



ASIPP

Preliminary results of commissioning of EAST full W upper divertor and deuterium retention & permeation of W irradiated by heavy ions

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Contents



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1. Status of W/Cu divertor on EAST
2. Retention/permeation of irradiated W
3. Summary & outlook



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W/Cu divertor on EAST

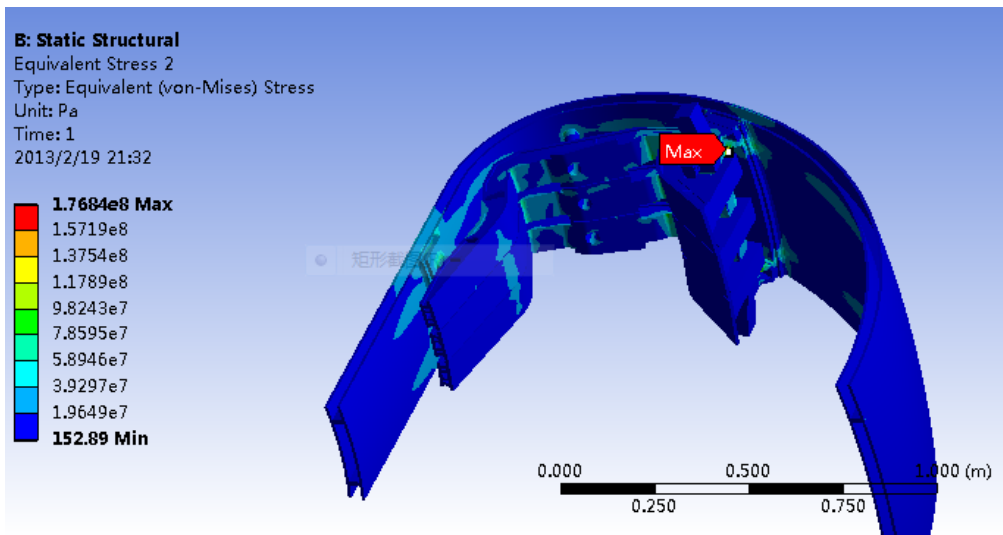
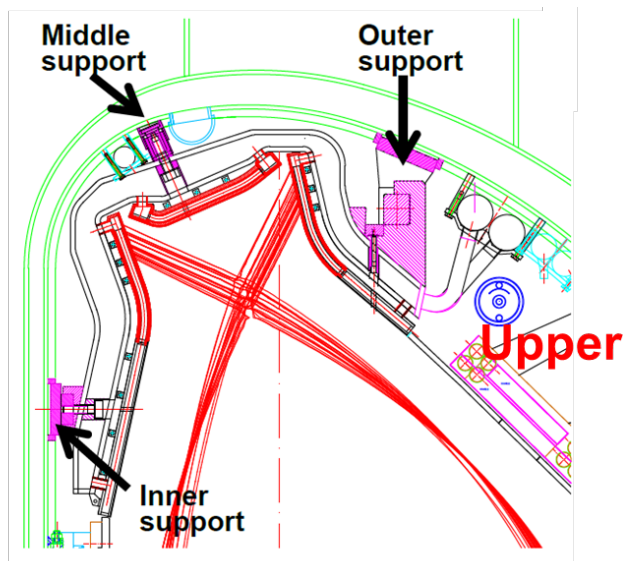
- Engineering



Structure Design and Evaluation



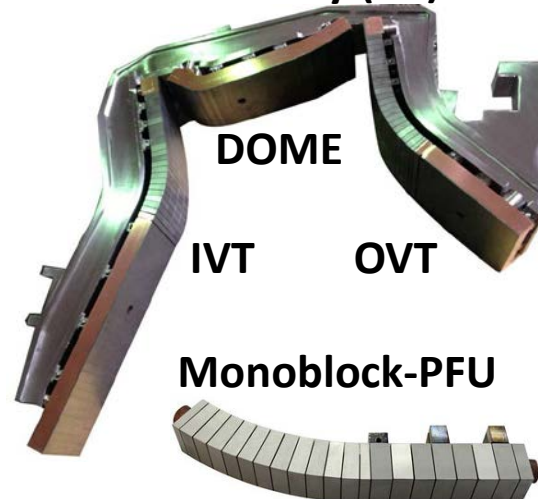
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Module = CB + IVT + DOME + OVT

EAST upper tungsten divertor has modular structure with 80 modules in total. Each module supported by inner/outer rail and middle support.

Cassete Body (CB)

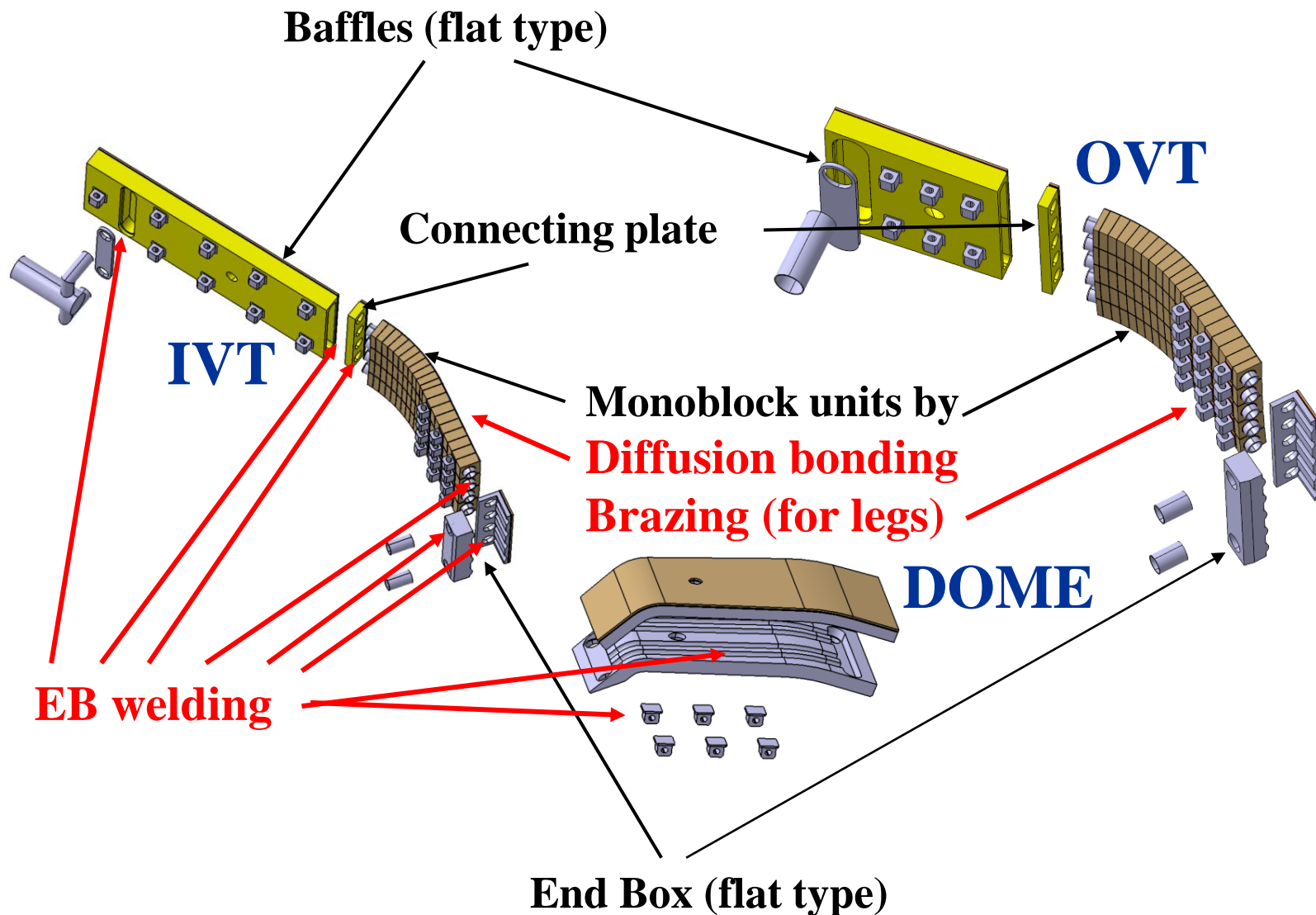




PFCs Configuration and Connections



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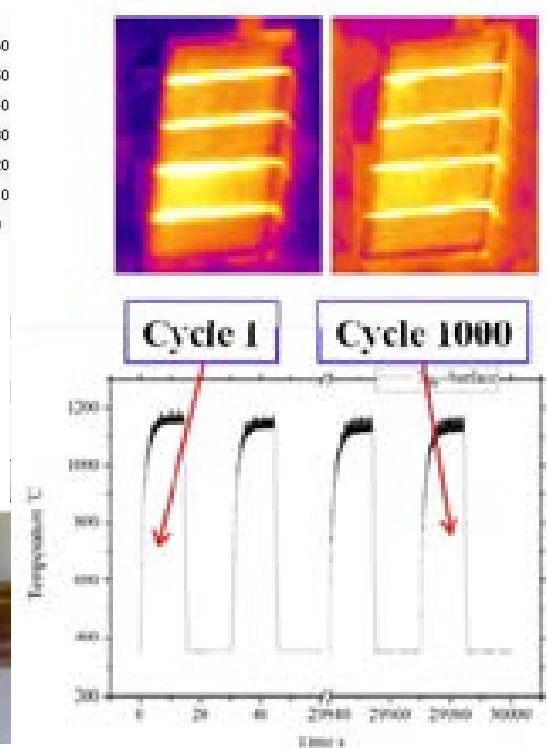
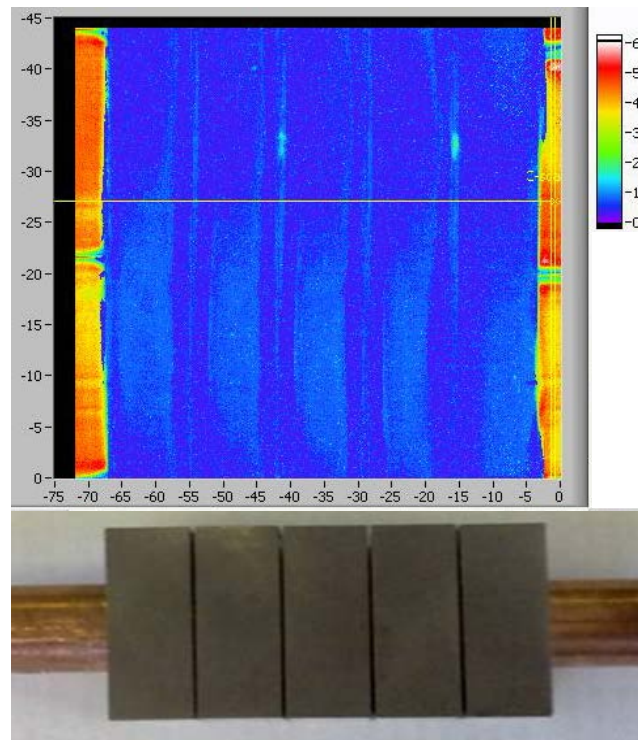
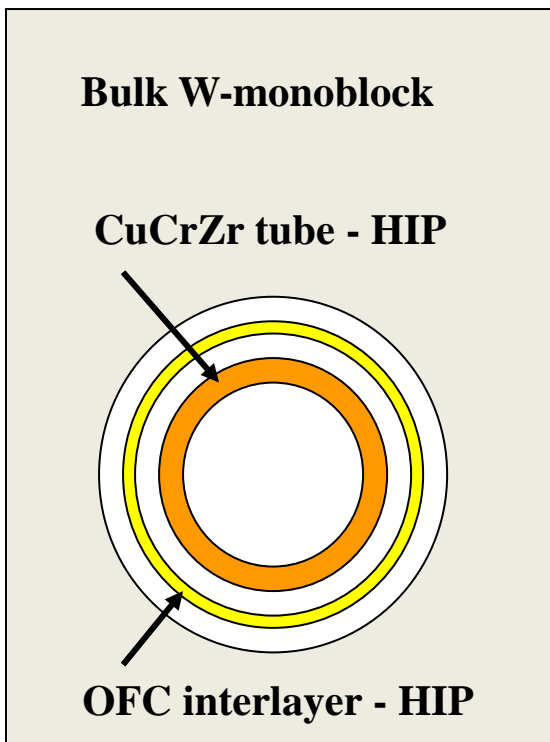


ITER-like Monoblock W/Cu PFUs



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- **HIP+HIP:** W/Cu mockups were manufactured successfully by a double Hot Isostatic Pressing (HIP) technology
- **NDT results:** Passed NDT check for dual bondings between W/OFC/CuCrZr tube; **HHF testing:** 10MW/m²-1000cycles



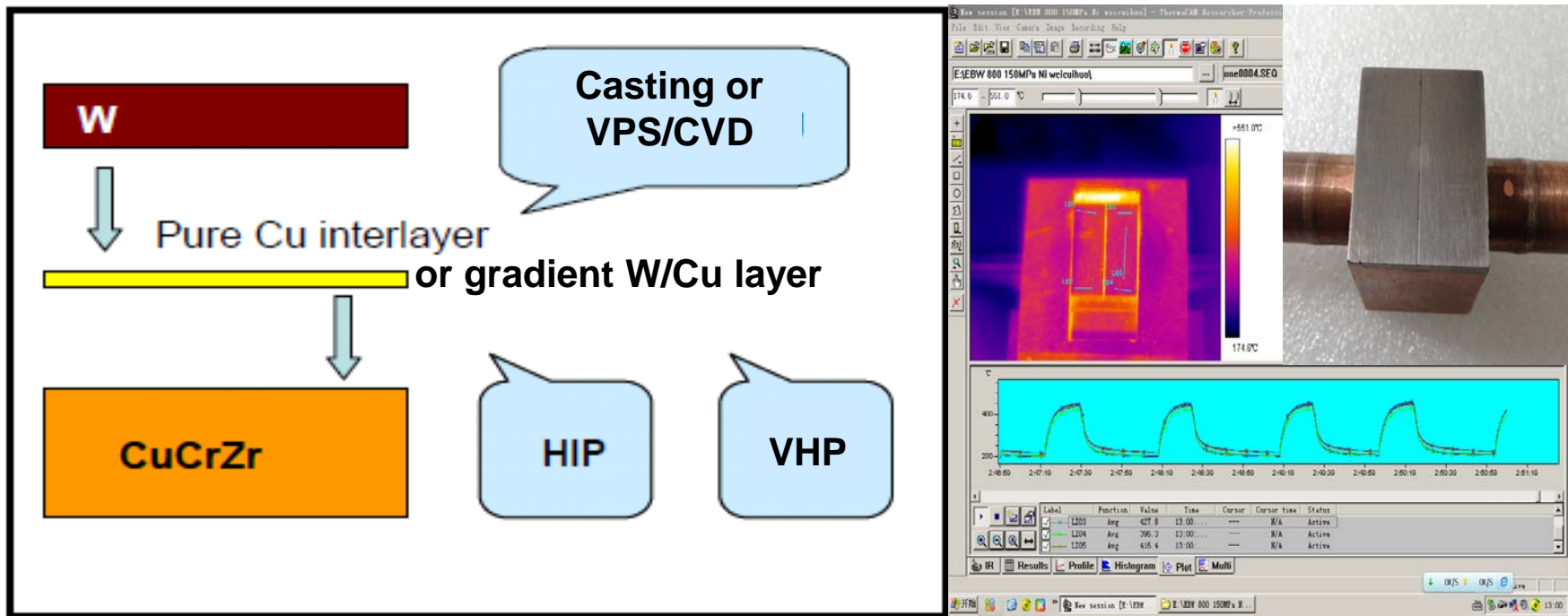


Flat-type W/Cu PFUs



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- **Casting + HIP:** The interface of W/Cu were joined by casting, and then the interface of Cu/CuCrZr was bonded by HIP at lower temperature of 500~600°C.
- **NDT results:** Passed NDT check for dual bondings between W/OFC/CuCrZr plate; **HHF testing:** 5MW/m²-1000cycles

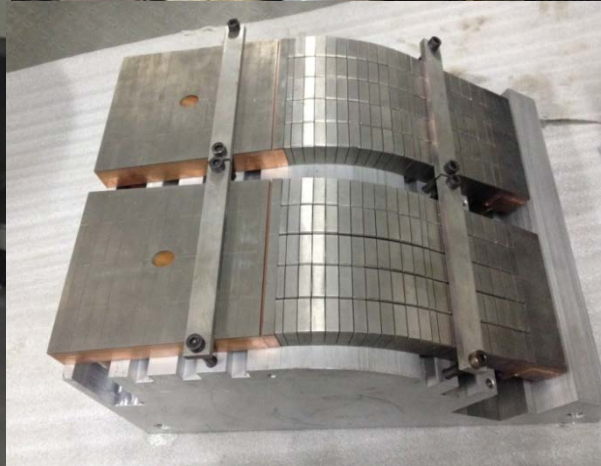
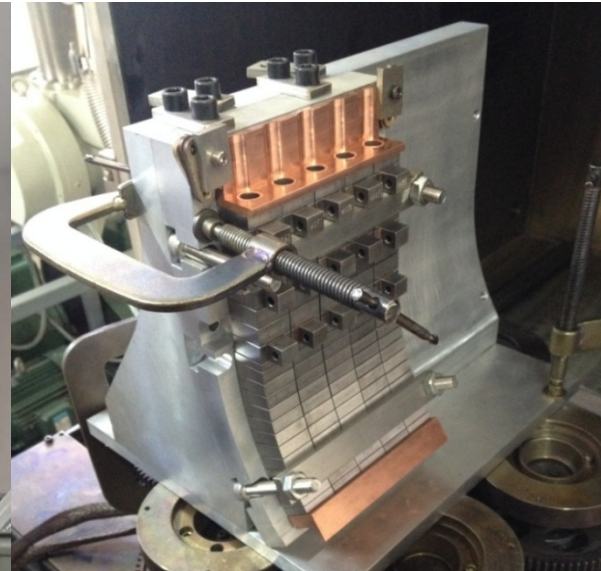




E-beam welding of PFCs



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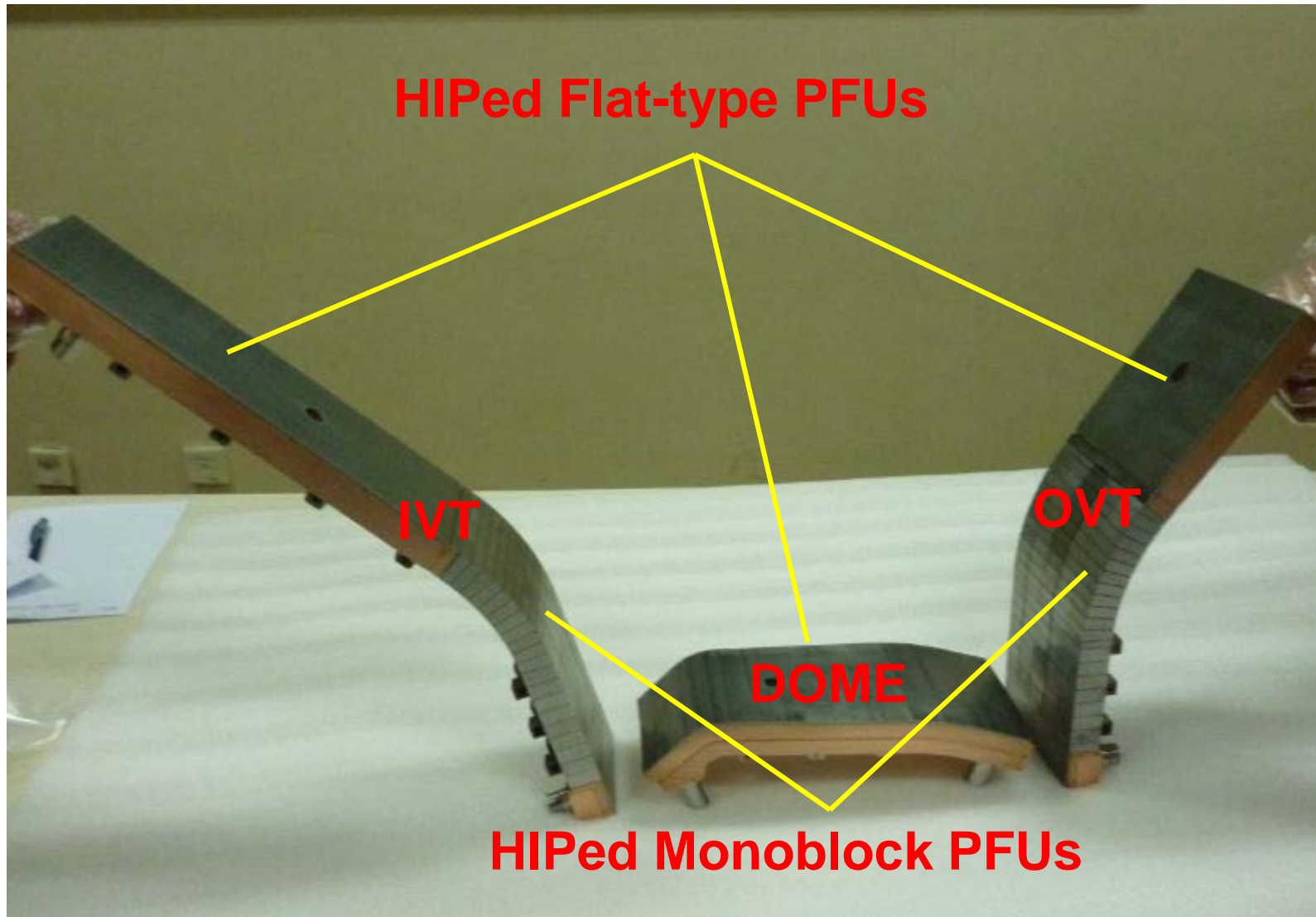




OVT/IVT/DOME-PFCs

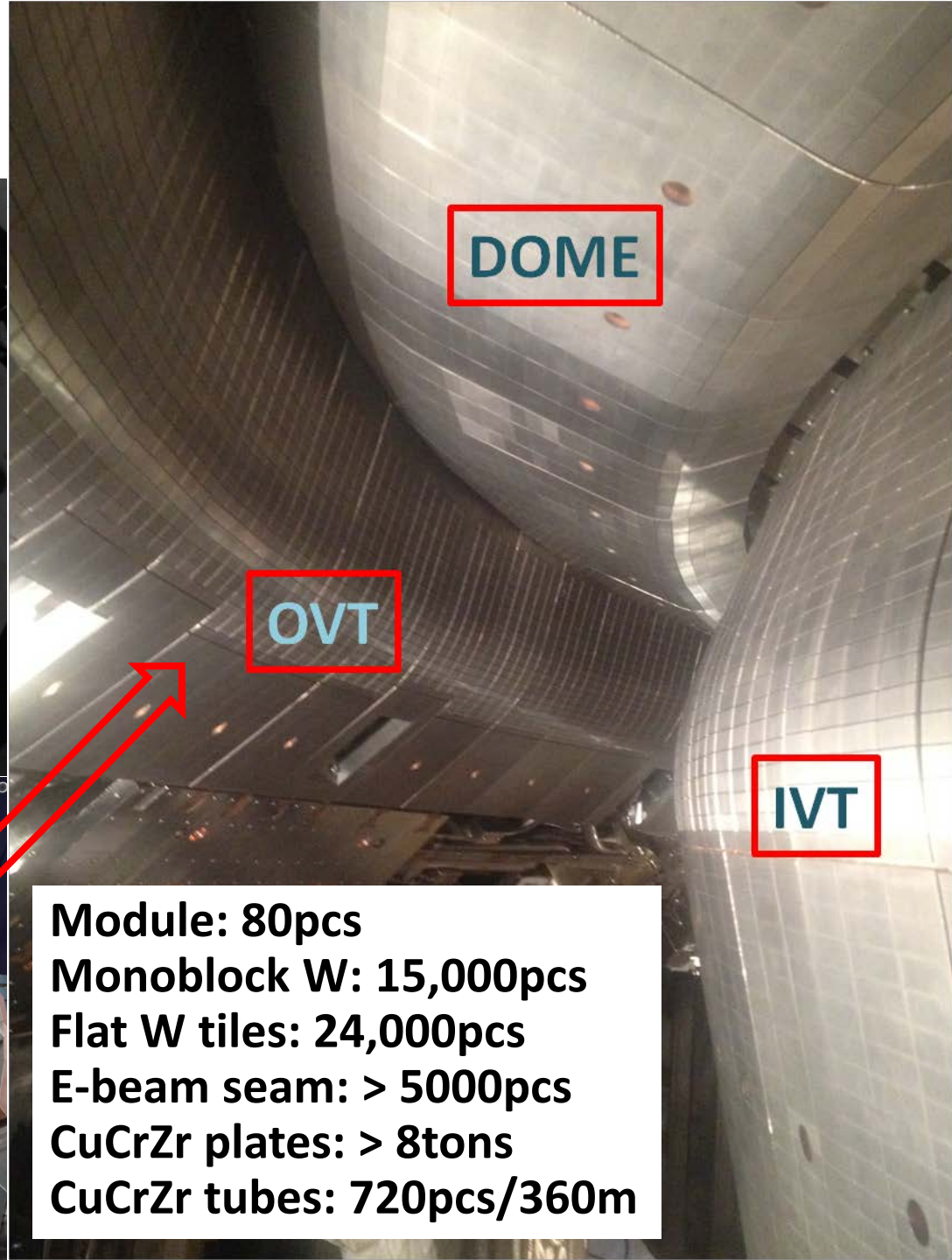


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Assembly



DOME

OVT

IVT

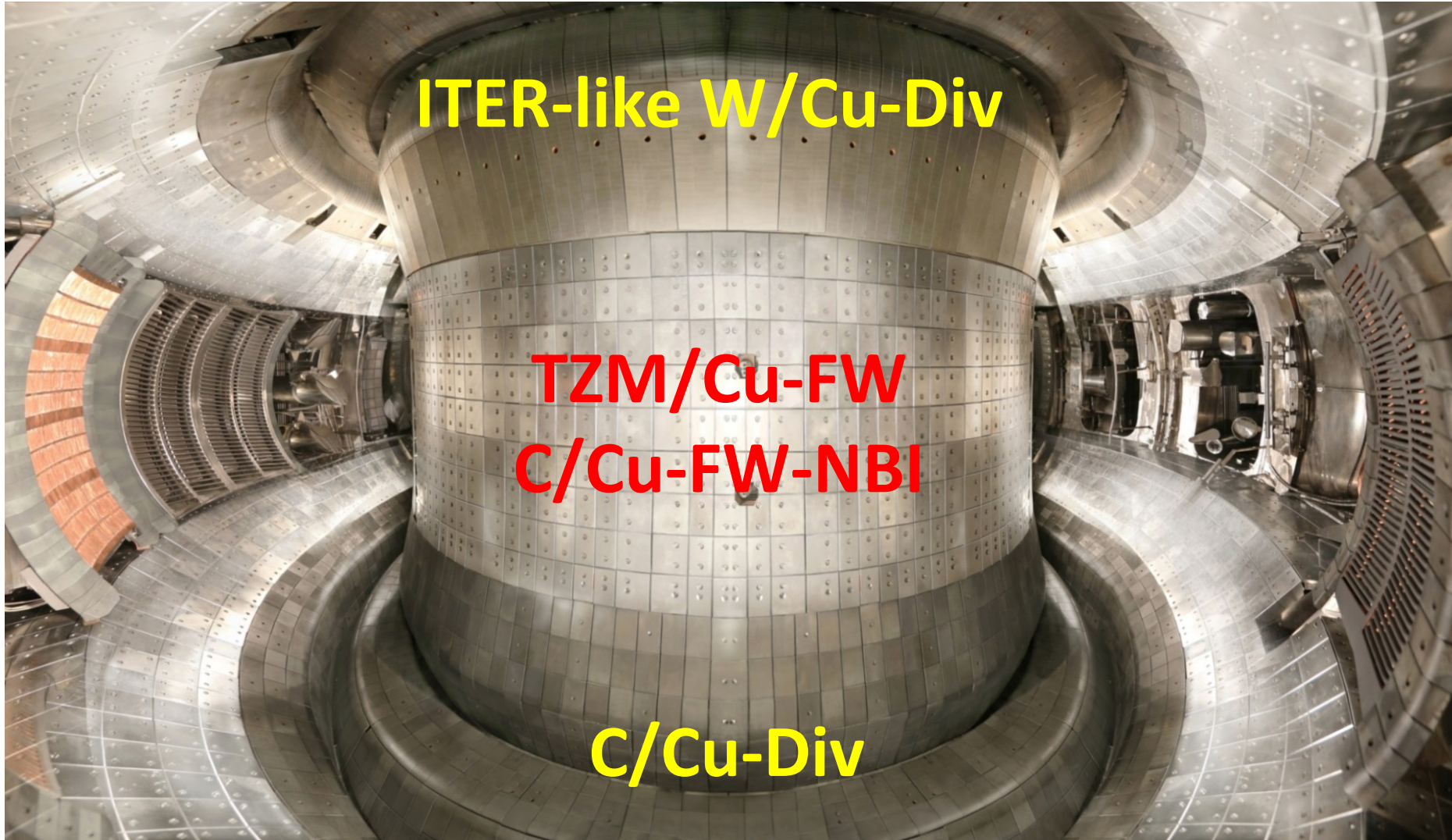
Module: 80pcs
Monoblock W: 15,000pcs
Flat W tiles: 24,000pcs
E-beam seam: > 5000pcs
CuCrZr plates: > 8tons
CuCrZr tubes: 720pcs/360m



Grandview of the whole PFMC



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W/Cu divertor on EAST

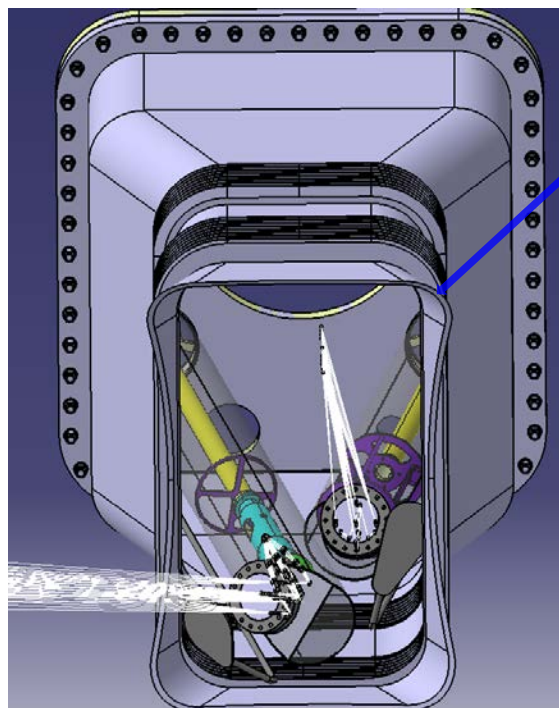
- Commissioning in 2014



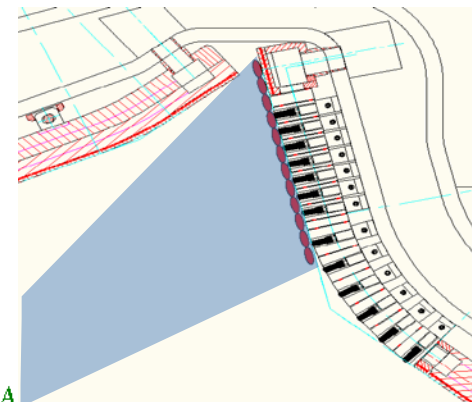
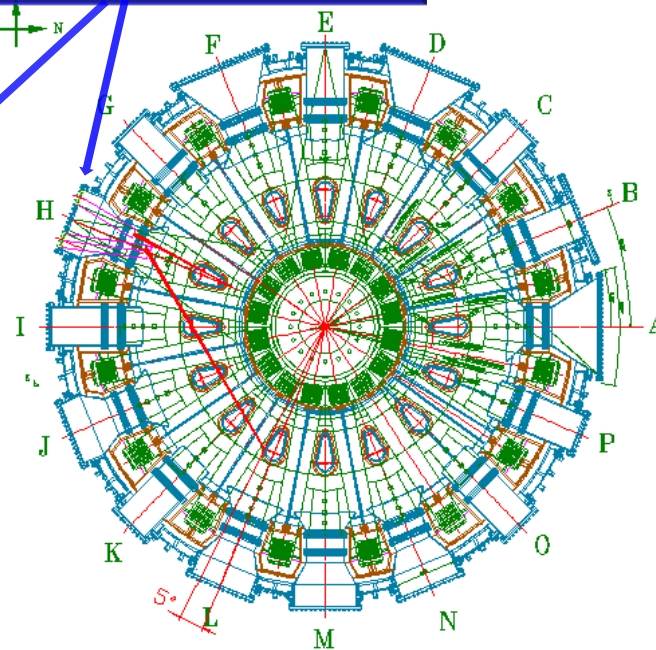
Monitoring W source on EAST



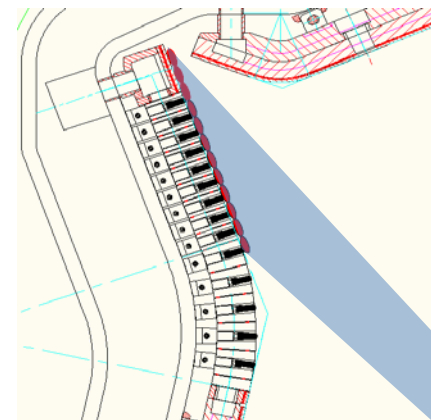
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H mid-plane port



Outer target plane
(H section)



Inner target plane
(L section)

Spectroscopy system: Spectrometer & Filter + PMT

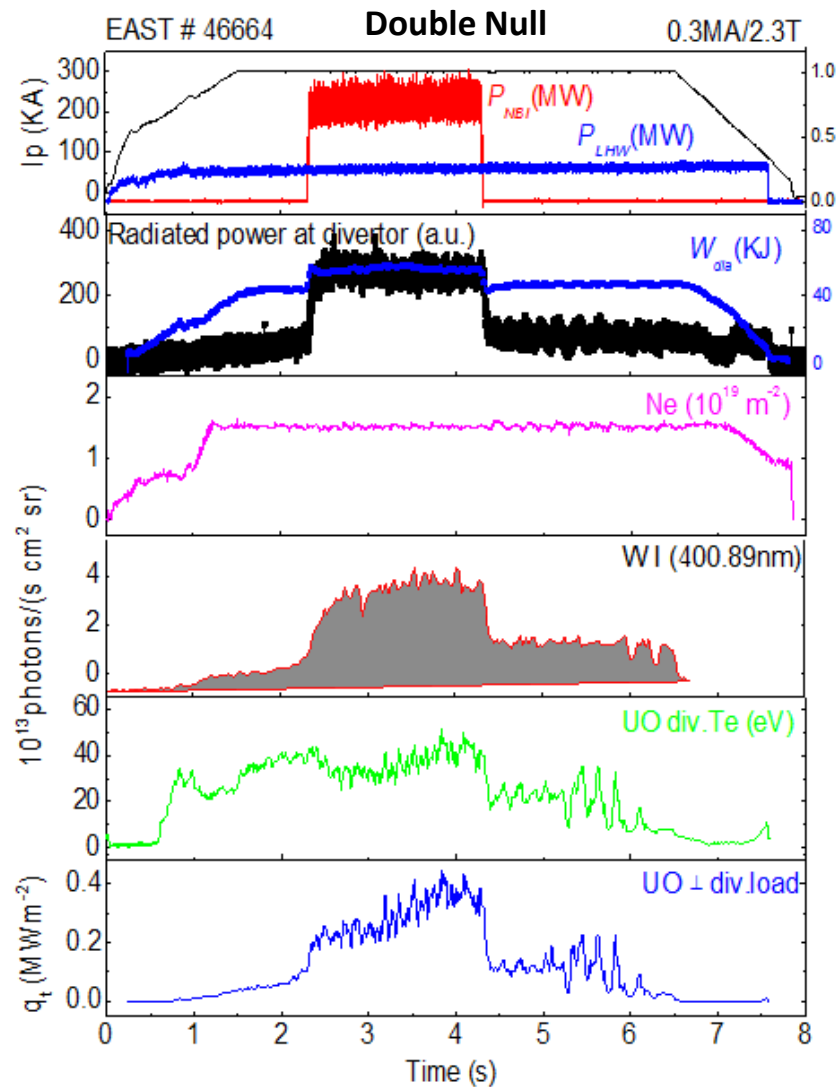
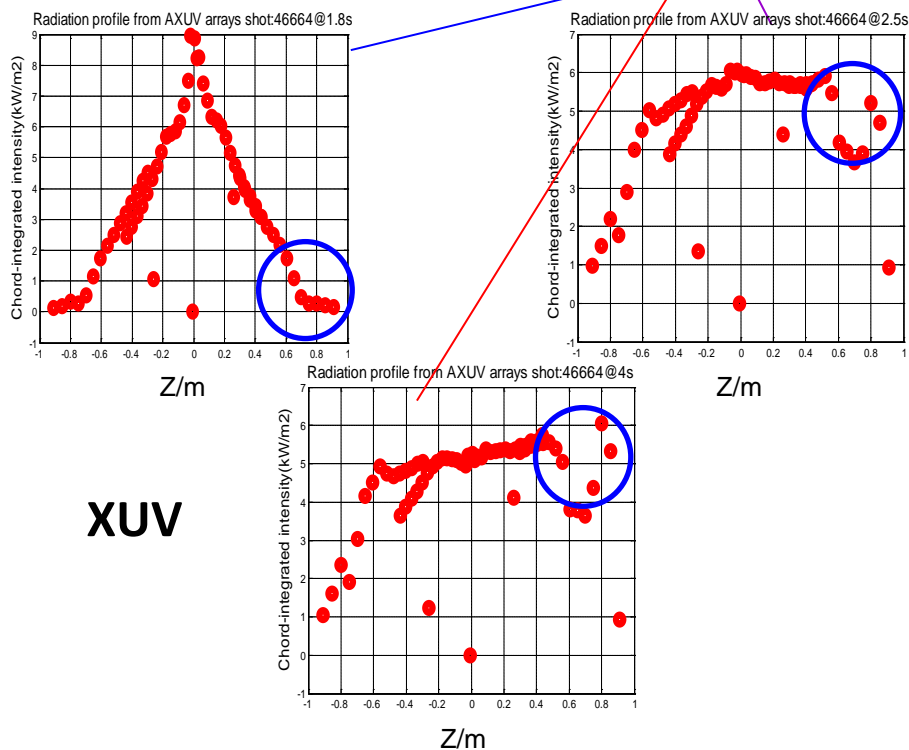
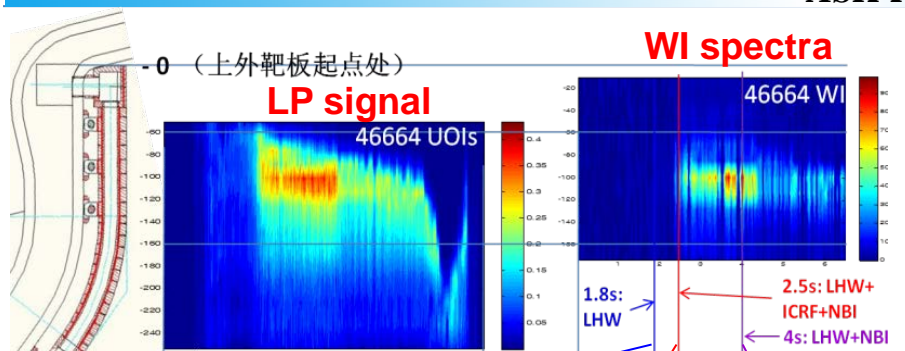
- 350nm - 800nm
- 300 / 600 / 1200 / 1800 / 3600 l/mm gratings
- Spatial resolution: 13 mm (poloidal)
- Time resolution: 200 Hz (12 channels, EMCCD)
Max. 20 kHz (PMT)



W source during NBI heating



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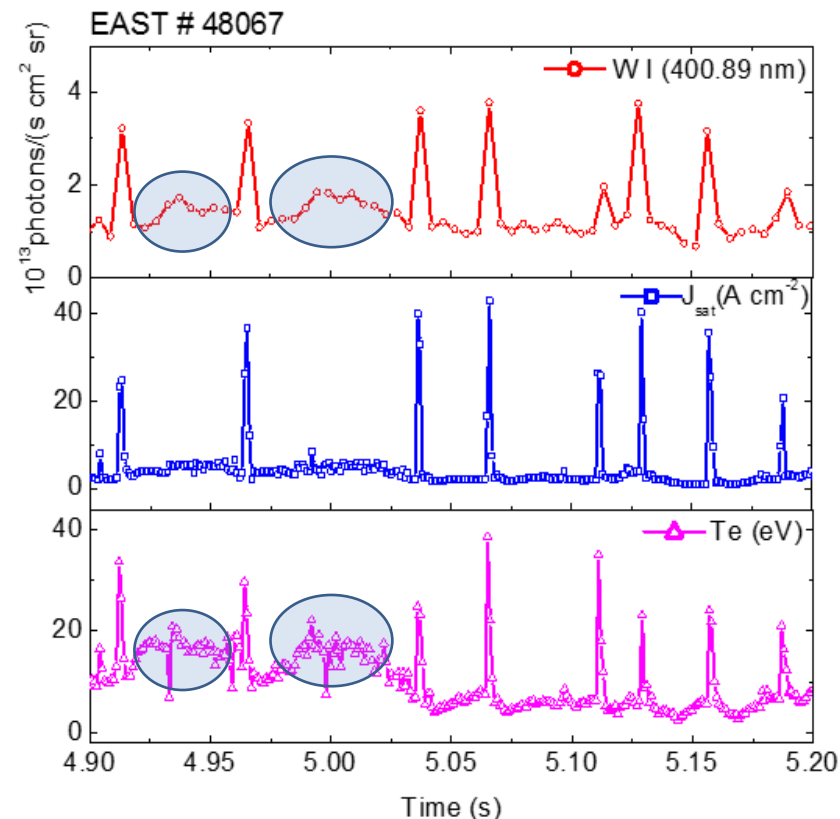
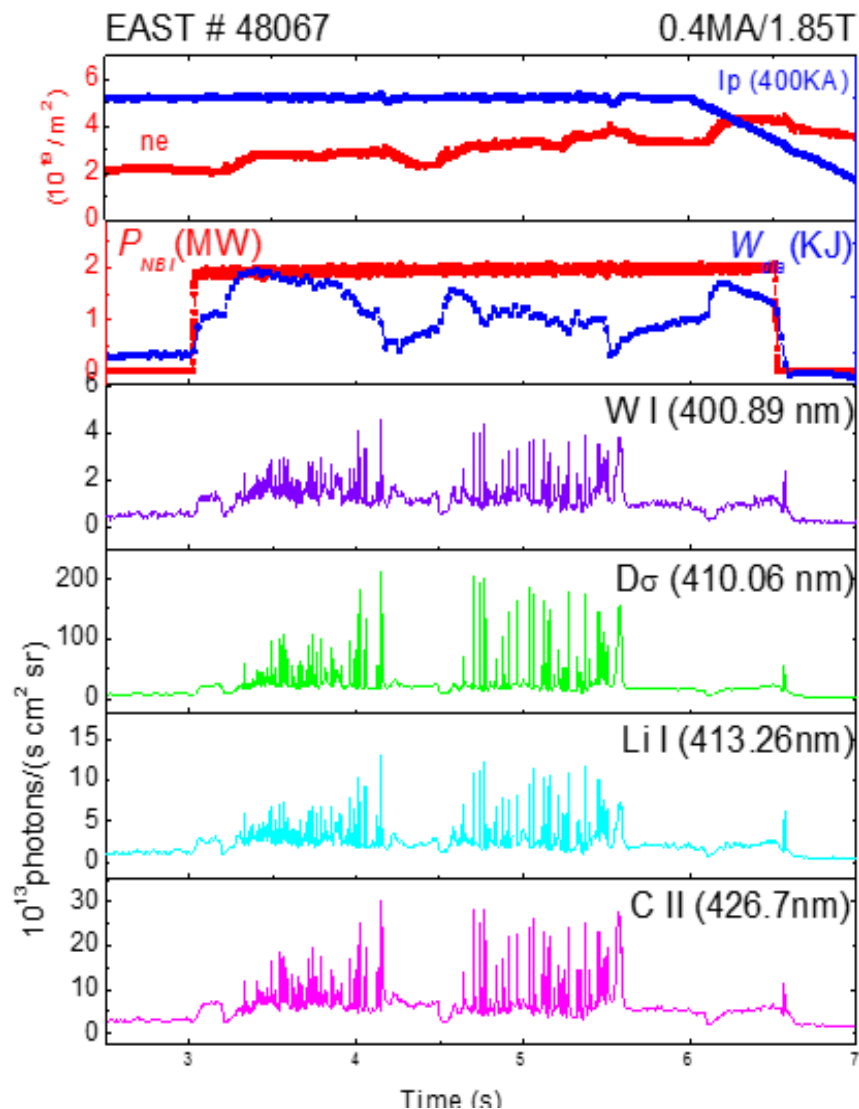




W source in type I ELMy H mode plasma



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400KA, LSN, NBI L+R 3MW

A strong W source appears during type I ELMy, together with other impurities including C, Li, etc., and inter-ELM vs Te

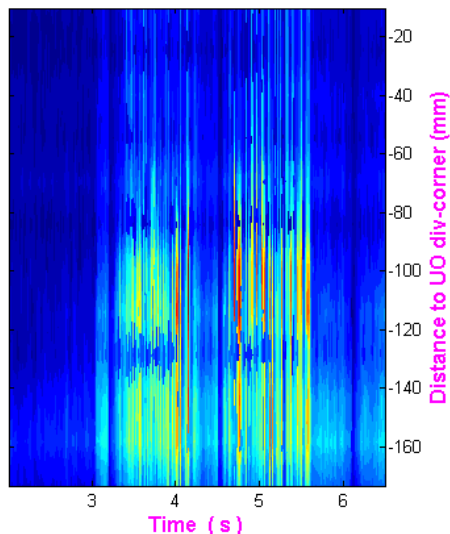


W source in type I ELMy H mode plasma



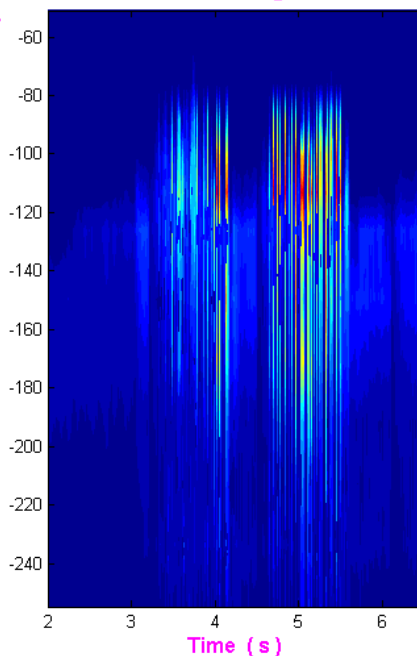
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W I (10^{13} photons $s^{-1}cm^{-2}sr^{-1}$) Shot #48067

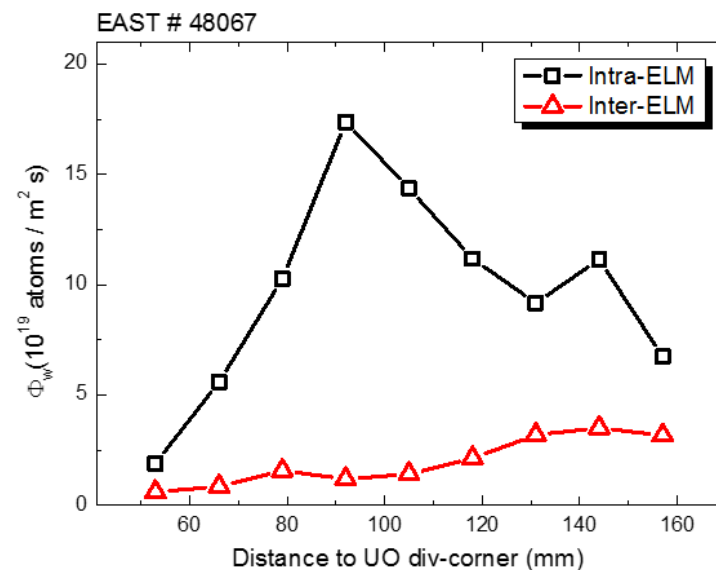
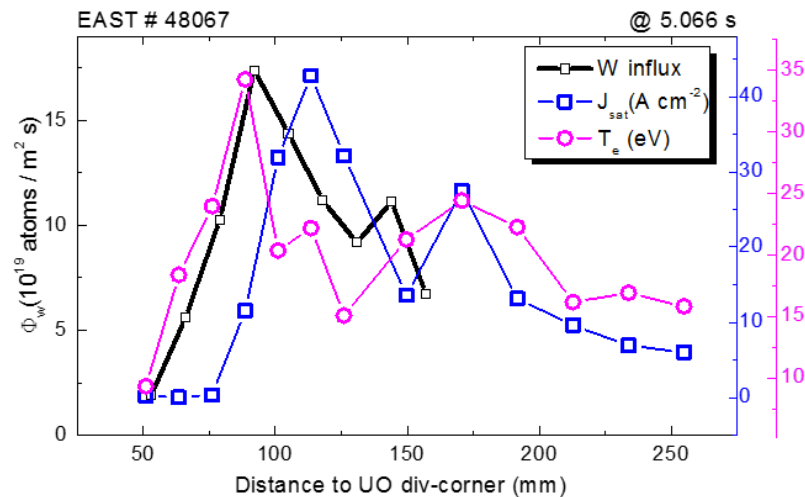


EAST # Shot :48067

UO Js ($A cm^{-2}$) @ Port D



Poloidal distribution of intra-ELM W influx presents a significant increase at strike point region, corresponding to T_e distribution at divertor.





Problems in 2014 campaign



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- Twice leak events during baking at 170°C, and first repaired and second evacuated (given-up).
- Baking at the end of campaign at 270°C to test PFCs and find put weak points (possible leaks).
- After campaign, 13 leaks in total were found, all related to e-beam welding (improper joint/connection design and insufficient non-destructive testing).
- Some W tile damages observed due to assembly issue.

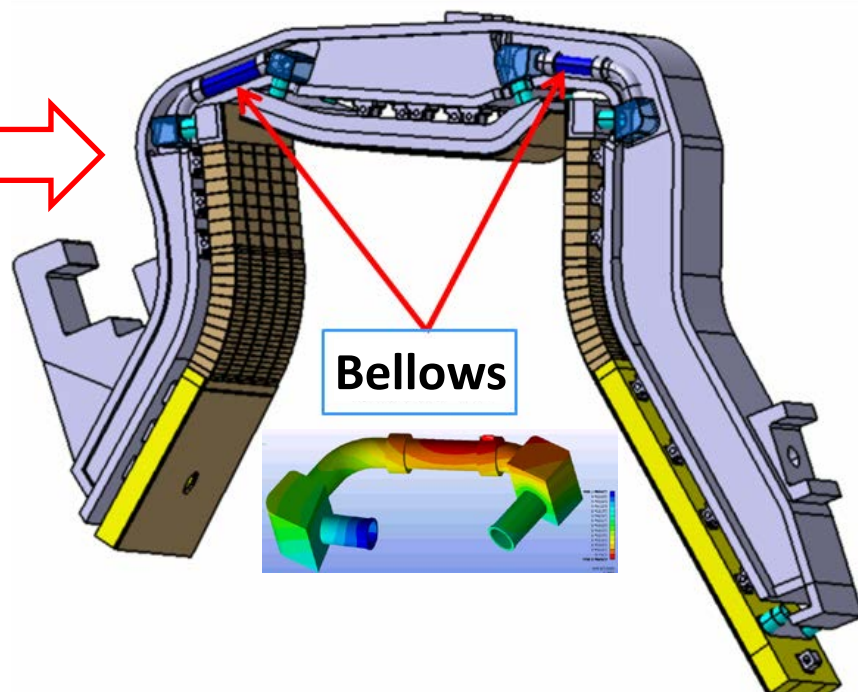
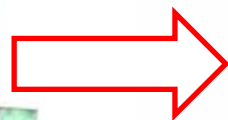
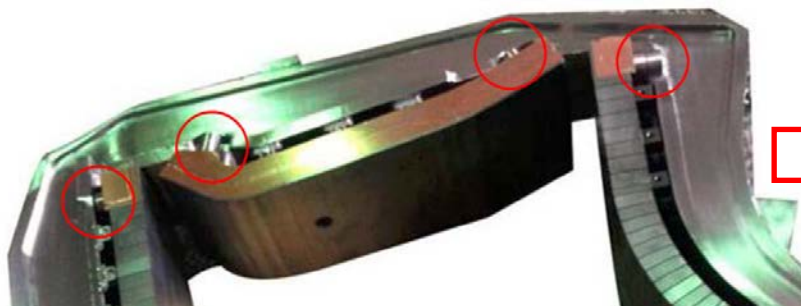




Repair/Optimization in 2014/5

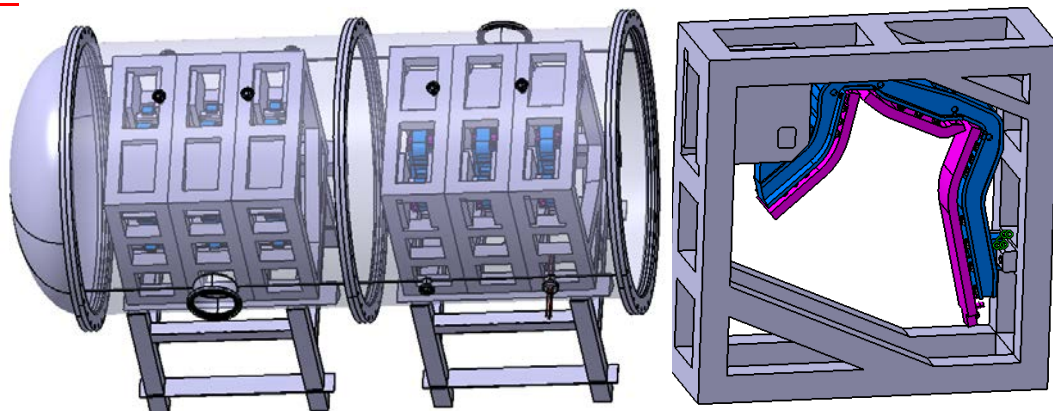


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Three measures for leak problem:

- Cooling tube connection:
TIG welding -> soft bellows
- QA&QC for EBW btw tube
and heat sink enhanced
- PFCs+CB baking at 250°C
prior to assembly in VV





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W/Cu divertor on EAST

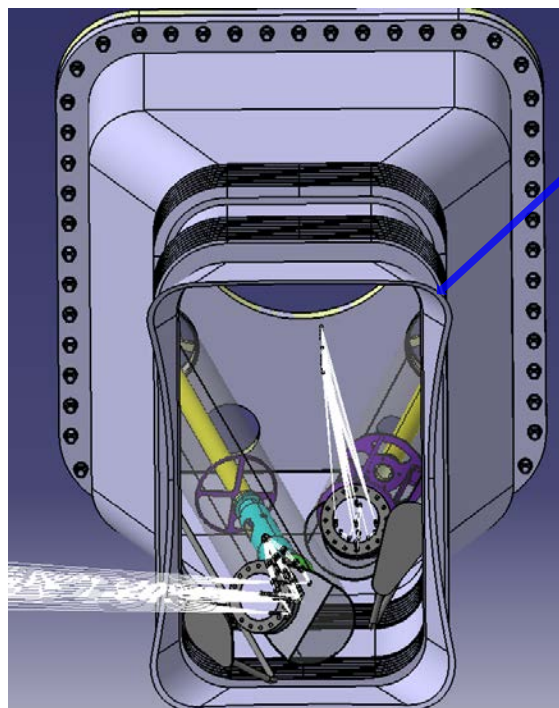
- Commissioning in 2015



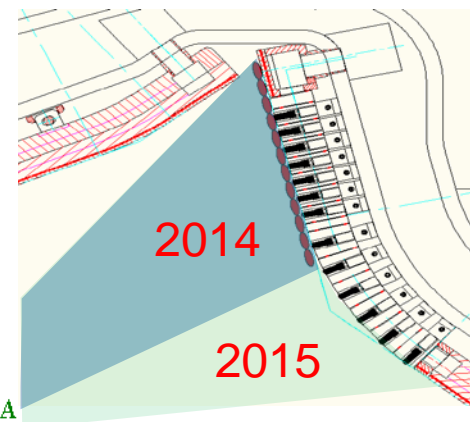
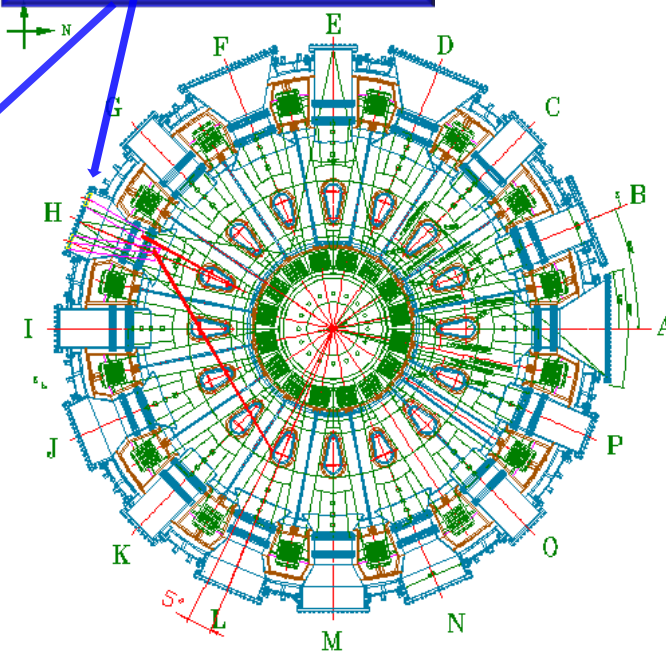
Monitoring W source on EAST



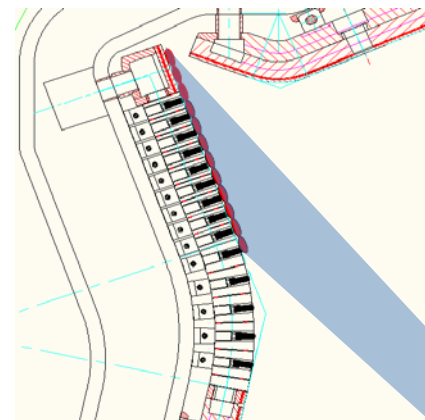
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H mid-plane port



Outer target plane
(H section)



Inner target plane
(L section)

Spectroscopy system: Spectrometer & Filter + PMT

- 350nm - 800nm
- 300 / 600 / 1200 / 1800 / 3600 l/mm gratings
- Spatial resolution: 13 mm (poloidal)
- Time resolution: 200 Hz (12 channels, EMCCD)
Max. 20 kHz (PMT)

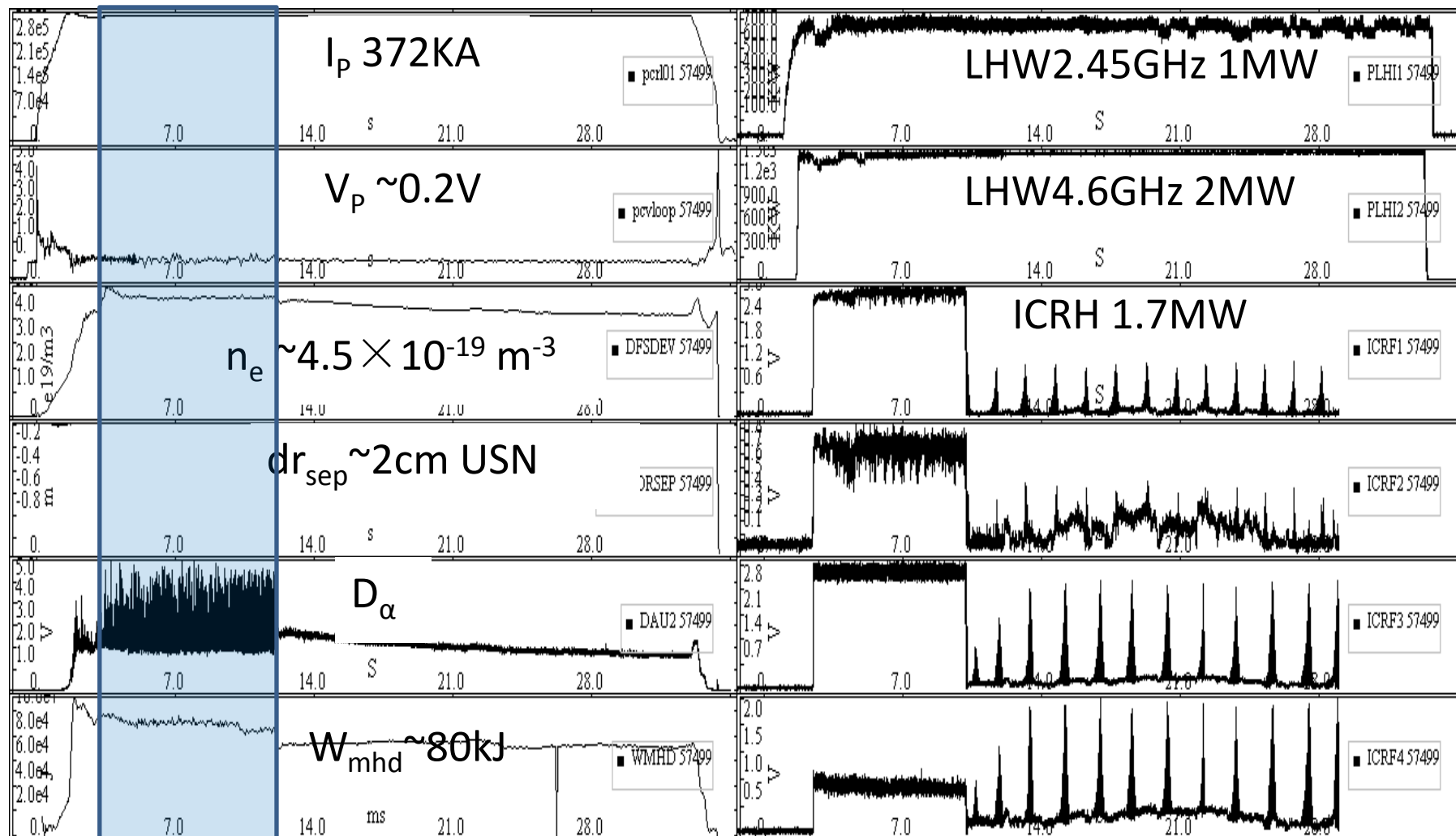


Shot 57499 (34s, H-mode/10s)



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Type III ELM



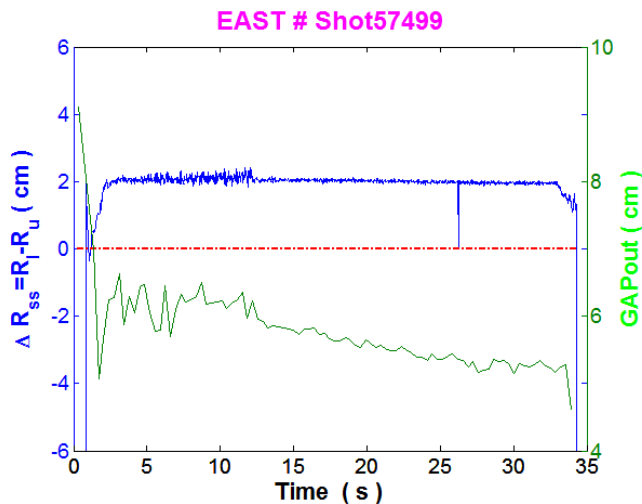
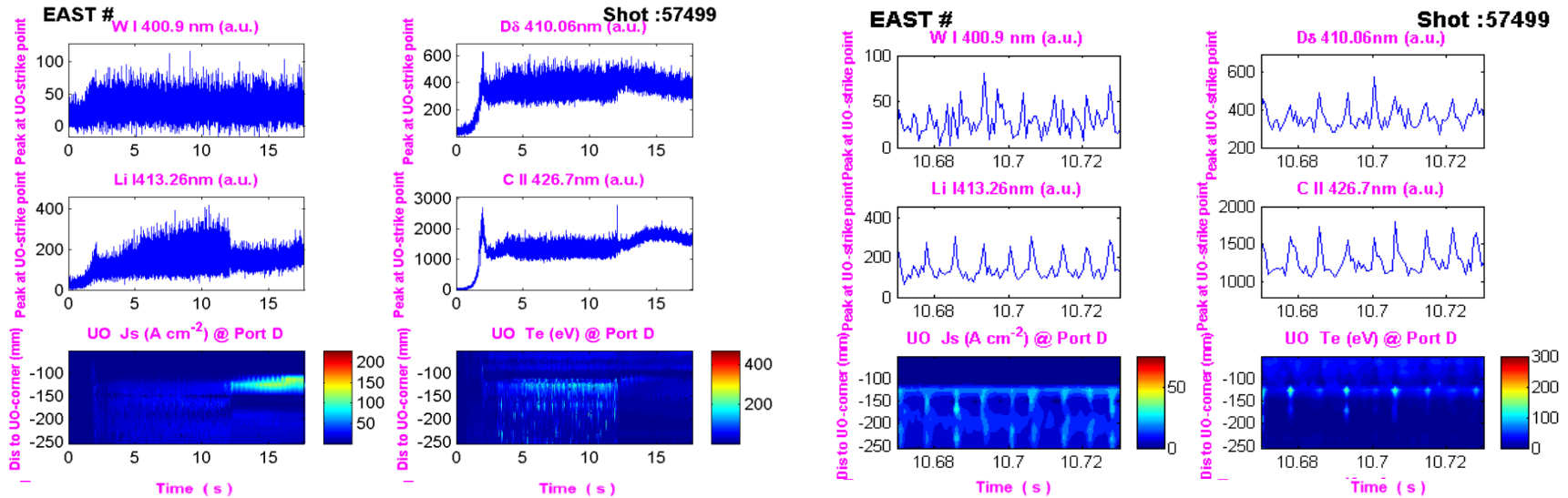


Shot 57499 (34s, H-mode/10s)

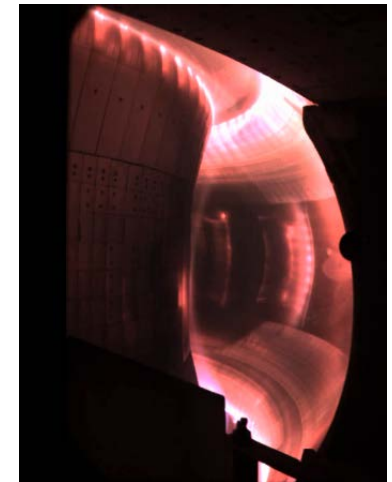
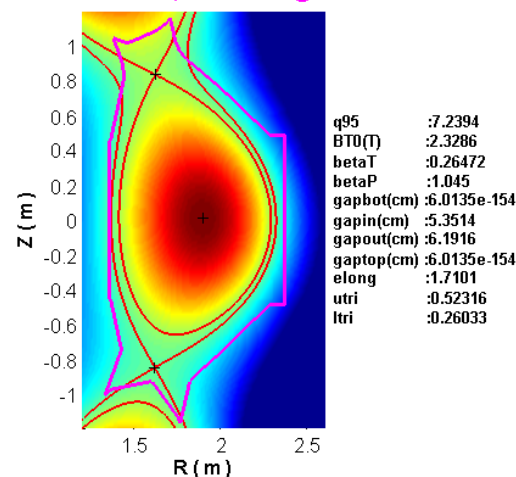


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Impurity data comparison btw spectroscopy and LP @ SP/OVT during H mode)



Shot#57499 poloidal flux @10.8s





Engineering performance



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- No leak till the end of the Spring campaign
- Survival from 200°C baking and discharges
- Thick Li coating existing !
- Careful cleaning of Li
- No significant damages found on PFMC !





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D retention & permeation
- W irradiated by heavy ions



Objectives & approaches

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Objectives

- To clarify the D retention mechanism in bulk damaged W
- To investigate the D permeation behavior through bulk damaged W

✓ W foils (1373 K, 2 h)

$\Phi 20 \times 0.11$ mm, $\Phi 20 \times 0.05$ mm

✓ HIRFL

122 MeV $^{20}\text{Ne}^{+7}$, 3.0×10^2 ions/m², ~713 K

✓ Defects characterization

PALS, DBS

✓ Deuterium retention

Gas exposure, TDS

✓ Deuterium permeation

Gas-Driven Permeation



Irradiation chamber with an energy degrader and cooling/heating stages at HIRFL-SFC.

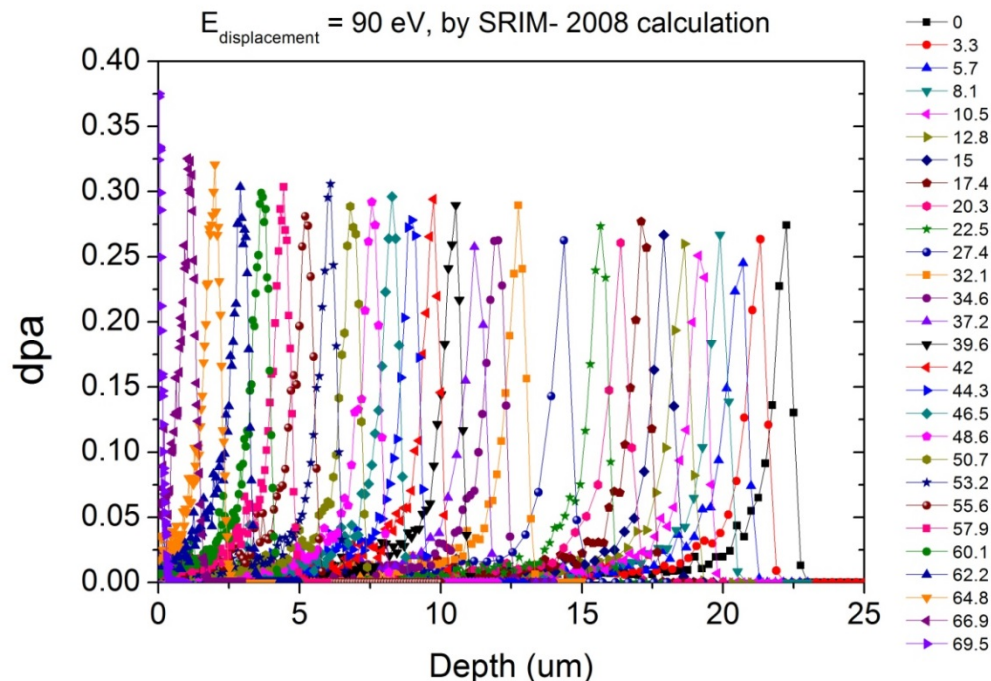


High energy heavy ions irradiation

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➤ Uniform distribution of irradiation defects in W

- High energy $^{20}\text{Ne}^{+7}$ ions (122 MeV), incident range is $\sim 22.0 \mu\text{m}$ in W;
- An energy degrader wheel made of 30 pieces of Al-foils revolving in front of the sample;
- A sample-double-side-irradiation mode.



A quasi-homogeneous distribution of atomic displacement damage to 0.3 dpa within a depth of $50 \mu\text{m}$ in W.

Estimated depth profile of displacement damage at one side of W. The right column shows the thickness (μm) of Al-foils used in the energy degrader.



Positron annihilation lifetime spectroscopy (PALS)

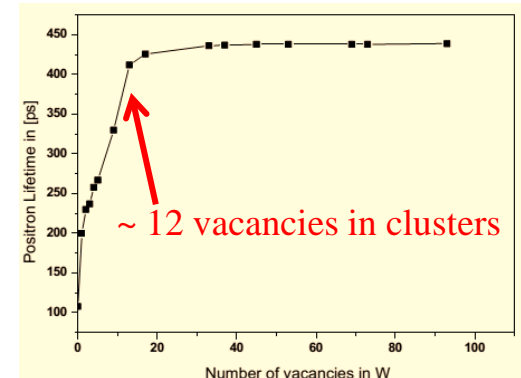
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- The measured PALS were analyzed based on a two-state trapping model by assuming:
- τ_1 - free positrons and positrons trapped at the mono-vacancies/dislocations,
 - τ_2 - positrons trapped at the vacancy clusters or voids.

Dislocations

Large vacancy clusters !

	Thickness of W (μm)	Lifetimes 1 (ps)	Intensities (%)	Lifetimes 2 (ps)	Intensities (%)
Annealed (1373 K)	50	139.6 ± 1.7	100	---	---
	110	139.5 ± 2.6	100	---	---
Irradiated	50	118.6 ± 3.4	59.5 ± 1.0	406.8 ± 5.2	40.5 ± 1.0
	110	118.9 ± 3.6	62.7 ± 1.3	366.4 ± 6.1	37.3 ± 1.3



Reference*:

- positron lifetime in defect-free W : $\tau_f = 105$ ps
- Correlation: positron lifetime vs.. the number of vacancies clusters in W (right figure)
- The lifetimes of positrons trapped at dislocation: slightly shorter than τ_{1V}

n-irradiated W/Mo:
 $\tau_2 = 350 \sim 470$ ps

* H.E.Schaefer, phys. stat. sol. (a) 102, 47 (1987);

T. Troev, E. Popov, N. Nankov, et al., J. Phys.: Conf. Ser. 207 (2010) 012033

M. Eldrup, X. Meimei-Li, L.L. Snead, et al., Nucl. Instr. Meth. B 266 (2008) 3602–3606

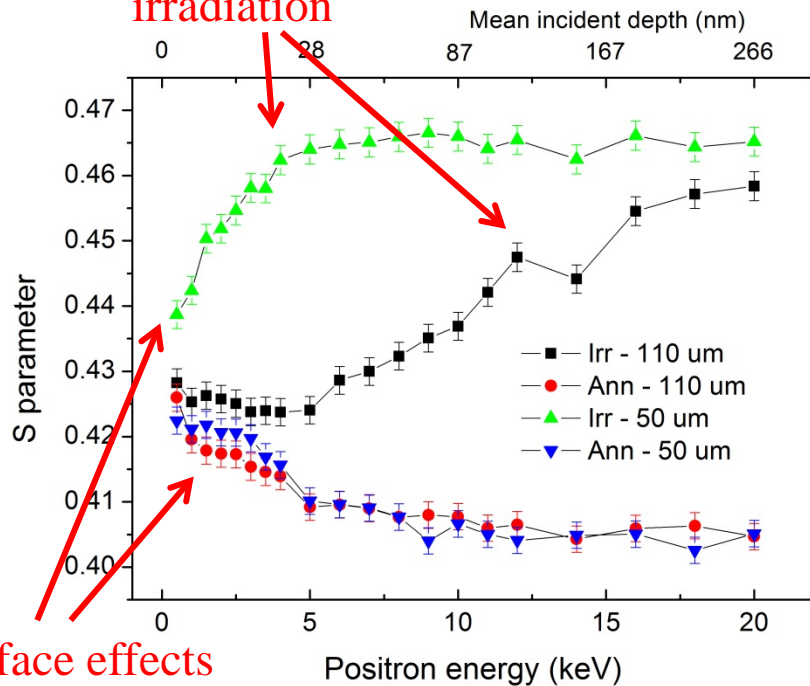


Doppler broadening spectroscopy (DBS)

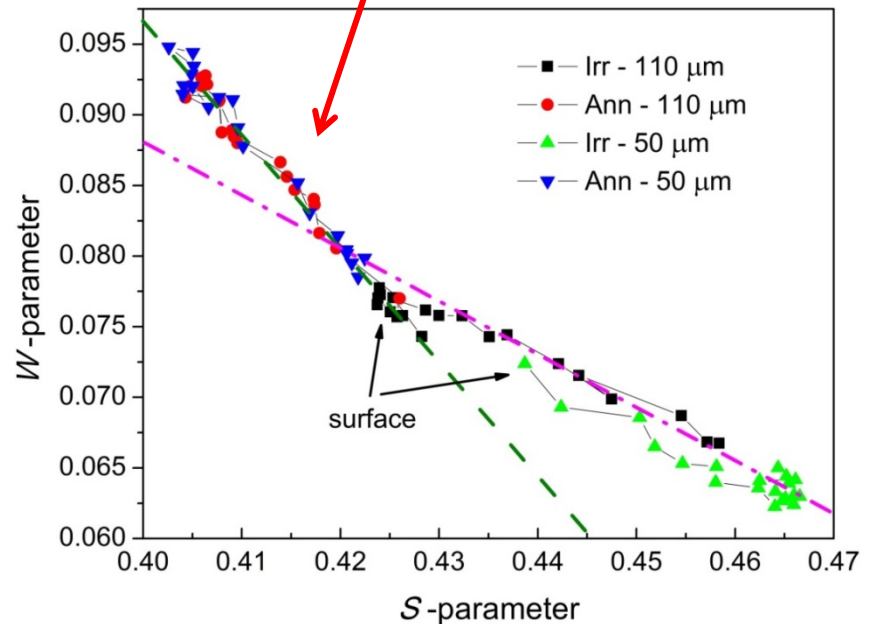
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- Slow positrons: 0.18 -20 keV, $R_{max} \approx 0.5 \mu m$ → near surface information

Increase of S parameter after irradiation



Change in slopes



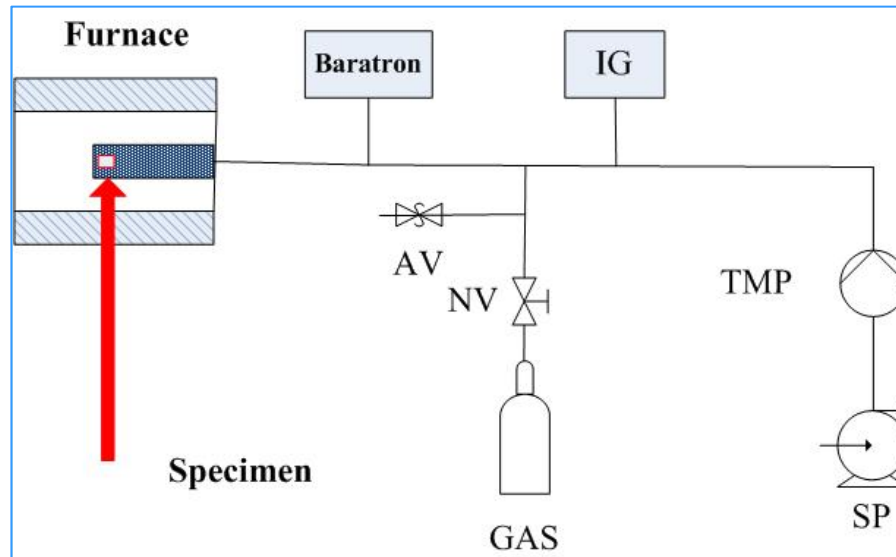
- ✓ Larger S parameter in S(E) plot & larger slope value in S-W plot for two irradiated tungsten: strong evidence of vacancy-type clusters formation under Ne-ion irradiation;
- ✓ Defects quasi-homogenously distributed at the near surface.



Deuterium bulk retention (D_2 gas loading)

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- D_2 gas loading to introduce an uniform distribution of D in the damaged W
(instead of plasma exposure: to avoid further modification of near surface structure under plasma irradiation & larger D retention in the plasma irradiated side of the damaged W)
- Focus on high temperature D bulk retention behavior



Sketch of the D_2 gas loading facility

Loading conditions:

773 K /1.3 bar;

Holding time: 2 h

(long enough to ensure the diffusion length of D > sample thickness)

Samples quick cooled by liquid

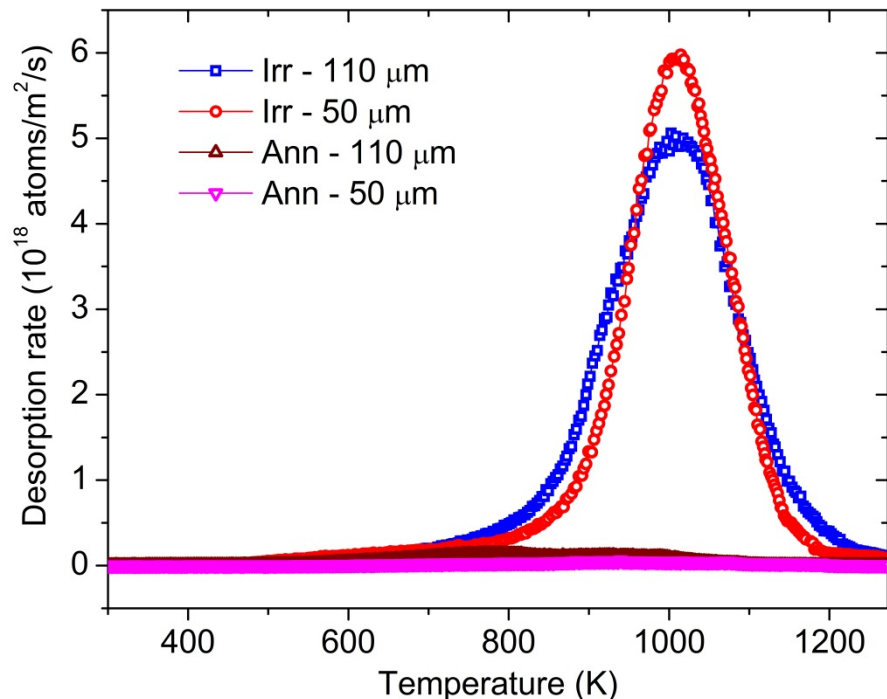
N_2



Deuterium bulk retention (TDS analysis)

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TDS: 1 K/s to 1173 K (infrared heating)



TDS spectra of D for annealed and Ne-ion irradiated W after D₂ gas exposure at 773 K for 2 h.

- ✓ **A high desorption peak at about 1010 K only for two irradiated W**
~1050 K for n-irradiated W [Hatano,et al]
- ✓ **Broad D desorption spectra from 730 K to 1173 K for the Ne-ion irradiated W**
 - i. 730->1173 K for the n-irradiated W
 - ii. 930-1050 K for the single energy W-ions irradiated W [Hatano,et al]
- ✓ **Trapping energy: ~1.8 eV** ; nano-voids/large vacancy clusters formed by cascade collisions or the clustering of mono-vacancies during Ne-ion irradiation

High-energy Ne-ion irradiation together with an energy degrader for tailoring the damage distribution may be an effective way to simulate defects production in bulk W under neutron irradiation.



Deuterium permeation through irradiated W

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➤ Building a GDP & TDS device at ASIPP

Downstream dual QMS system

TDS furnace & sample



Upstream gas supply & gauge

GDP furnace & gasket-type sample

Parameters for GDP device

- Background pressure: 10^{-5} Pa
- Driven D_2 pressure: 10^3 - 10^5 Pa
- Sample temperature: RT-1000 K
- Sample sealed by VCR couplings
- Signal monitored using QMS

We tried to study the D permeation behavior through irradiated W with the GDP device. Unfortunately, the irradiated W was very brittle and irradiated samples were limited, we **failed to seal the samples**.



Effects of non-irradiated defects on D permeation

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Irradiated W samples failed at sealing, then we focused on the GDP of non-irradiated W specimens which possess different defect types and densities through thermal treatment.

Materials:

- Rolled W : 973 K/2 h
- Annealed W: 1373 K/2 h
- Recrystallized W: 1673 K/2 h

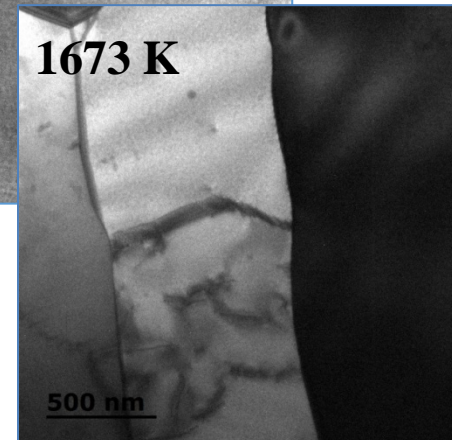
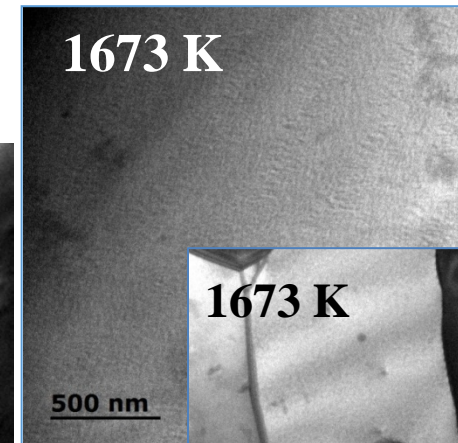
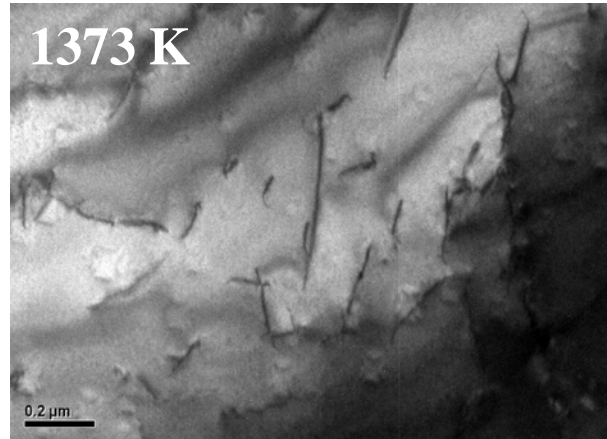
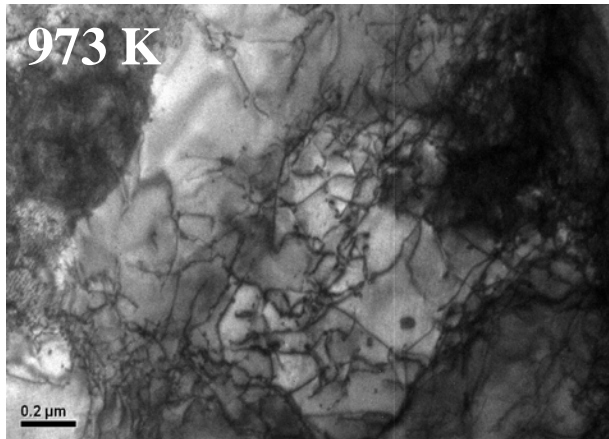


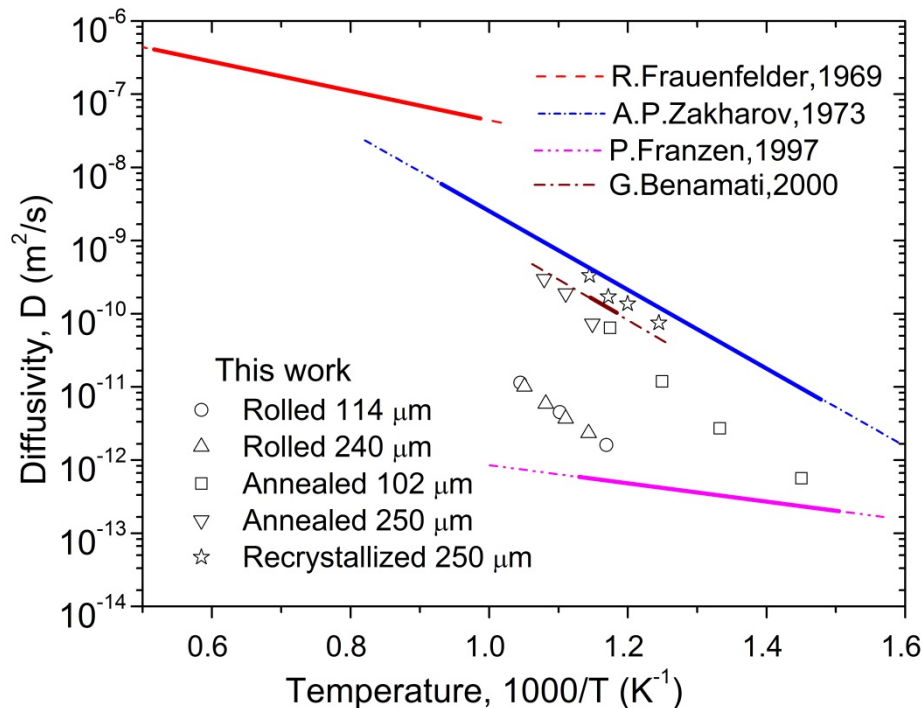
Fig. Bright field TEM observation of rolled/annealed/recrystallized W.

- Rolled W: a dense network of dislocations;
- Annealed W: dislocations fragmented into shorter individual lines or disappeared;
- Recrystallized W : large areas free of dislocations.



Diffusion coefficient of D in various W foils

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Diffusivity of D in rolled/annealed/recrystallized W compared with data from literatures

$$J = -D_L \frac{dc_L}{dx} \quad (\text{lattice diffusion})$$

$$J = -D_{eff} \frac{dc}{dx}, \quad (c = c_L + c_t)$$

(diffusion with trapping sites)

$$D_{eff} = \frac{D_L}{1 + \frac{n_T}{n_L} \exp\left(\frac{E_t}{RT}\right)}$$

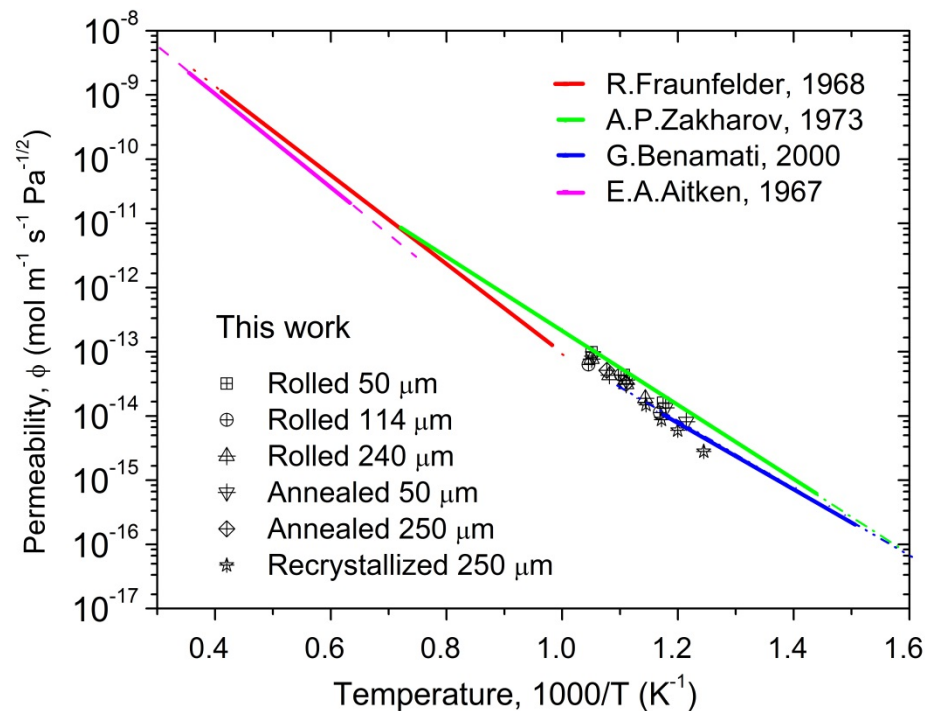
[Oriani' work, 1970]

- ✓ The higher the density of defects (n_T), the lower the effective diffusivity (D_{eff}). Thus the measured diffusivity of D in rolled W, annealed W, recrystallized W increases one by one.
- ✓ If there are trapping sites, $D_{eff} < D_L$; only when the experimental temperature is high enough, the trapping effects can be neglected, then $D_{eff} \approx D_L$.



Permeation coefficient of D in various W foils

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- ✓ Invariable with the dislocation density;
- ✓ Comparable to the literature data which were measured by hydrogen gas-driven permeation method.

Permeability of D in rolled/annealed/recrystallized W compared with data from literatures.

The defects in non-irradiated W does not significantly affect the cross section of D diffusion fluxes, hence the permeability of D is almost not affected by the defects in W.



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Summary and outlook



Summary & Outlook



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- EAST upgraded its upper divertor into full W/Cu-PFCs, employing HIP technology and e-beam welding in 2014.
- Commissioning in 2014 found leaks and damages, and after repairing and optimization, commissioning in 2015 was successful to the end of the Spring campaign.
- Plasma-tungsten interactions (PWI) have been studied in 2014-5 campaigns via monitoring the upper divertor by means of spectroscopies and Langmuir probes.
- A quasi-homogeneous atomic damage (mainly large vacancy clusters) achieved to 50 μm deep in W by heavy ion irradiation.
- High D desorption peak and broad D desorption spectra, similar to that from n-irradiated W, have been observed.



Thanks for your attention!
Welcome to collaborations!