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H isotope retention in Be and Be mixed materials

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Introduction: Challenges for ITER and beyond



Fusion reaction:T(D,α)nCentral Temperature:20 keVTypical Density:10²⁰ m⁻³Fusion Power:500 MWPlasma Volume:840 m³Plasma Current:15 MA

700 m² Be first wall
100 m² Tungsten divertor
50 m² CFC strike point



Nuclear Fusion Reaction









The role of material walls





1. VACUUM CONDITIONS

Unlike the sun, a fusion plasma can only be maintained under ultra high vacuum conditions base pressure $\approx O(10^{-8} \text{ mbar})$

2. EXTRACTION OF POWER

The α -particle power and auxiliary injected power used to heat the plasma must be finally extracted through the plasma facing wall

Power carried by neutrons is converted to heat in blanket wall neutrons also breed tritium in blanket

3. HELIUM REMOVAL

The removal of the helium ash requires thermalisation and neutralisation of plasma ions



ITER: 400 MW in neutrons,
100 MW from alpha particle heating
about 50 MW auxiliary heating
→ about 150 W heating power to walls
About 50 % go to divertor, rest to main chamber walls

Main chamber wall:

700 m², about 450 to 500 MW, about <1 MW/m²

Divertor:

About 75 MW Radial extend of power carrying layer O(cm) Inclined target plates increase strike area by about a factor of 6

→ Mean power flux density at divertor strike areas in ITER: 10 MW/m²

(in transient events [ELMs, disruptions] up to factor of 10-100 higher)

 \rightarrow Radiative cooling to distribute power over a larger area

Requires development of appropriate plasma-facing components

For comparison:Hot plate0.05-0.1 MW/m²Oxy-acetylene torch100 MW/m²

Particle fluxes







Material erosion – life time of components and plasma impurities

Transport and redeposition of eroded material

Tritium inventory (in particular in redeposited layers)

Handling of power fluxes (plasma-facing components)

Neutron damage and activation of materials

Physical sputtering yield and harmfulness of elements in the plasma are main selection criteria for first wall material

PSI/PMI processes



- Physical Sputtering
- Chemical Erosion
- Radiation Enhanced Sublimation
- Photon Induced Desorption
- Evaporation & Sublimation
- Brittle destruction
- Melting & Splashing
- Arcing
- Neutron Induced Damage

Erosion by energetic D ions: physical and chemical sputtering



For beryllium and tungsten theoretical and experimental yields agree very well

Carbon shows additional erosion with only weak dependency on impact energy

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Material erosion under extreme power load





FOR METALS: Splashing Formation of droplets Formation of dust





FOR CARBON: Above a certain power load (threshold) emission of debris **₽→ BRITTLE DESTRUCTION**





4 main topics:

Surface Processes

Tritium inventory

Migration in fusion devices

Materials and components

Devices:

Tandem Acc. Laboratory

AUG, TS, ITER

AUG, JET

High heat flux test facility, GLADIS, JET, W7-X

I. Surface Processes



Formation of mixed materials:

Here: Be – W – O system

- Investigation of surface alloys and mixed phases by XPS
- Lab experiments and experiments at BESSY (Berlin Electron Syncrotron)
 - Chemical information from peak position
 - Compositional information from intensity
 - Depth information from variation of photon energy
- Which chemical phases do form?
- What are reaction temperatures?
- What is the role of diffusion?
- How do these phases interact with the boundary plasma?



II. H retention and release in pure Be

Prior work:

Be – W system and pure Be

- PhD thesis work of Matthias Reinelt
 - H retention in Be single crystals
 - TMAP simulation of desorption spectra
- PhD thesis work of Martin Oberkofler (just finished)
 - > H retention in polycrystalline Be
 - TMAP simulation of desorption spectra

Reinelt et al., New Journal of Physics 11 (2009) Oberkofler et al., Nucl. Instrum. Methods 267 (4) (2009)

low flux/fluence implantation at 0.3-3keV: retention saturates above 10¹⁷D cm²

change in release pattern: low fluence: only high-T release

high fluence: + low-T release





II. H retention and release in pure Be

IPP

Reinelt et al., New Journal of Physics 11 (2009) Oberkofler et al., Nucl. Instrum. Methods 267 (4) (2009)

low flux/fluence implantation at 0.3-3keV with 10¹⁵D cm⁻² s⁻¹:





Formation of mixed materials:

Here: Be – W – C system

- Some model systems as themodynamically stable phases (Be₂C and Be₁₂W)
- Deposition of mixed material films with arbitrary composition in Romania (MEdC, C. Lungu et al.)
- Characterisation at IPP
 - Ion beam analysis
 - > XPS
- H implantation in HCS
 600 eV D₃⁺ (200 eV/D)
- Thermal release (TESS)

Thermionic Vacuum Arc deposition device



C. P. Lungu, I. Mustata, V. Zaroschi et al., Phys. Scr. T128 (2007) 157



Formation thermodynamically stable phases:

Be₂c

- Evaporation of thin C film onto Be substrate
- Annealing at about ? K

Be₁₂W

- Evaporation of thin W film onto Be substrate
- Annealing at about 1200 K







Thermal desorption spectra of D from:

(a) Be and compounds (Be12W and Be2C),

(b) Be and Be-W mixed deposited layers (W concentrations of ~10 % and ~60 %)

(c) Be and Be-C mixed deposited layers (C concentrations of \sim 8 % and \sim 50 %).

For pure Be and Be₂C samples the temperature was held for 20 min. at 623 K

D implantation to samples was performed at room temperature.



Thermal desorption spectra of D from Be and Be₂C layers as a function of time





Thermal desorption spectra of D from Be layers implanted at different temperatures (RT, 423 K and 573 K).





Remaining fraction of D derived from TDS spectra of

(a) Be-W system,
(b) Be-C system
(c) Be implanted at different
temperatures (320 K, 423 K and 573 K).

Note that each curve is normalized to retention in pure Be implanted at RT.





D remaining fraction at 623 K (350 °C) in Be-containing samples as a function of each impurity (W or C) concentration in Be. Note that each number is normalized to retention in pure Be.



- Deuterium retention and release behaviour of Be-containing materials were investigated for the ITER wall (240 °C, 513 K) and divertor (350 °C, 623 K) baking temperatures
- In pure Be loaded with D at T < 340 K, D is predominantly released around 420-470 K within a relatively sharp desorption peak.
- Operation at elevated temperatures reduces the retained D amount, but the remaining D is less efficiently out-gassed at 623 K.
- Admixture of W or C changes the D release behaviour resulting in less efficient D removal by baking.
- Especially the presence of C in Be shifts the D release to higher temperature

Missing information:

- Release at increased holding times (so far max. 20 min)
- Release from thick films (so far only implanted films)

New experiments:

- Investigation of redeposited films produced in PISCES and magnetron sputtered Be/D films (D/Be ratios up to 0.3) with thickness up to 1 µm (Invited talk T. Schwarz-Selinger, PFMC-13)
- Influence of longer holding times at ITER temperatures (planned for 2011)





- ⇒ magnetron sputtered Be/D films are suited as a model system to study retention and release
- \Rightarrow release of D from Be at 240°C and 350°C has a very large time constant
- ⇒ even a several hours bake at 240°C or 350°C does not release all D (a D/Be of 0.8% or 0.2% remains, respectively) (deeper trap sites that cannot be drained)
- ⇒ codeposits grown in different ways (energies, growth rates) show similar release features
- ⇒ the total amount of D released above 350°C is the same for all codeposits investigated, equivalent to D/Be ≈ 1%
- ⇒ multilayer codeposits show the same release as single layer codeposits (BeO interface is no transport barrier)



- investigation of surface processes in ternary systems (Be W O, Be N)
- H retention in Be materials
- Modeling of Be release (TMAP and new code developments) (& collaboration with A. Allouche - DFT calculations of e.g. Be migration energies)
- D retention in mixed materials (increased holding times)
- Thick film model systems? (collaboration with MEdC Bucharest)
- MD simulations (W-Be-H potential, first tests performed)
- Global impurity transport modeling (Schmid /Reinelt) needs better input data

III. Needed data (personal view)



Sputtering yields (H, D, T, He, Ne, Ar, N, ...)

- energy dependence
- flux dependence
- ➢ fluence dependence
- ➤ temperature dependence

Compare plasma erosion vs. ion beams more individual ion species in plasma additional neutral species e.g., H₂ plasma: H⁺, H₂⁺, H₃⁺, & H⁰ laboratory plasmas: H⁰/H_{ion} \approx 100

Projectile enrichment in near surface layer (= retention) may lead to reduction of sputter yields (e.g., W/N) or enable chemical sputtering (e.g., C/H, enhanced yield!)

Don't forget possibility of impurity sputtering in plasma experiments

Redeposition in high flux plasma experiments requires proper modeling of net yields



Different types of redeposited layers

- Intensive characterization required
- Stoichiometry (IBA)
- Chemical states (XPS)
- Phases, crystalline phases, alloys, …
- Microstructure porosity, impurities, surface structure and morphology

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Mixed materials!