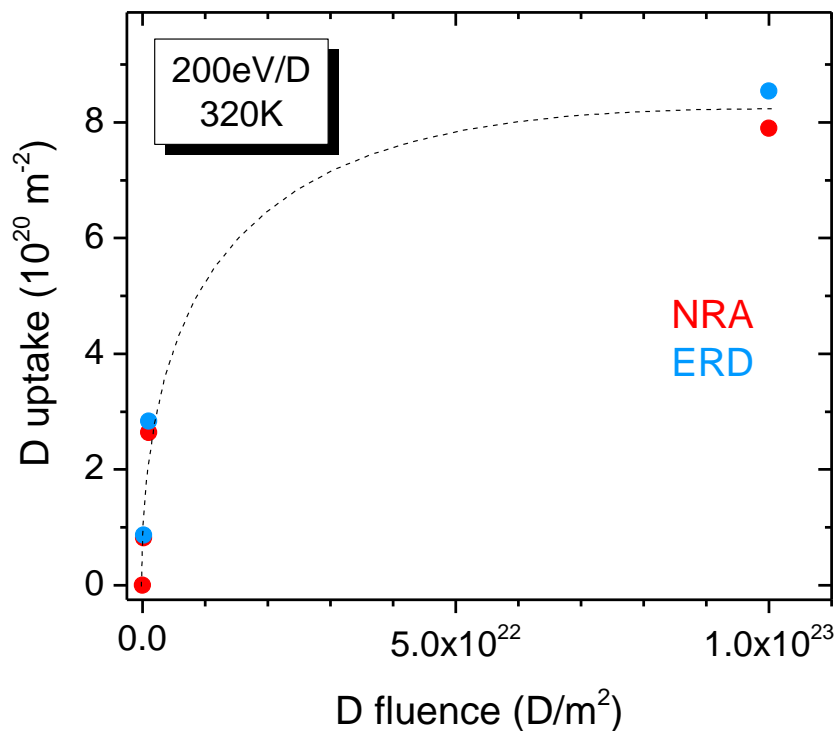
- ≈ 200 nm PISCES-B Be:H codeposit on polished tungsten (H/Be=0.3)
- exposed to D ion beam

\Rightarrow only surface near exchange @ 310K \Rightarrow dense material

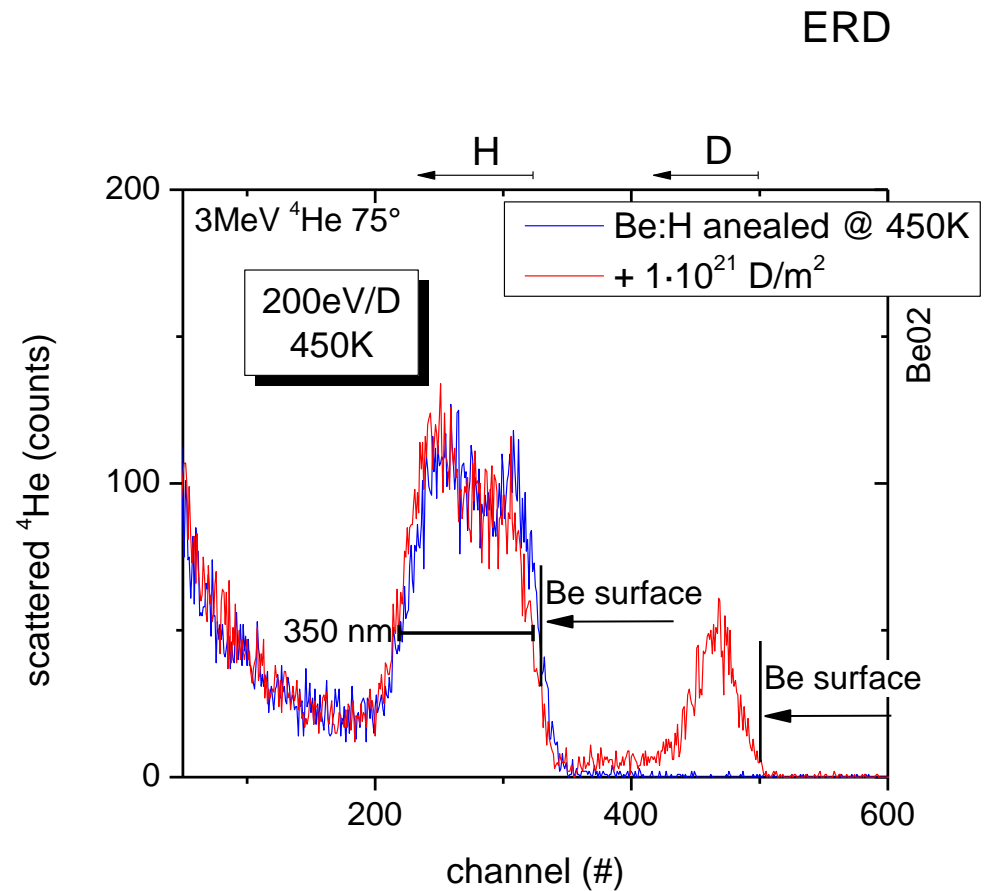


- ≈ 200 nm PISCES-B Be:H codeposit on polished tungsten (H/Be=0.3)
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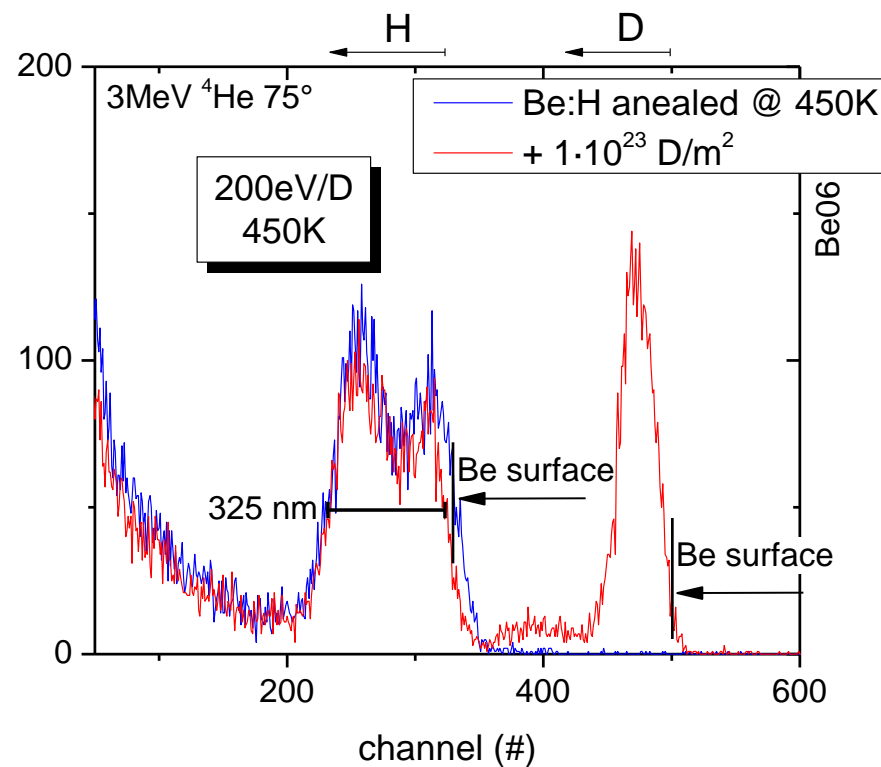
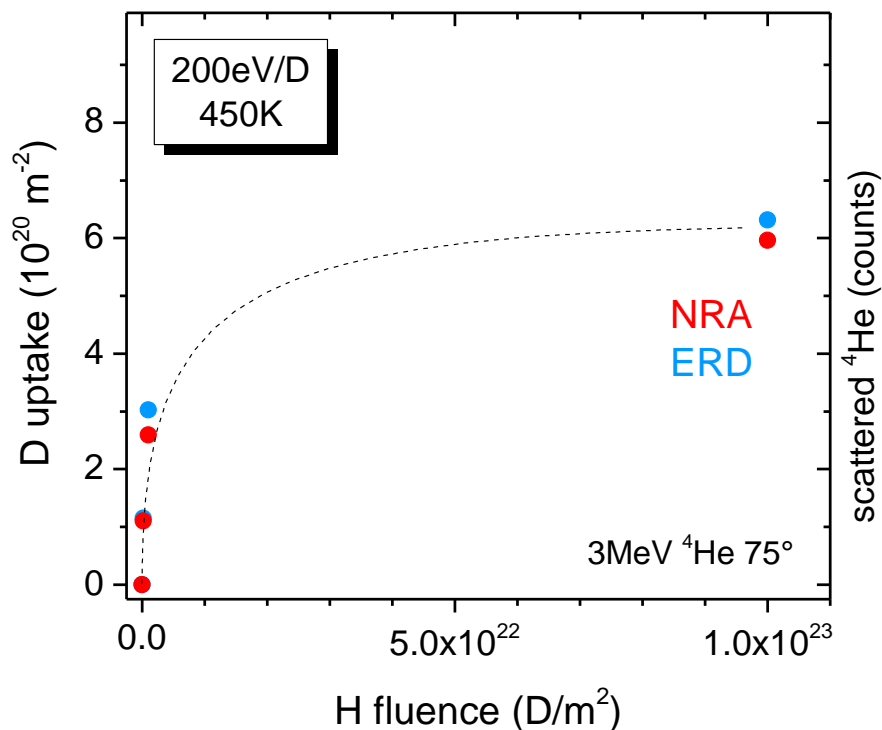
- ≈ 350 nm magnetron sputtered Be:H codeposit on polished tungsten (H/Be=0.04)
- exposed to D ion beam



- ≈ 325 nm magnetron sputtered Be:H codeposit on polished tungsten ($D/Be=0.04$)
- exposed to D ion beam

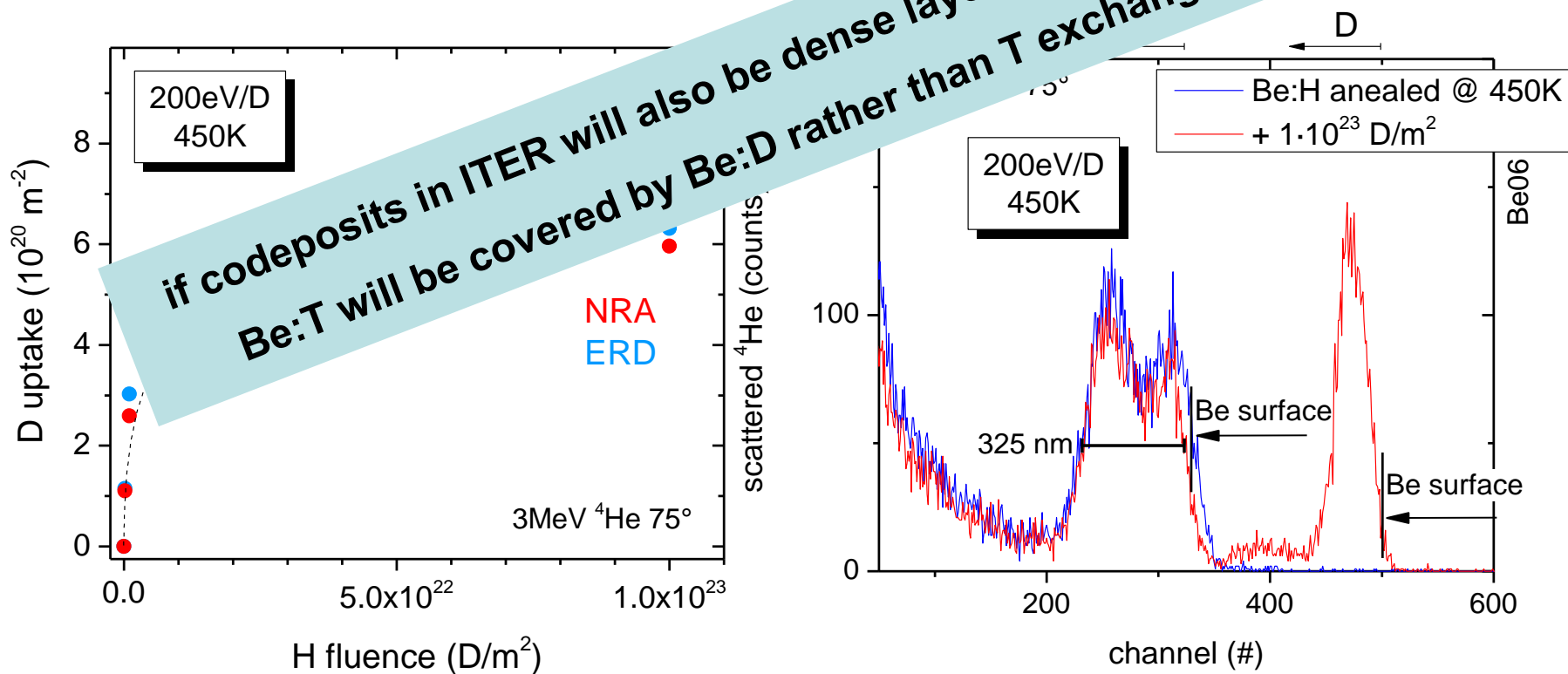
\Rightarrow uptake > release! + indication for deeper penetration?

ERD



- ≈ 325 nm magnetron sputtered Be:H codeposit on polished tungsten (D/Be=0.04)
- exposed to D ion beam

\Rightarrow uptake > release! + indication for dense layers (no open porosity)



**if codeposits in ITER will also be dense layers (no open porosity)
Be:T will be covered by Be:D rather than T exchanged by D!**

- More high temperature experiments and experiments on bulk Be were planned but Be operation ended at HCS
- Ideal experiment:
H/D exchange at the same ion energy and sample temperature at which D/H was implanted during growth to see quantitatively the exchange rather than a new equilibrium state

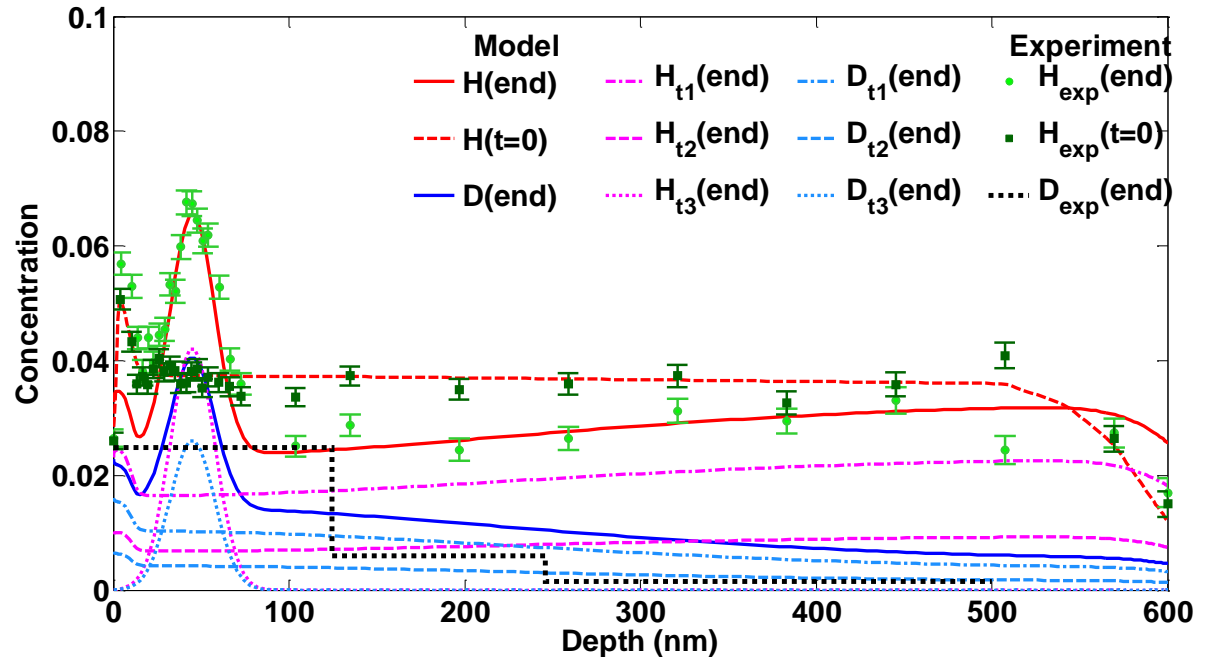
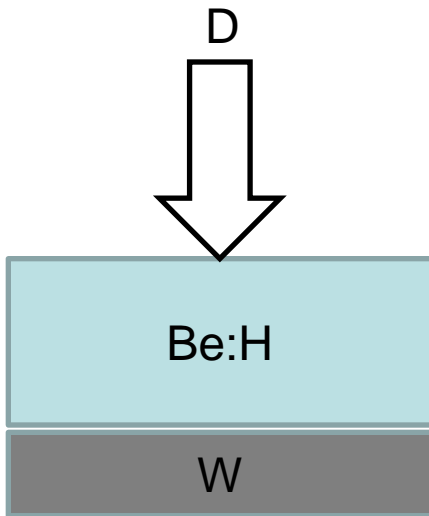
Initiated by D. Kogut and D. Douai CEA

Same procedure as in 2014:

- Be:H preparation on rough W substrate by magnetron sputtering at PISCES lab
- Exposure to D ion beam at MEFPhI
- Analysis at
 - Ruhr Uni Bochum by ^{15}N NRA for H depth profile and at
 - IPP Garching D content with NRA and exchange with ERDA

Drawback:

- High energy D implantation (2.5keV/D compared to $< 100\text{eV/D}$ during growth)
- Inhomogenous implantation spot
- Limited number of samples



- Distinct H peak in the D implantation zone!
- DITMIX diffusion trapping model with Baldwins input data describes observation assuming trap creation by high energy D implantation (2.5keV/D compared to < 100eV/D during growth)
- See D. Kogut et al. Phys. Scr. T167 (2016) 014062 (6pp)

D/H isotope exchange experiments

- Exchange of H by D (or vice versa) is possible even at room temperature with high rates
- Clear high temperature experiments are lacking
- In terms of tritium removal for ITER:

Despite indications for deeper hydrogen penetration in Be:H in net deposition areas dense Be:T codeposits will be simply covered up by Be:D (until delamination: dust issue?) if only running a high performance shot in D₂.

Cavity experiments at PISCES

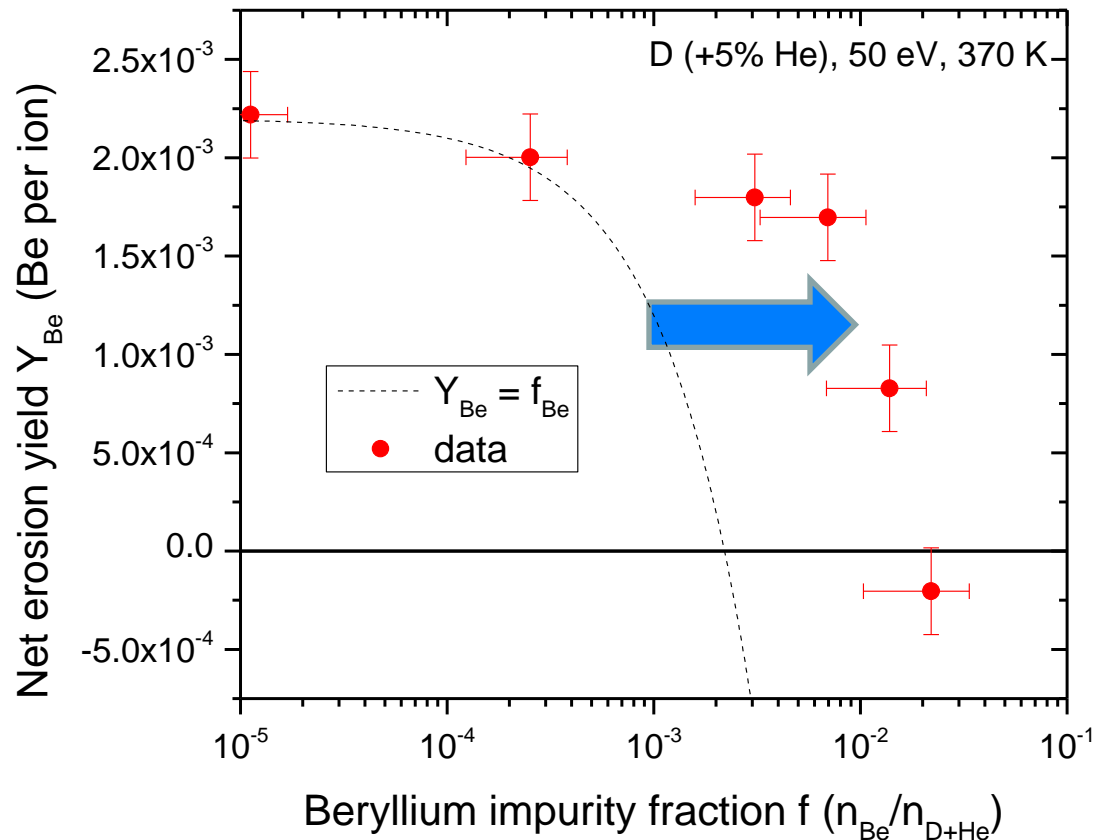
T. Schwarz-Selinger ^a, T. Dittmar ^{b,c}, R. Doerner ^b

^a *Max-Planck-Institut für Plasmaphysik, EURATOM Association, Garching, Germany*

^b *University of California San Diego, La Jolla, USA*

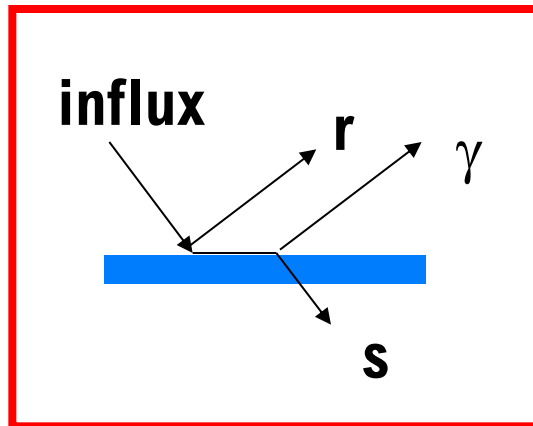
^c *Forschungszentrum Jülich, Germany*

Seeding series at 50 eV ion energy (-65 V bias), 370 K shows:



Erosion yield of 0.22% can only be compensated by seeding 2.2% Be!

The Cavity technique:



r ... reflection
γ ... transformation
(H⁰ → H₂)
s ... sticking

} $\beta = s + \gamma$

$$s + \gamma + r = 1$$

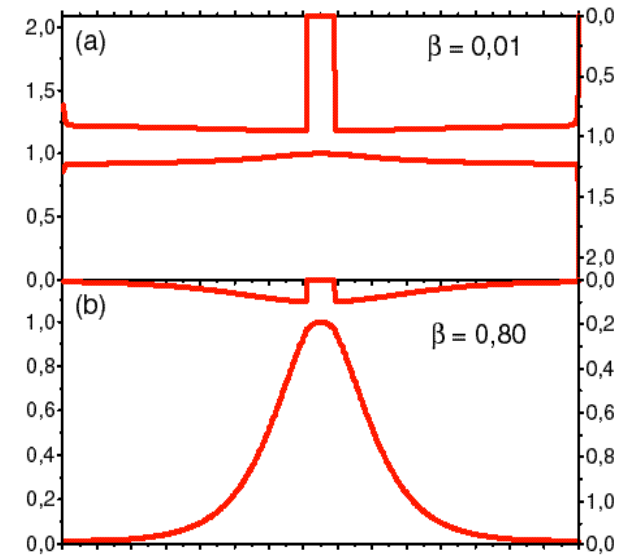
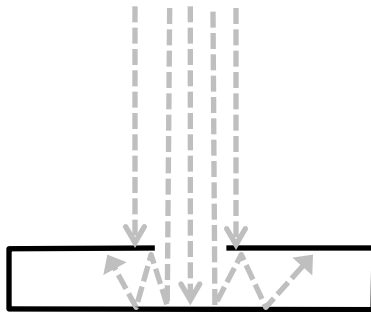
adapted from Ch. Hopf et al., JAP 87 (6), 2719 (2000)

- Here: $\gamma = 0$
- If erosion can be neglected: $\beta = s$

⇒ Analysing deposition pattern of cavity / Hohlraum / pillbox structures

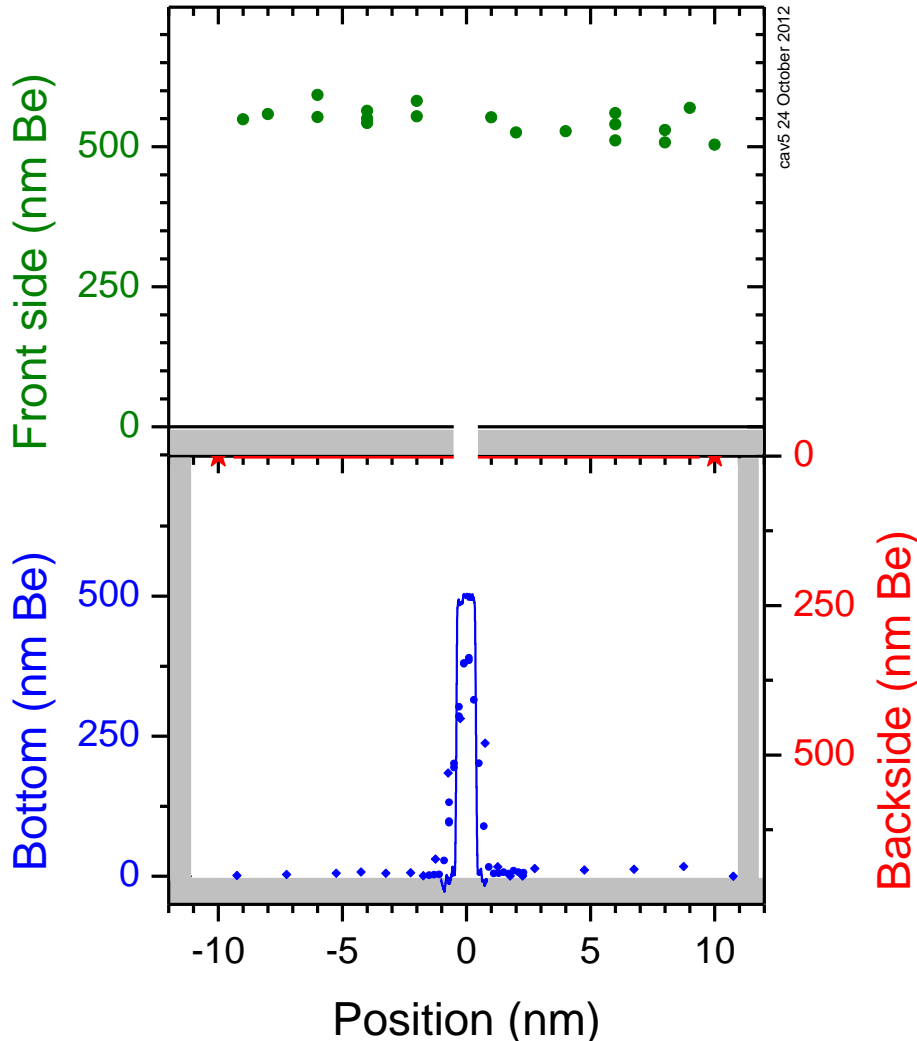
⇒ Analysing deposition pattern of cavity / Hohlraum / pillbox structures:

➤ Profiles inside the cavity



Ch. Hopf et al., JAP 87 (6), 2719 (2000)

➤ Here also: Comparing total amounts inside and outside of the cavity



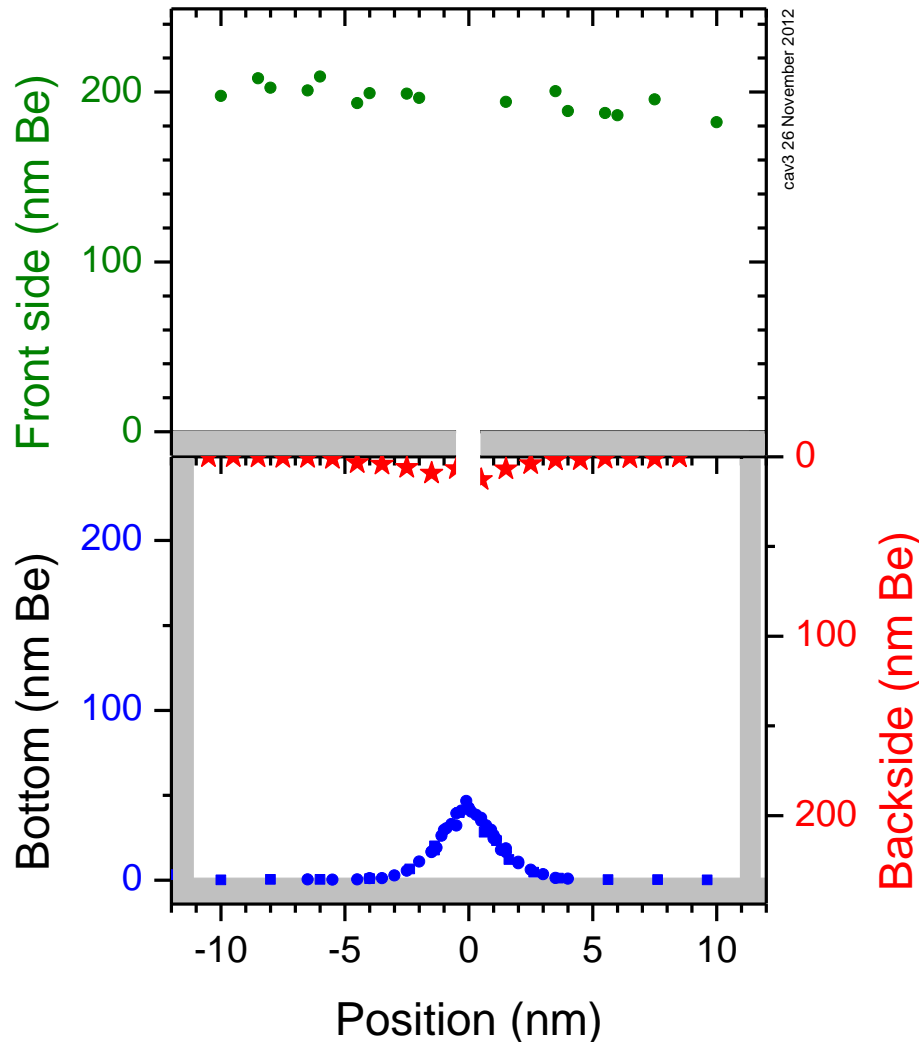
PISCES-B witness plate:

- $3.3 \cdot 10^{19} \text{ Be m}^{-2}\text{s}^{-1}$,
- 2000 sec,
- 300 K

Result:

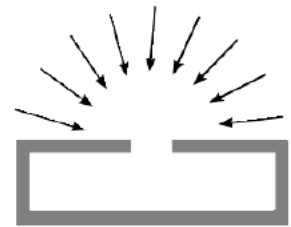
- Deposition at the bottom is a direct image of the gap (in terms of width and thickness)
- Integral Be amount inside of cavity = expected one from top
- No deposition at the back side of the lid

- ⇒ Surface loss probability $\beta = 1$
- ⇒ No erosion: **sticking $s = 1$**



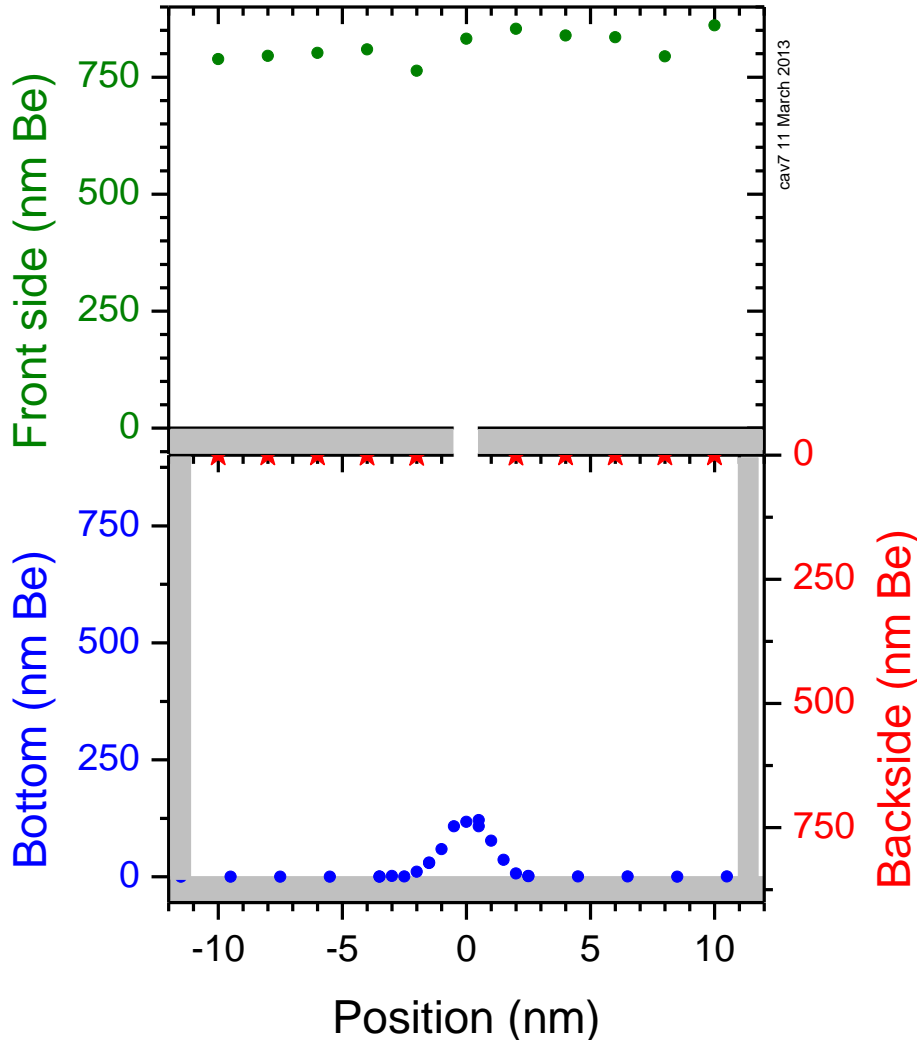
PISCES-B witness plate:

- $0.65 \cdot 10^{19} \text{ Be m}^{-2}\text{s}^{-1}$,
- 3600 sec,
- 300 K



Result:

- Deposit contains D! **D/Be = 1%!!!**
- Integral Be amount inside of cavity = expected one (top)
- $\frac{1}{4}$ of the total deposition at the back side of the lid
- ⇒ **Surface loss probability $\beta < 1?$**
- Deposition at the bottom is spread out due to collisions



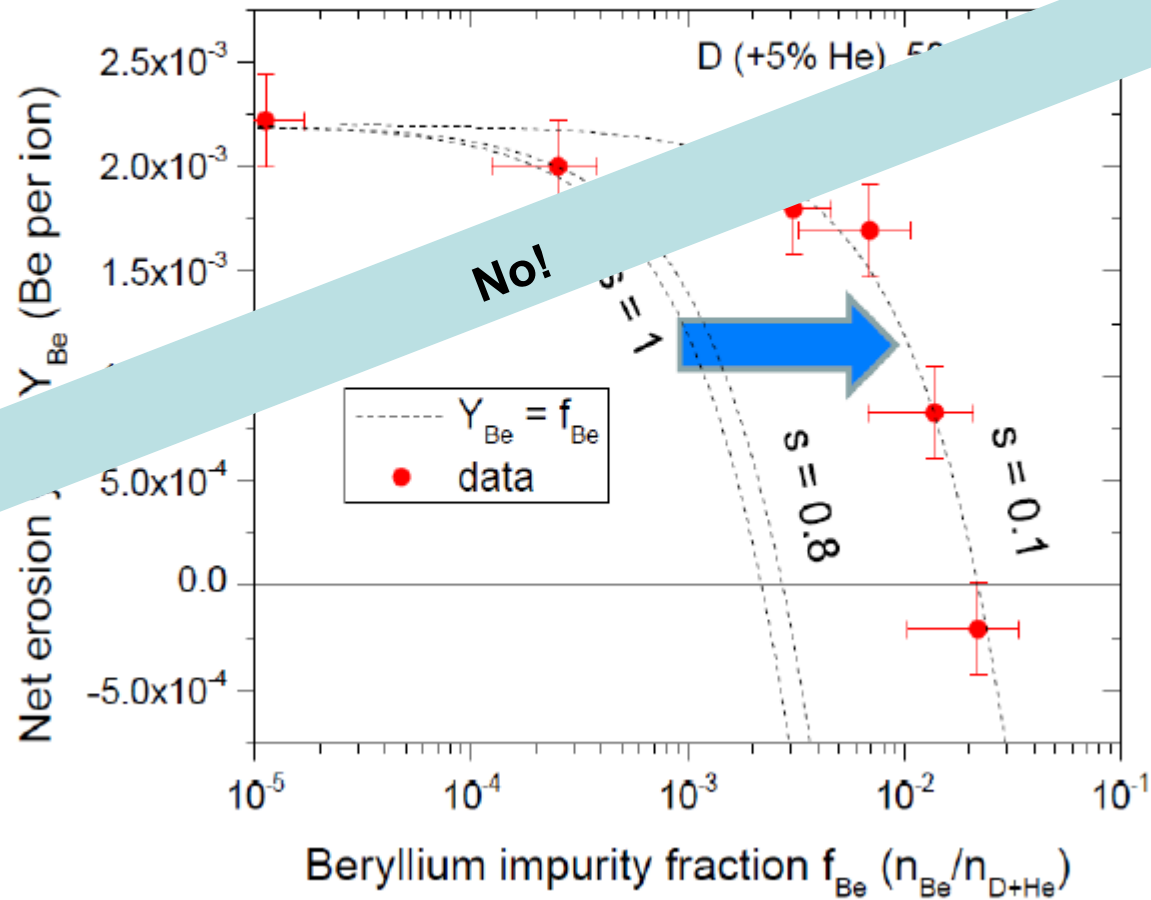
PISCES-B target plate:

- $2.6 \cdot 10^{19} \text{ Be m}^{-2}\text{s}^{-1}$,
- 3770 sec,
- $< 400 \text{ K}$

Result:

- Integral Be amount inside of cavity = $\frac{1}{2}$ of expected one (top)
- no deposition at the back side of the lid
- ⇒ Surface loss probability $\beta < 1?$
- Deposition at the bottom is spread out due to collisions

Can a surface loss probability between 0.8 and 1 explain the Be seeding experiments?



Beryllium film deposition in cavity samples in remote areas of the JET divertor during the 2011-2012 ITER-like wall campaign

S. Krat^{1,2}, M. Mayer², U. von Toussaint², P. Coad³, A. Widdowson³,
Yu. Gasparyan¹, A. Pisarev¹ and JET Contributors*

EUROfusion Consortium, JET, Culham Science Centre, Abingdon,
OX14 3DB, UK

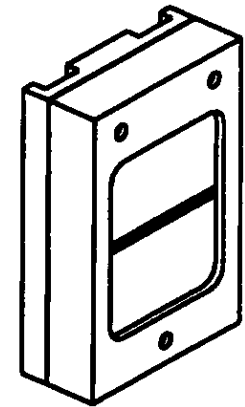
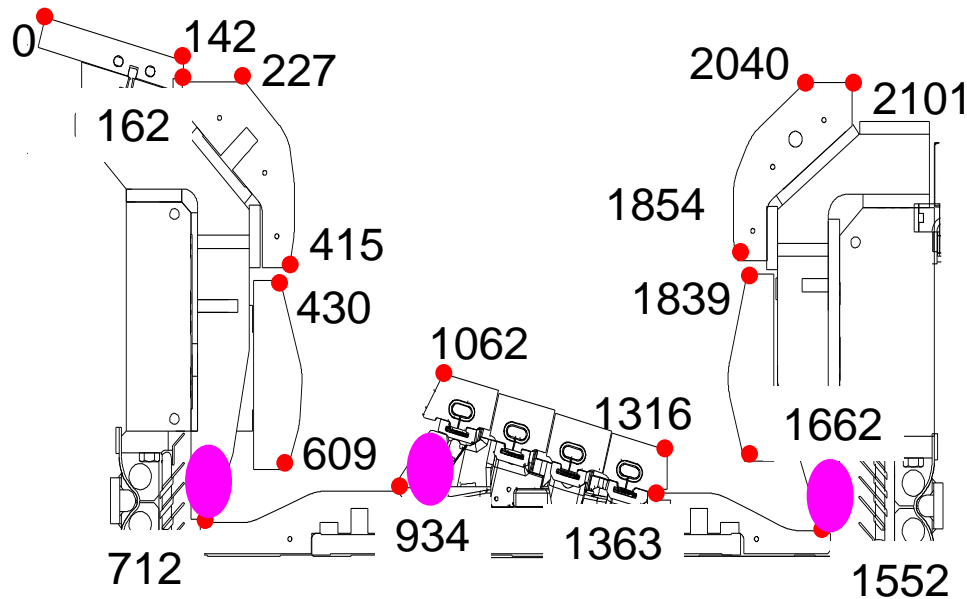
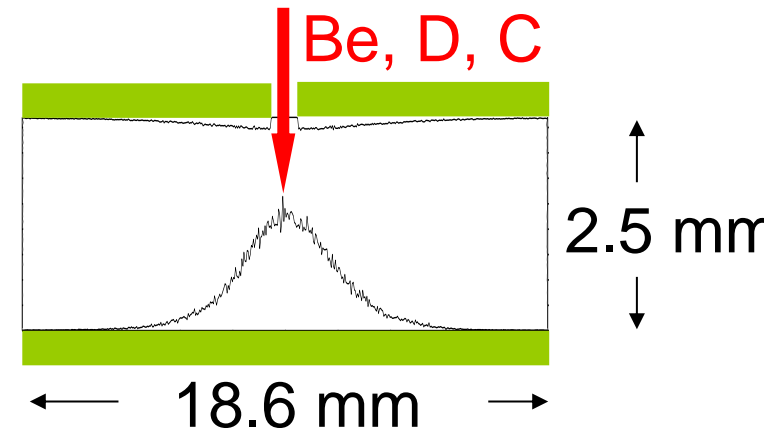
¹*National Research Nuclear University “MEPhI”, Moscow Kashirskoe sh. 31, 115409, Russia*

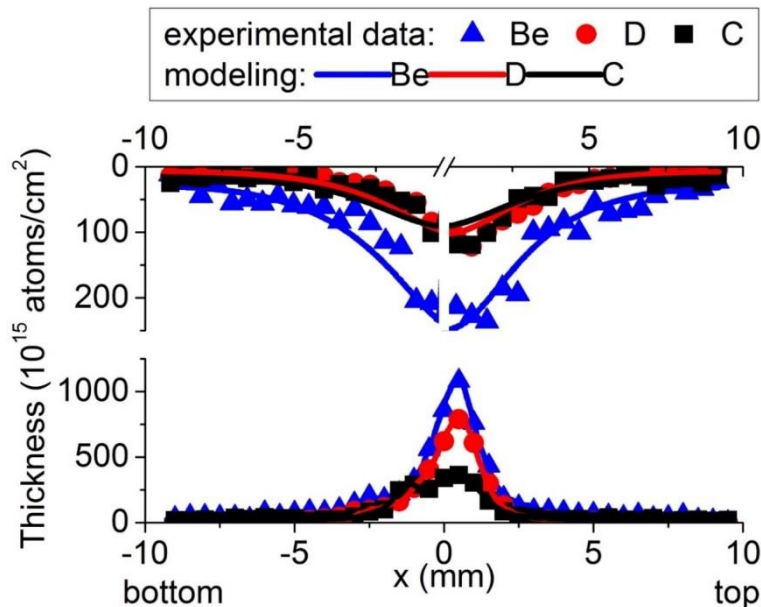
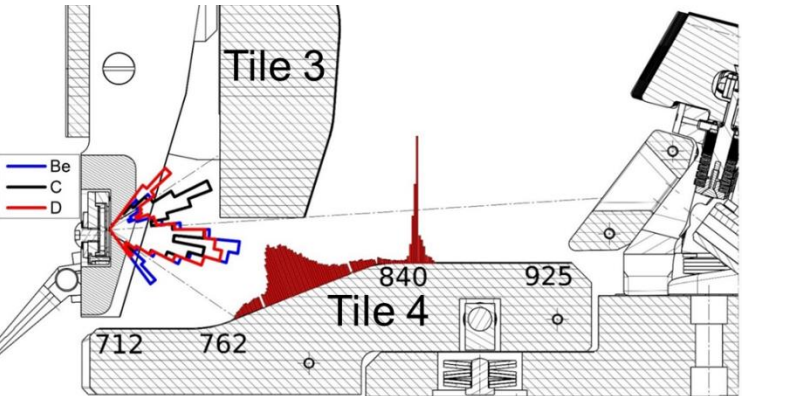
²*Max-Planck-Institut für Plasmaphysik, Boltzmannstr. 2, 85748 Garching, Germany*

³*Culham Science Centre, EURATOM/UKAEA – Fusion Association, Abingdon, Oxfordshire OX14 3DB, UK*

** See the Appendix of F. Romanelli et al., Proceedings of the 25th IAEA Fusion Energy Conference 2014, Saint Petersburg, Russia*

- Exposed in the 2011-2012 ILW campaign
- Silicon samples in stainless steel housing
- 2.4 MeV $^3\text{He}^+$ NRA at inner top and bottom for Be, D, and C





- The Be source is almost similar to the C source in JET-C for the inner divertor: Most particles originate from the sloped central area of divertor tile 4, from the direction about 15° below the horizontal plane, near the strike points on this tile
- A large amount of C originates from **the rear side of tile 3**. Parasitic discharges or chemical erosion of C by deuterium atoms might be the cause
- The D source is a mixture of Be and C sources
- Relatively large amount of C, with $C/Be=0.5$
- Large D content with $D/(Be+C)=0.3$
- For D and Be, highly sticking species are present ($S>0.7$). Only low sticking species for C ($S<0.5$)

- Relatively large amount of C, with $C/Be=0.5$
- Large D content with $D/(Be+C)=0.3$
- How to explain low sticking species for Be?
Erosion?

| | Two-component model | One-component model | Ratio top/bottom |
|----|-------------------------------------|---------------------|------------------|
| Be | $91\% \cdot 0.33 + 9\% \cdot 0.97$ | 0.4 | 0.46 |
| D | $48\% \cdot 0.49 + 52\% \cdot 0.76$ | 0.64 | 0.68 |
| C | $17\% \cdot 0.01 + 82\% \cdot 0.53$ | 0.44 | 0.53 |

- At least cavities act as pinhole camera [to locate sources](#)

- Film deposition in shadowed areas of the JET divertor was analyzed using NRA.
- The Be deposition rate is **34 times smaller** than the C deposition rate during the 2005-2009 JET-C campaign. D accumulation rate was **45 times smaller**
- Be and C originate mostly from nearby strike point locations, they are likely reflected or re-eroded from W divertor tiles; some C in the inner divertor originates from the rear side of tile 3, where it might originate due to parasitic discharges or erosion by atomic deuterium
- There are indications that re-erosion plays a significant role in particle transport in the outer divertor and under tile 5.
- Low effective sticking coefficients dominate for all components, with $s \leq 0.3$. This cannot be explained by atomic reflection or re-erosion.

S. Krat^{1,2}, M. Mayer¹, C. Porosnicu³

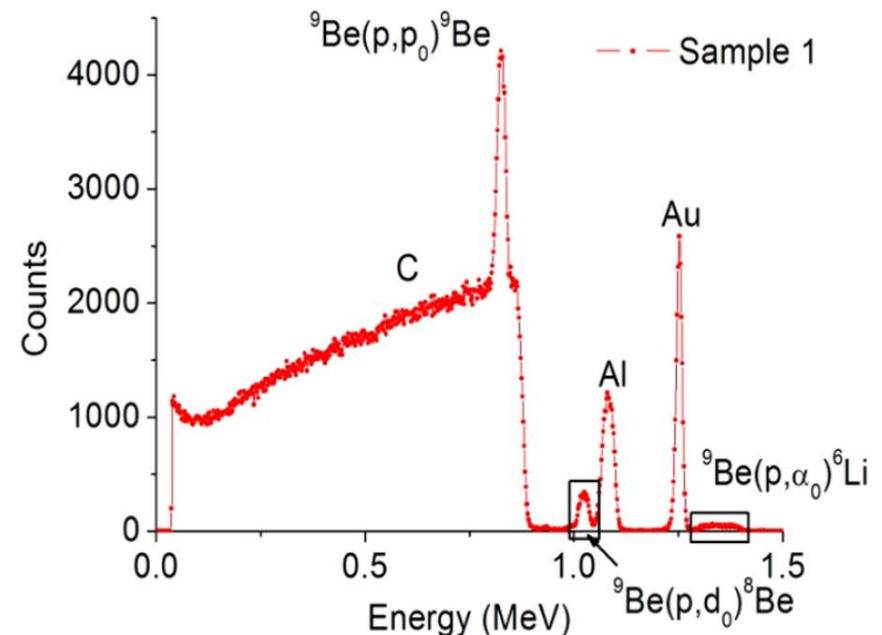
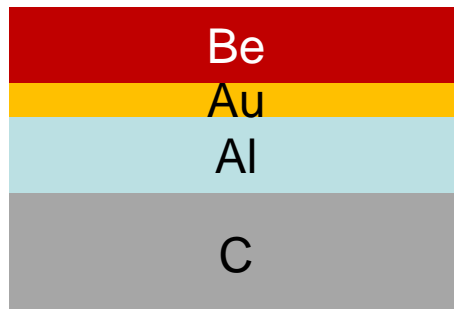
¹Max-Planck-Institut für Plasmaphysik, Garching, Germany

²National Research Nuclear University MEPhI, Moscow, Russia

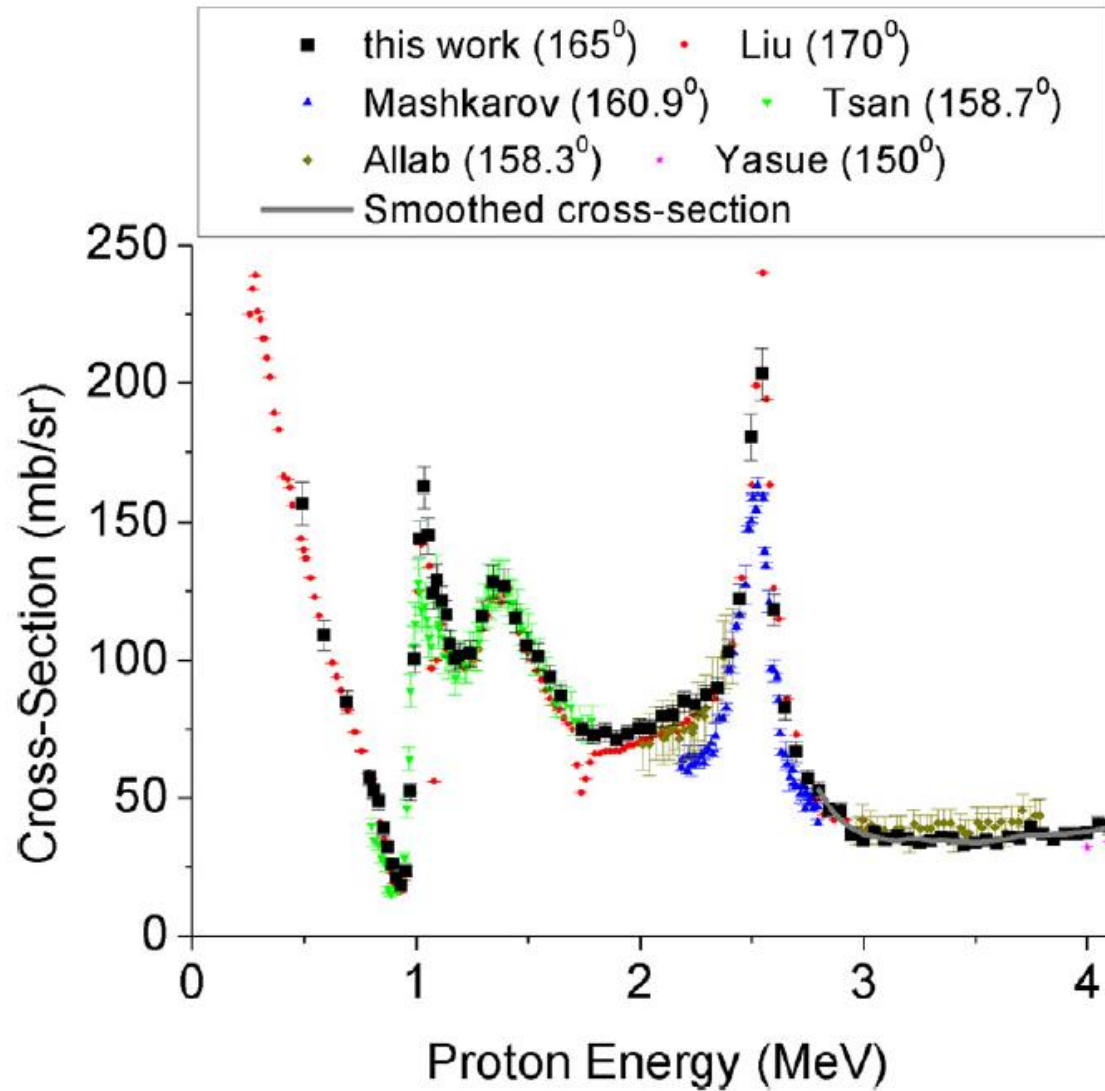
³NILPRP, Association EURATOM-MEdC, Bucharest, Romania



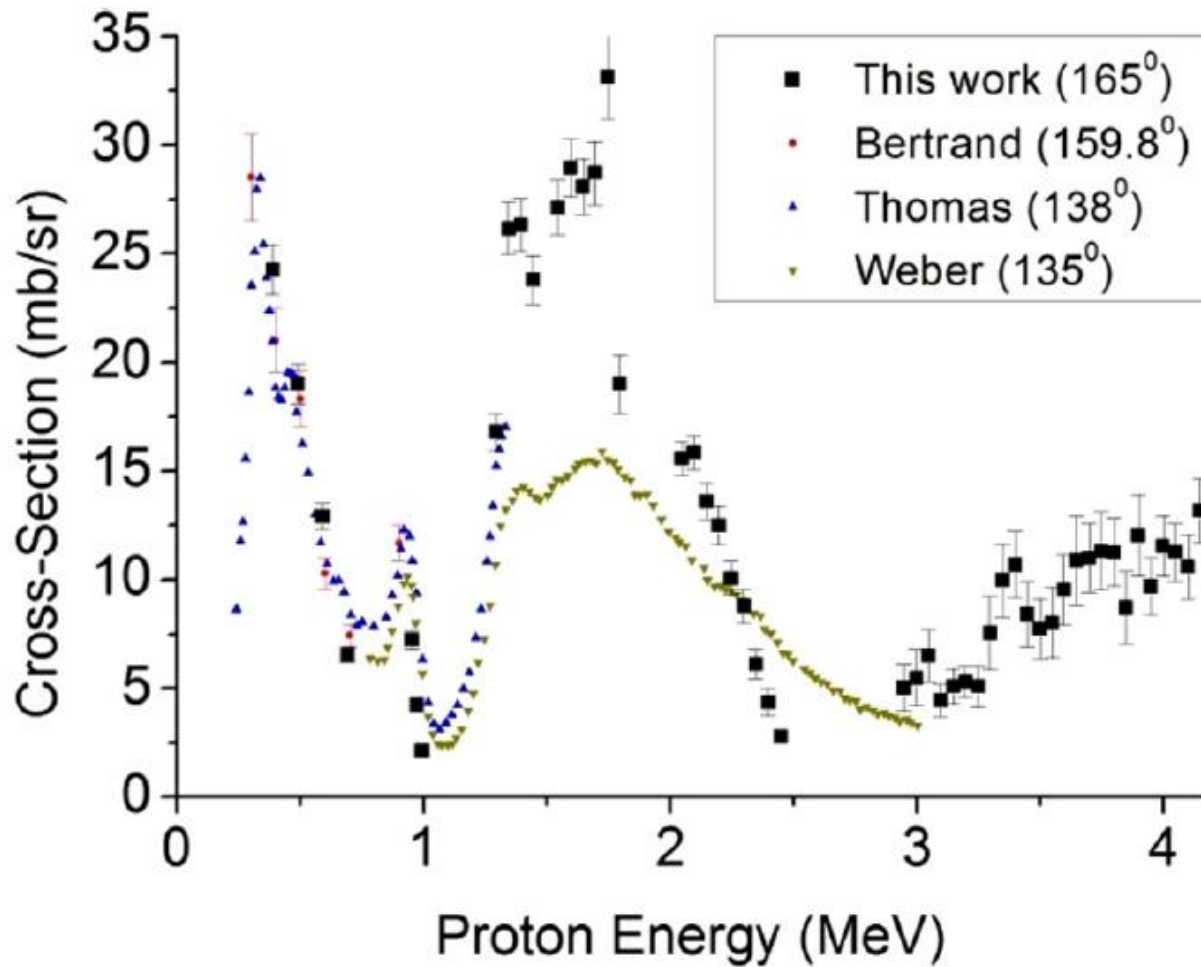
- ${}^9\text{Be}(p,p_0){}^9\text{Be}$ EBS between 400keV and 4150keV (non RBS > 230keV)
- ${}^9\text{Be}(p,d_0){}^8\text{Be}$ NRA between 400keV and 4150keV
- ${}^9\text{Be}(p,\alpha_0){}^6\text{Li}$ EBS between 400keV and 1300keV
- Scattering angle of 165°
- Measured on Be thin films and benchmarked on bulk films



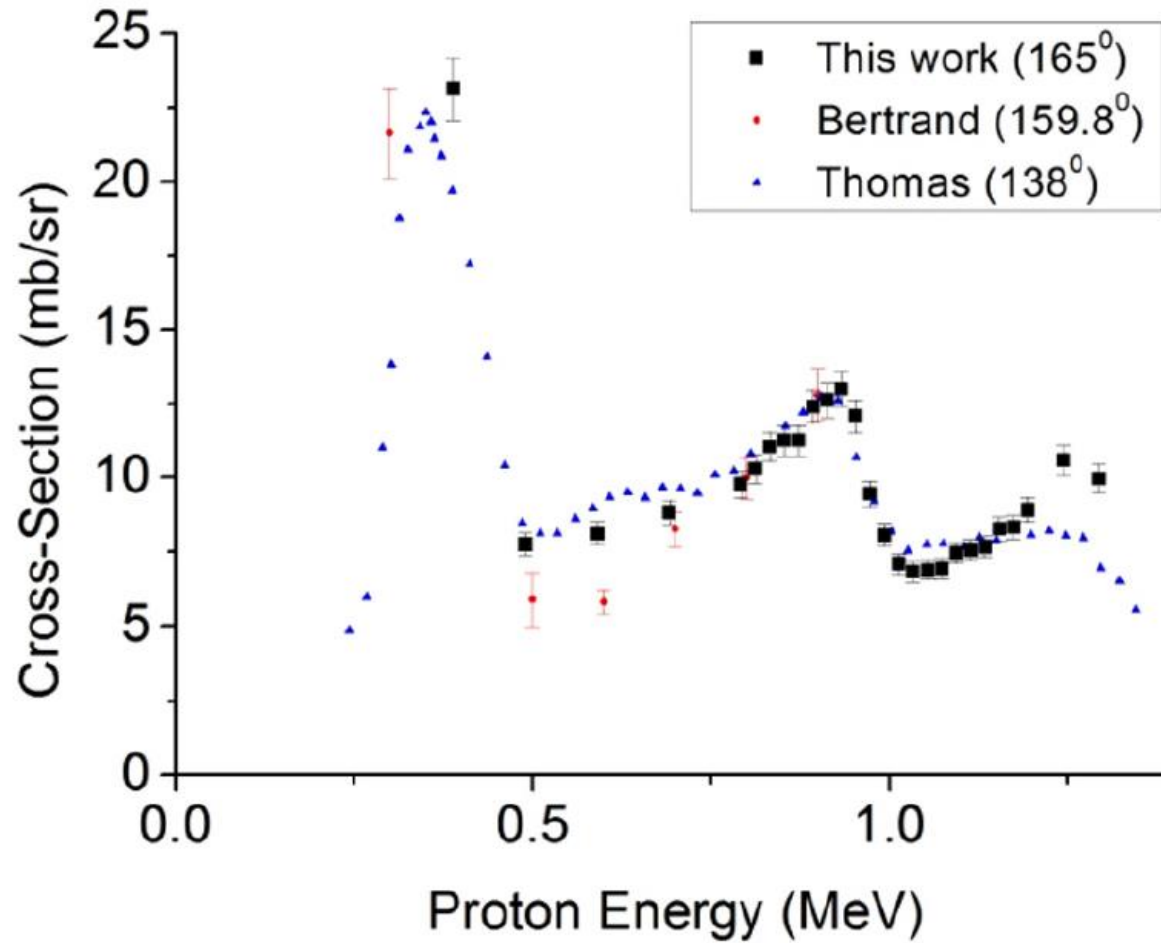
${}^9\text{Be}(p,p_0){}^9\text{Be}$ elastic scattering



${}^9\text{Be}(p,d_0){}^8\text{Be}$ elastic scattering



${}^9\text{Be}(p, \alpha_0){}^6\text{Li}$ elastic scattering





JET inner wall inserts

S. Krat

JET



This work has been carried out within the framework of the EUROfusion Consortium and has received funding from the Euratom research and training programme 2014-2018 under grant agreement No 633053. The views and opinions expressed herein do not necessarily reflect those of the European Commission.

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¹Max-Planck-Institut für Plasmaphysik, Garching, Germany

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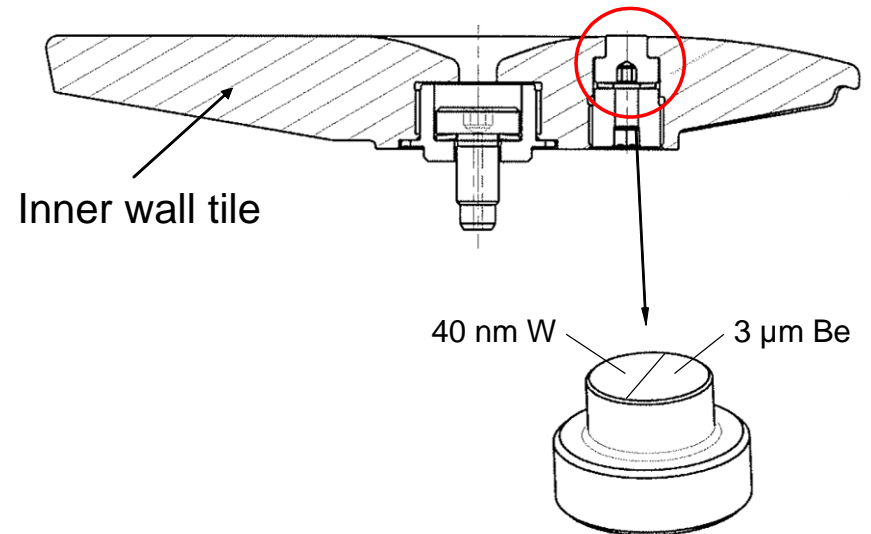
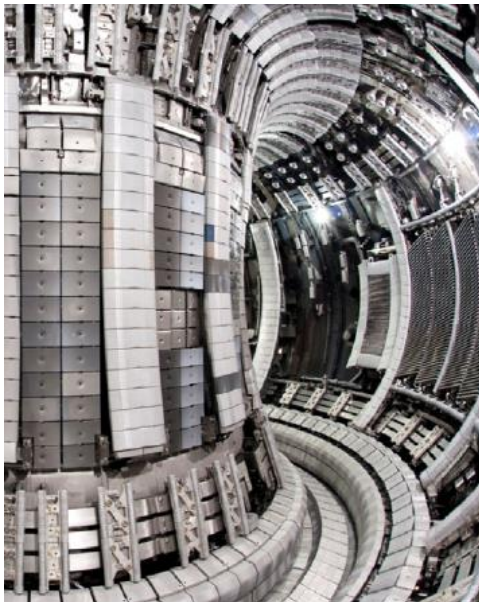
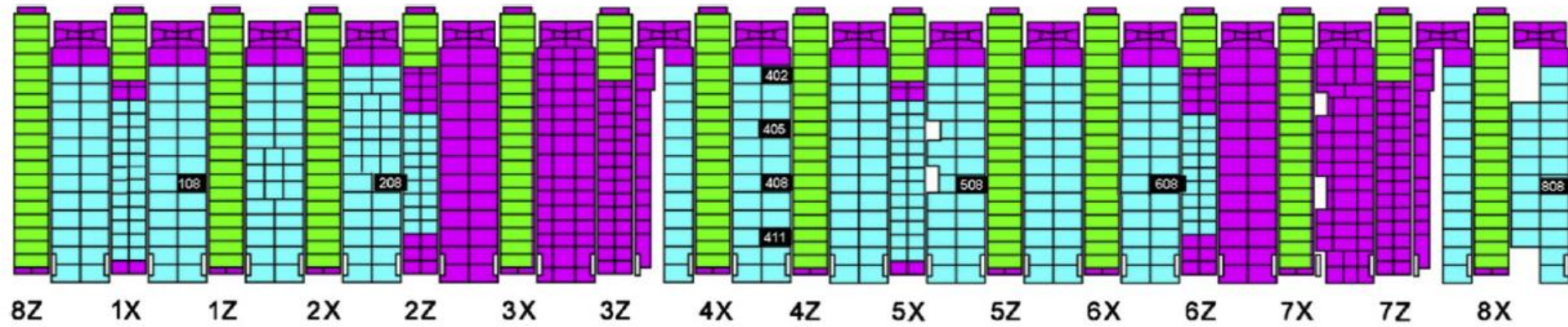
³NILPRP, Association EURATOM-MEdC, Bucharest, Romania

⁴Culham Centre for Fusion Energy, Culham Science Centre, Abingdon, UK

** See the Appendix of F. Romanelli et al., Proceedings of the 25th IAEA Fusion Energy Conference 2014, Saint Petersburg, Russia*



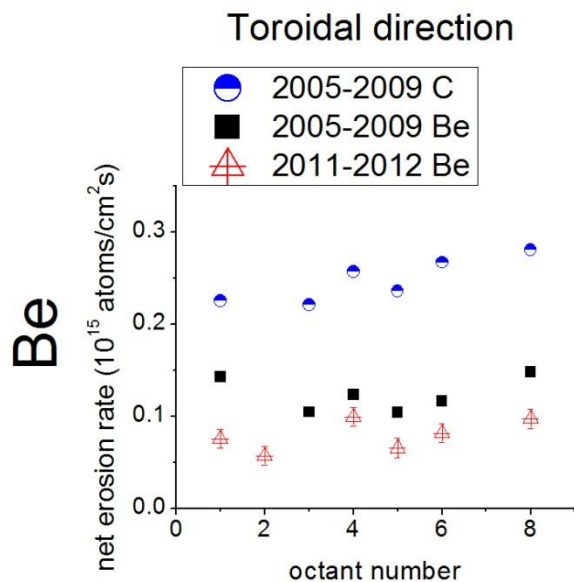
Long-Term Samples: Erosion at inner wall cladding



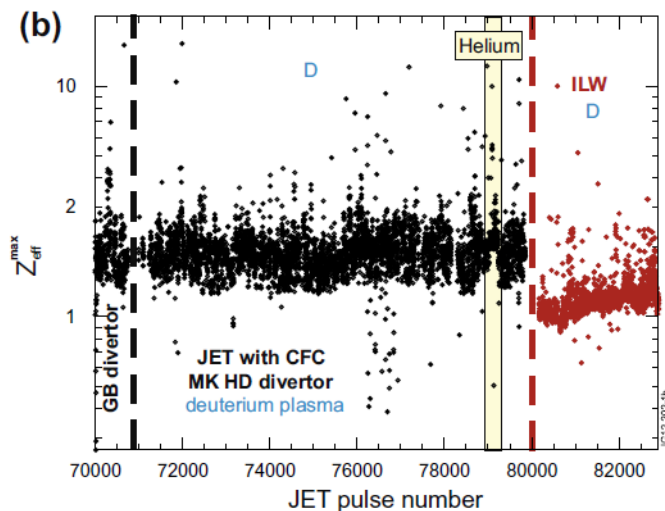
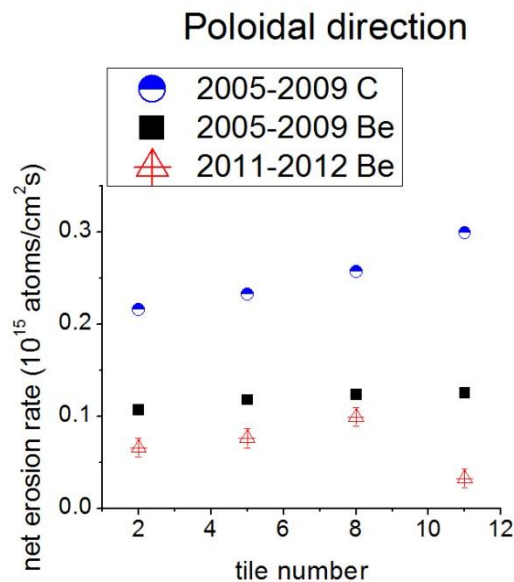
Erosion at inner wall 2011-2012

- Main chamber erosion: Erosion of Be decreased by factor ~ 5 compared to C
 - Decrease of plasma impurity concentration from $\sim 1 - 2\%$ C to $0.1 - 0.2\%$ Be
- \Rightarrow Decrease of net impurity source from inner wall
- \Rightarrow Reason: Absence of chemical erosion by low-energy neutral hydrogen

M. Mayer et al., JNM 438 (2013) S780



S. Krat et al., JNM 456 (2015) 106



S. Brezinsek et al., JNM 438 (2013) S303

ILW discharge campaigns

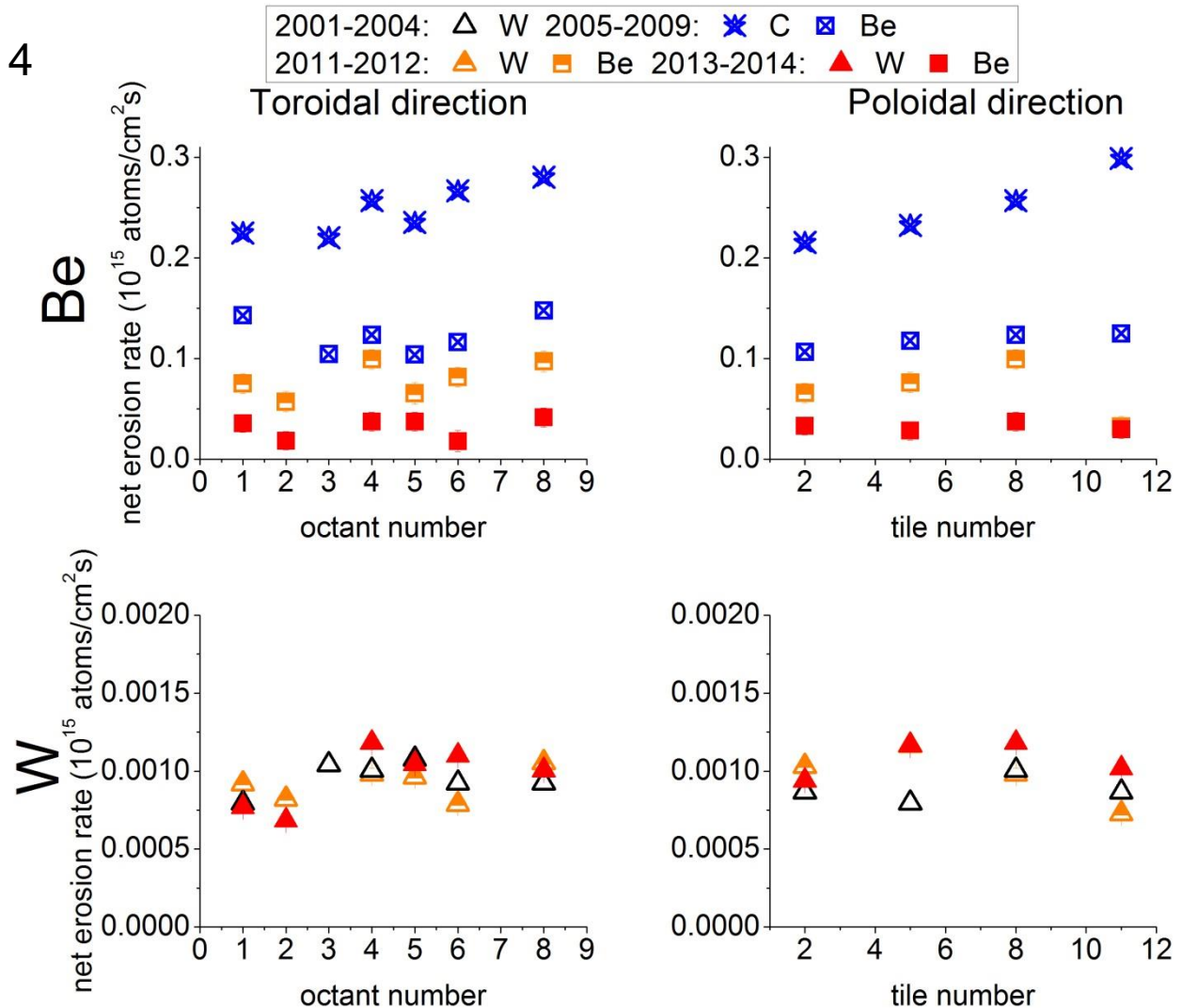


| Discharge campaign | Number of discharges | Number of successful discharges ($I_p > 0.7$ MA) | Total discharge time ($I_p > 0.7$ MA), 10^4 s | Divertor phase discharge time, 10^4 s | Limiter phase discharge time, 10^4 s |
|--------------------|----------------------|---|--|---|--|
| 2011-2012 | 3812 | 2819 | 6.41 | 4.51 | 1.9 |
| 2013-2014 | 4150 | ??? | 7.12 | 5.09 | 2.03 |

Erosion at inner wall 2011-2014



- Erosion of Be in 2013-2014 decreased by factor 1.8 compared to 2011-2012
- Erosion of W remained constant



- Inner wall erosion monitors 2016 – 201? are foreseen
- Inserts to be provided by CCFE
- W coating at IPP (M. Mayer)
- Be coating at NILPRP (C. Lungu)
- IBA pre-analysis at IPP (M. Mayer)

Status: Waiting for inserts



Erosion and deposition in the JET divertor during the first ILW campaign

M. Mayer

JET



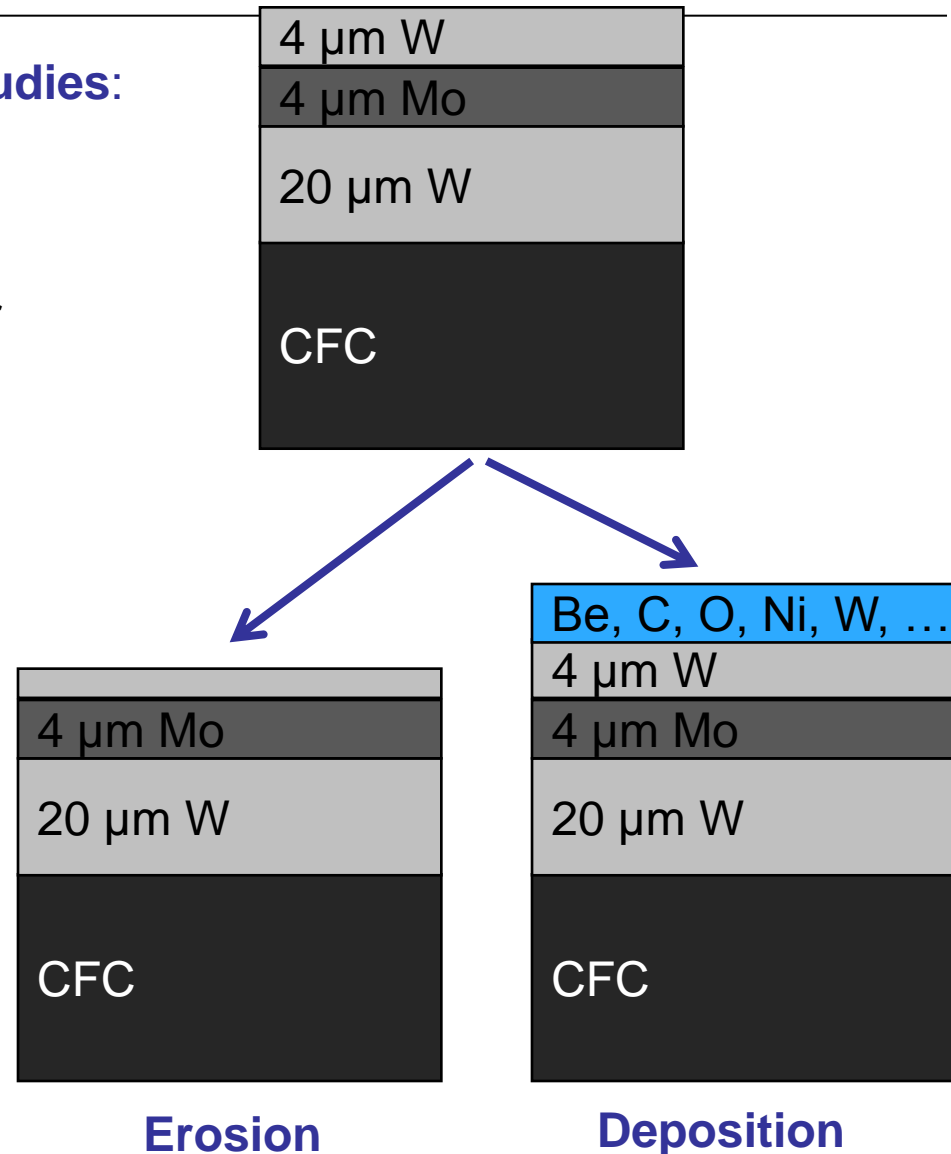
This work has been carried out within the framework of the EUROfusion Consortium and has received funding from the Euratom research and training programme 2014-2018 under grant agreement No 633053. The views and opinions expressed herein do not necessarily reflect those of the European Commission.

Experimental: Marker layers in JET-ILW

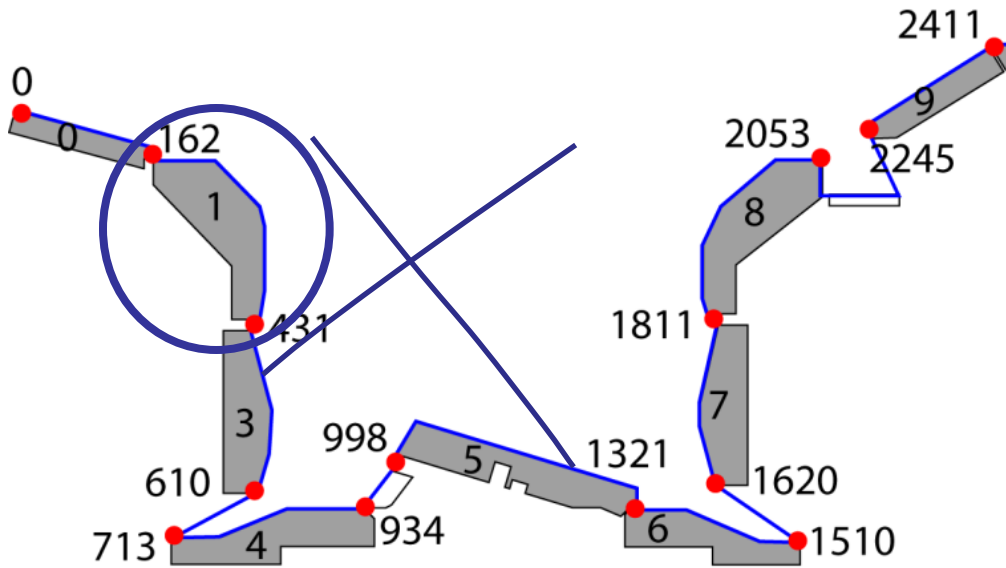
Marker layers for erosion/deposition studies:

4 μm W

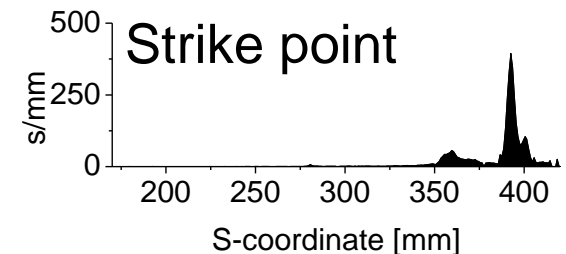
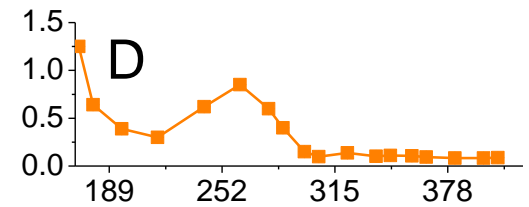
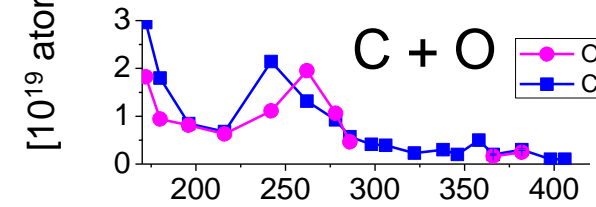
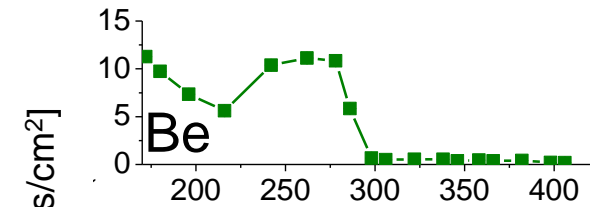
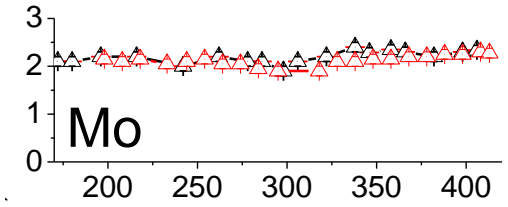
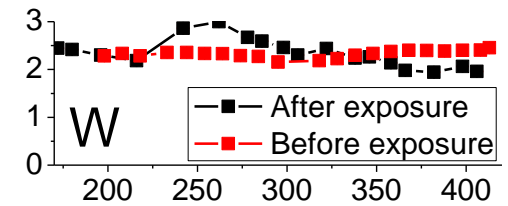
- 4 μm Mo interlayer for separation of thick 20 μm W coating from marker layer
- Non-destructive analysis by RBS + NRA before and after exposure
- Exposed 2011 – 2012:
13 h divertor plasma phase + 6 h limiter



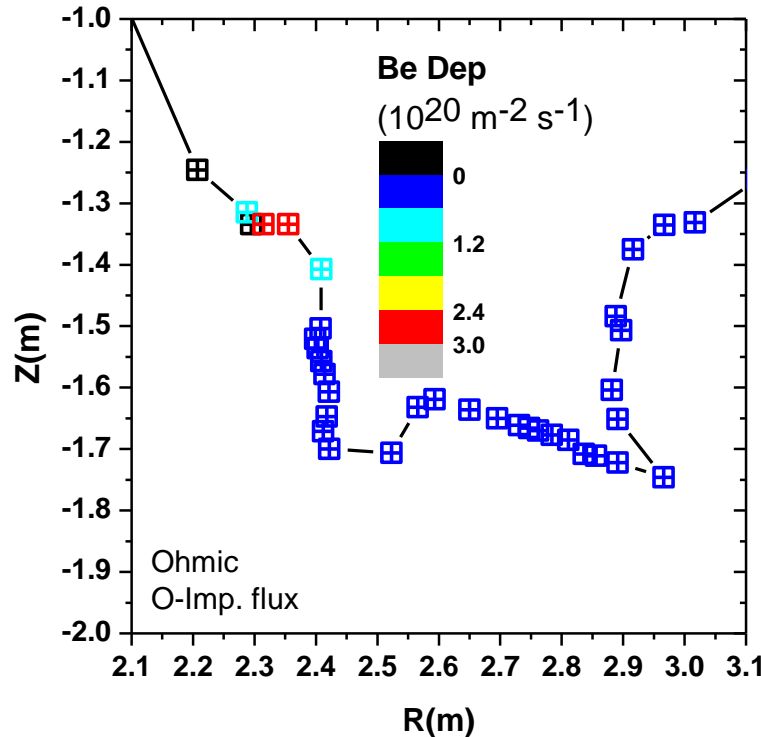
Tile 1: JET-ILW 2011-2012



- Large amounts of deposited Be on horizontal and sloping part (s = 160 – 280 mm)
- Deposition of W on sloping part
- 10 – 20% of C and O in thick deposits
- Erosion of W at bottom vertical part of the tile
- Homogeneous depth profile of D in deposits



❖ Erosion & Deposition in JET-ILW [1]



❖ Absolute amounts:

➤ WalIDYN Be dep. Rate on „Apron“:
(Equilib. value)
 $0.5 \text{ to } 2.8 (10^{20} \text{ Be m}^{-2} \text{ s}^{-1})$

➤ Experimental (post mortem) [2]
Average over many different
plasma scenarios and other
uncertainties
 $0.2 \text{ to } 0.3 (10^{20} \text{ Be m}^{-2} \text{ s}^{-1})$

Qualitative match to post mortem results [2,3]

- Main chamber is net erosion zone
- Be deposition mainly on tile 1 („Apron“)
- Rest of divertor no net Be deposition

[1] K. Schmid et al. *J. Nucl. Mat.* 463 (2015) p. 66

[2] J. P. Coad et al. *Phys. Scr.* T159 (2014) 014012

[3] M. Mayer et al. *PFMC 2015, Physica Scripta*, in print

→ WalIDYN too high by trend, but close given the large uncertainties

- PSI studies involving Be sample preparation or Be sample implantation has come to an end at IPP Garching
- Activities are still possible for experiments where no significant Be amounts are mobilized.
- Analysis of JET marker tiles and wall inserts will be continued
- Collaboration with colleagues from PISCES is being continued (mostly ion beam analysis, but also XPS depth profiling)
- Walldyn modeling on Be transport for Jet and ITER will be continued
- Studies on isotope exchange and Be cavities need to be published
- Collaboration with colleagues from MEdC is being largely reduced (mixed-material co-deposited D, O or N -containing layers by TVA)

D retention in and release from Be mixed materials

K. Sugiyama, C. Porosnicu, W. Jacob, I. Jepu, C.P. Lungu, Nuclear Materials and Energy 6 (2016) 1

Erosion at the inner wall of Jet

S. Krat, Yu Gasparyan, A. Pisarev, I. Bykov, M. Mayer, G. de Saint Aubin, M. Balden, C.P. Lungu, A. Widdowson, JET-EFDA contributors, J. Nucl. Metr. 456 (2015) 106

Erosion and Deposition in JET

M Mayer, S Krat, W Van Renterghem, A Baron-Wiechec, S Brezinsek, I Bykov, P Coad, Yu Gasparyan, K Heinola, J Likonen, A Pisarev, C Ruset, G de Saint-Aubin, A Widdowson and JET Contributors, Physica Scripta, T167 (2016)

Cross sections for IBA analysis

S Krat, M. Mayer, C. Porosnicu, Nuclear Instrum. Meth B 358 (2015) 72

Modelling beryllium migration in JET

K. Schmid, K. Krieger, S.W. Lisgo, G. Meisl, S. Brezinsek, J. Nucl. Mat. 463 (2015) 66